



Recent advances and breakthrough for SRF cavities at FNAL

Martina Martinello

12 March 2021

Outline

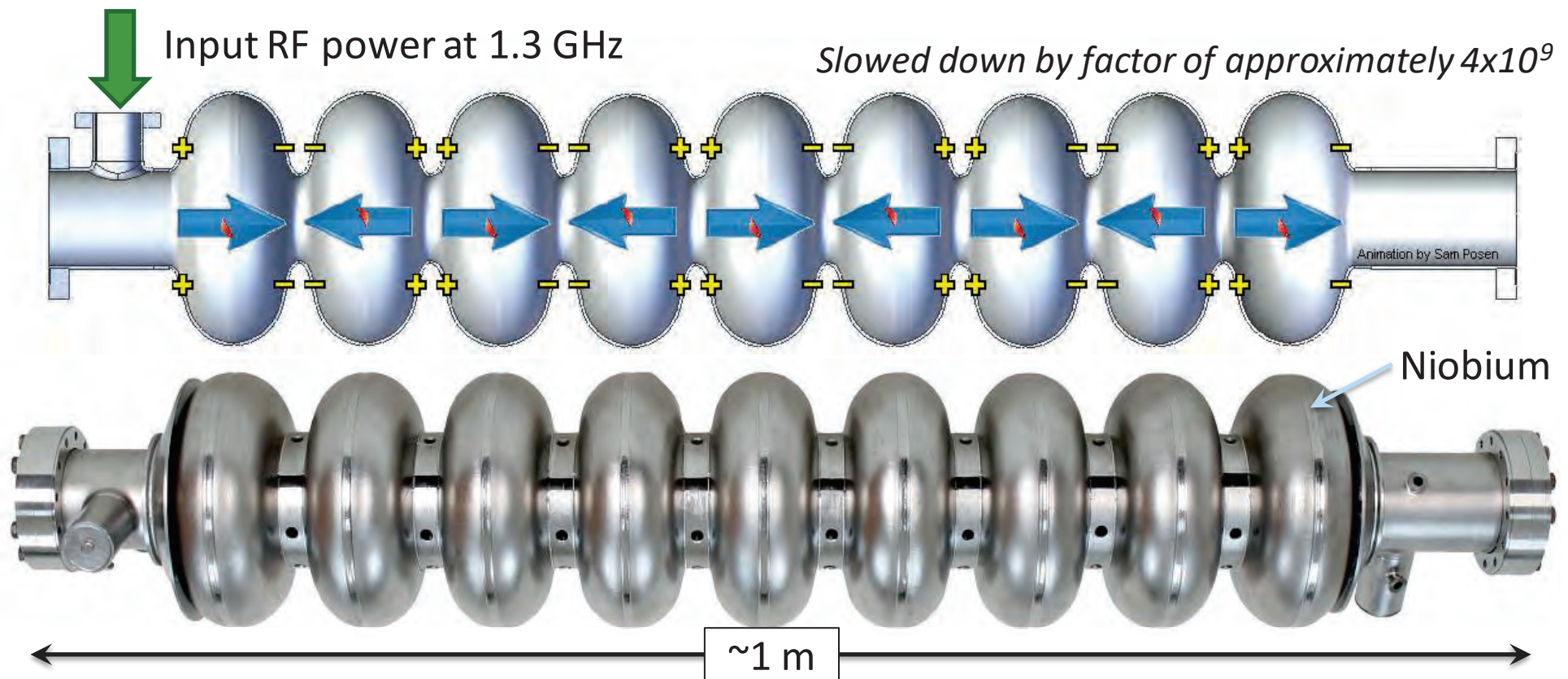
- Introduction to SRF cavities
- State-of-the art surface treatments
- Toward a better understanding
 - BCS surface resistance: new insights and optimization
 - Residual resistance: understanding and minimizing degradation due to trapped flux
- Technology improvement for SRF-based projects
- Conclusions

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- **Introduction to SRF cavities**
- State-of-the art surface treatments
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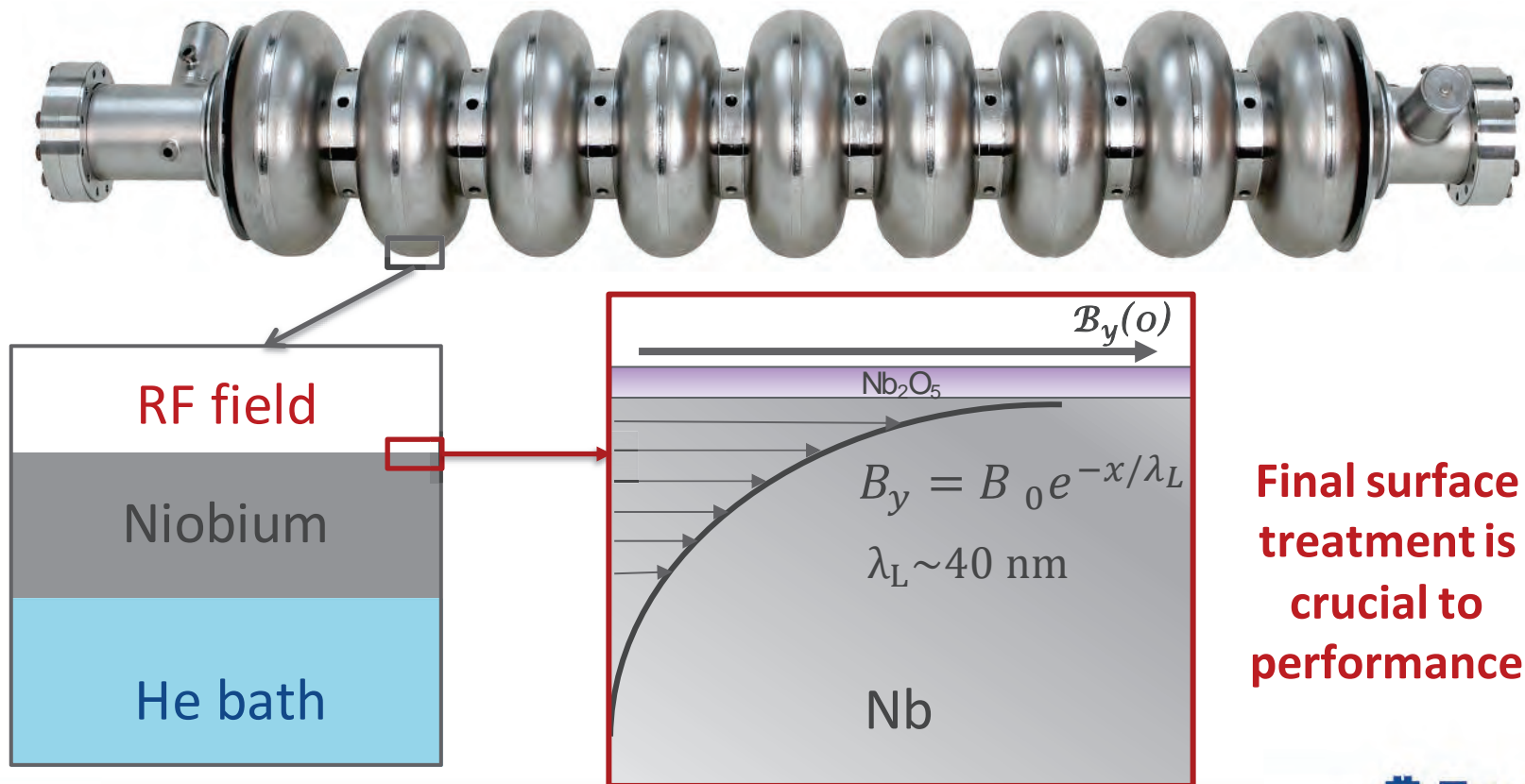
Particle Acceleration via SRF Cavities

- Superconducting Radio-Frequency (SRF) cavities are electro-magnetic resonators characterized by **very low power dissipation**
- The electric field provide acceleration to charged particles



Superconducting RF Cavities

- Niobium ($T_c=9.2$ K), T operation 2 - 4.5 K
- RF surface resistance \rightarrow fighting against $n\Omega$
- Losses concentrate on the first ~ 100 nm of the **inner surface**



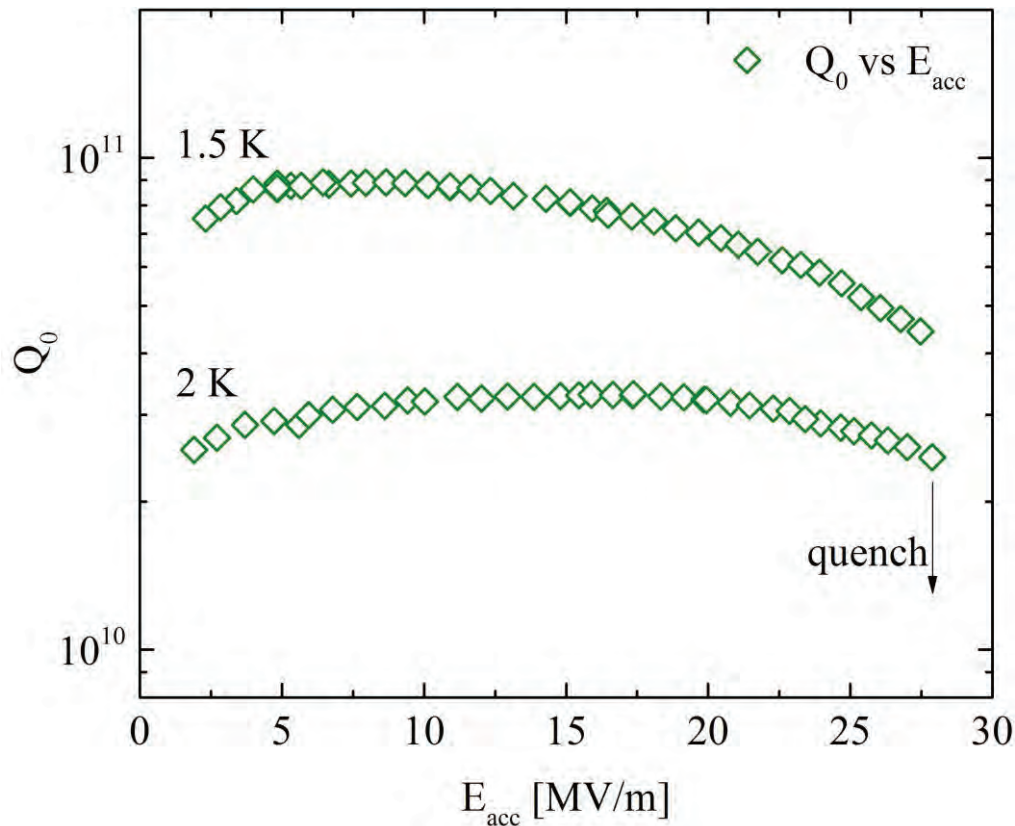
Beam view, inside the cavity



Extreme attention to surface treatments and surface cleanliness are mandatory



SRF Cavities figure of merits: Q_0 vs E_{acc}



Q-factor (Q_0):

$$Q_0 = \frac{G}{R_s} = \frac{\omega_0 U}{P_d}$$

High $Q_0 \rightarrow$ lower power consumption.
Limited by $n\Omega$ of surface resistance.

Accelerating field (E_{acc}):

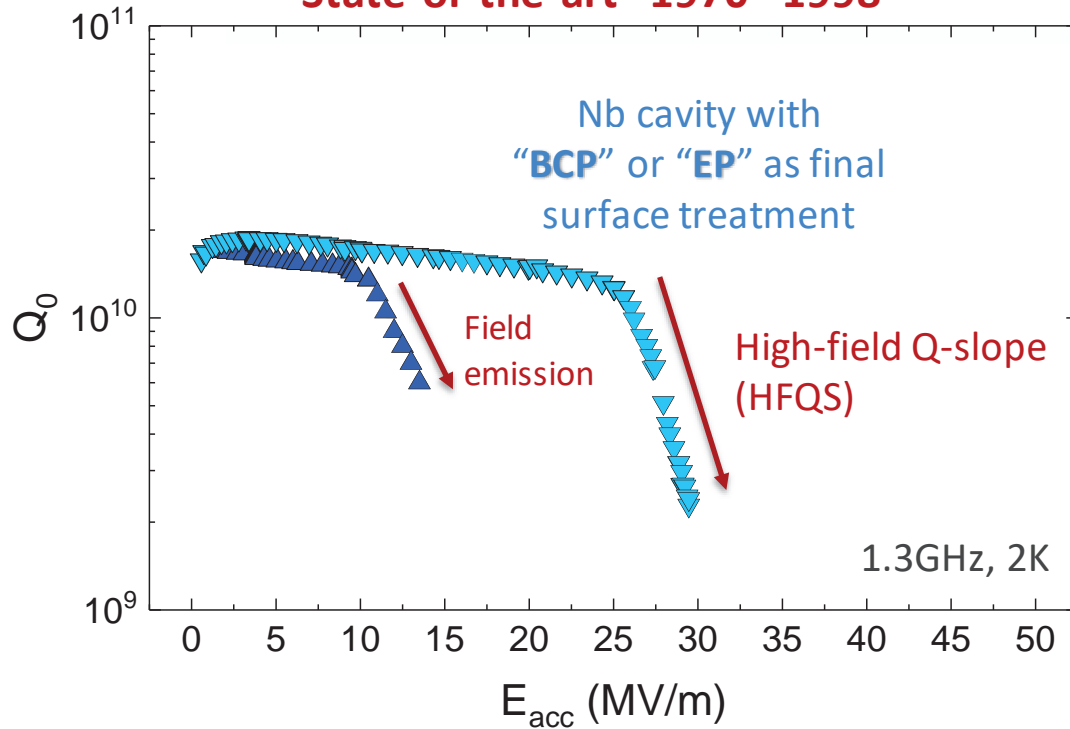
Determine the energy transferred to charged particles.

High $E_{acc} \rightarrow$ lower accelerator length.
Limited by quench of the SC state.

High Q at high gradients may reduce both **capital and operational costs** of accelerators

Historical progress of SRF technology

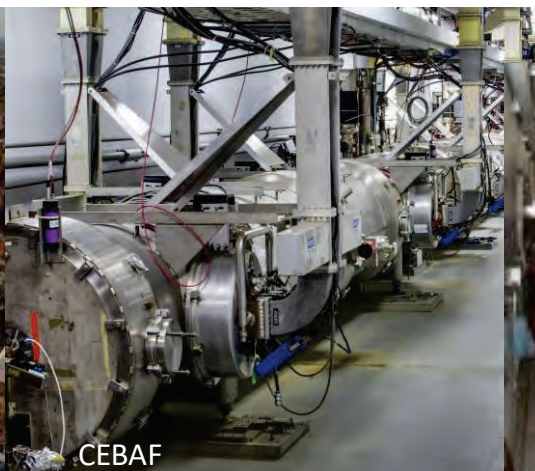
State-of-the-art ~1970 - 1998



BCP=Buffer chemical polishing
EP=Electro-polishing

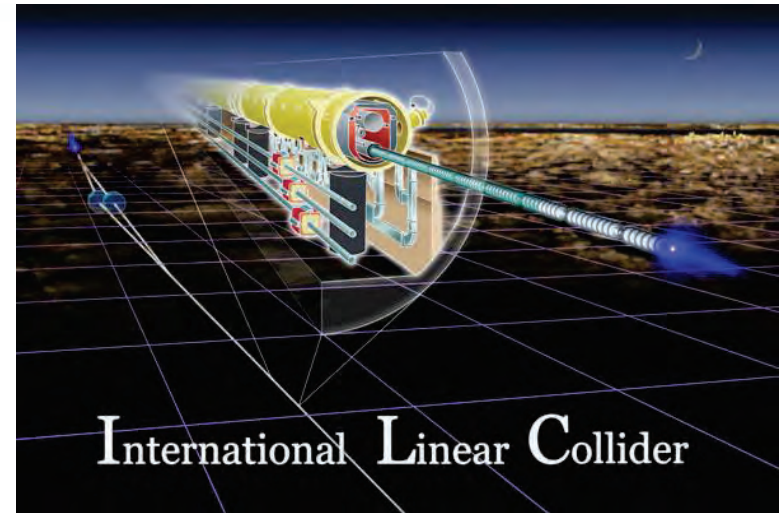
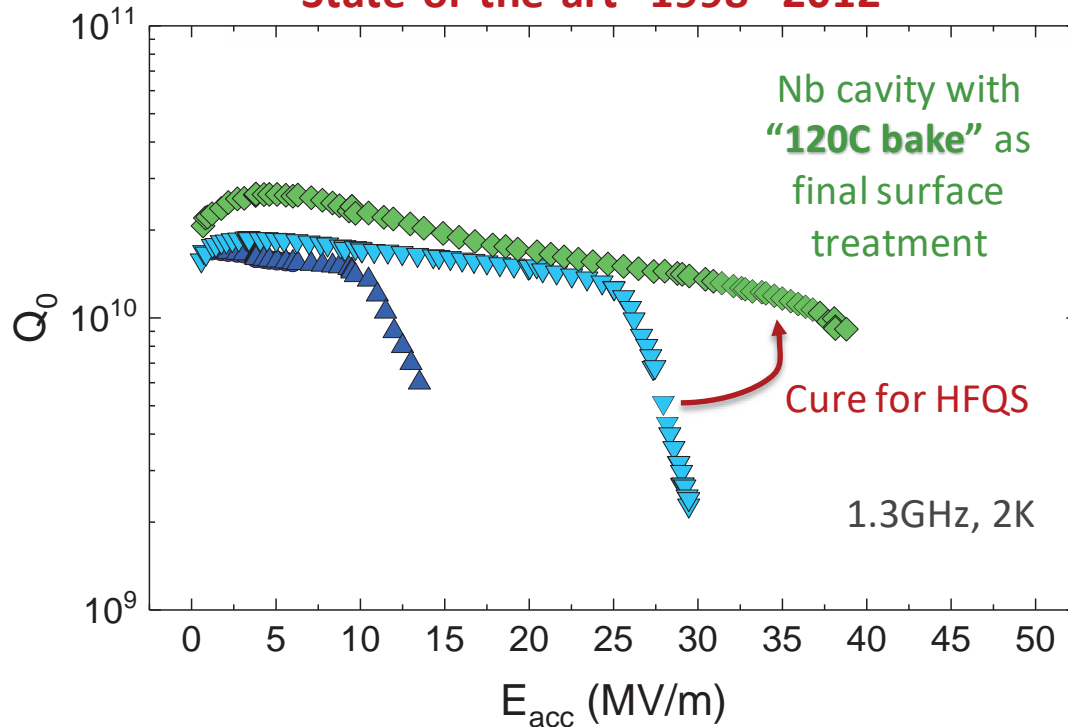
Enabled first SRF-based particle accelerators:

- **CEBAF**
- **CESR**
- **TRISTAN**
- **HERA**
- **SNS**
- ...



Historical progress of SRF technology

State-of-the-art ~1998 - 2012

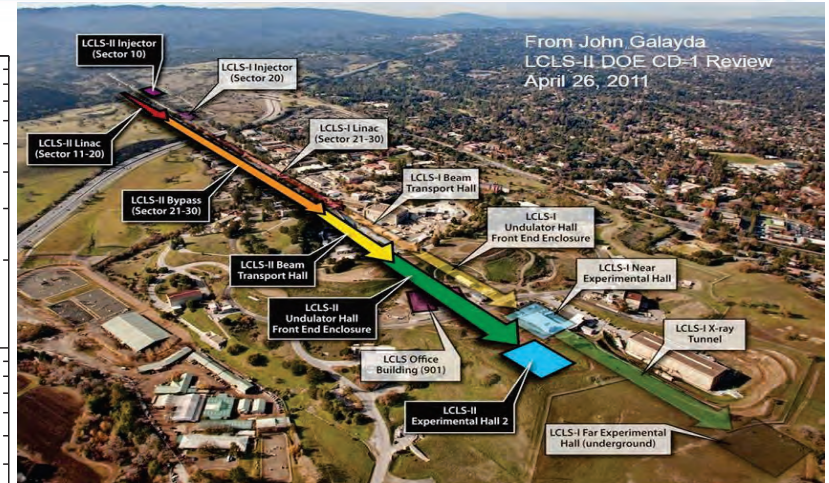
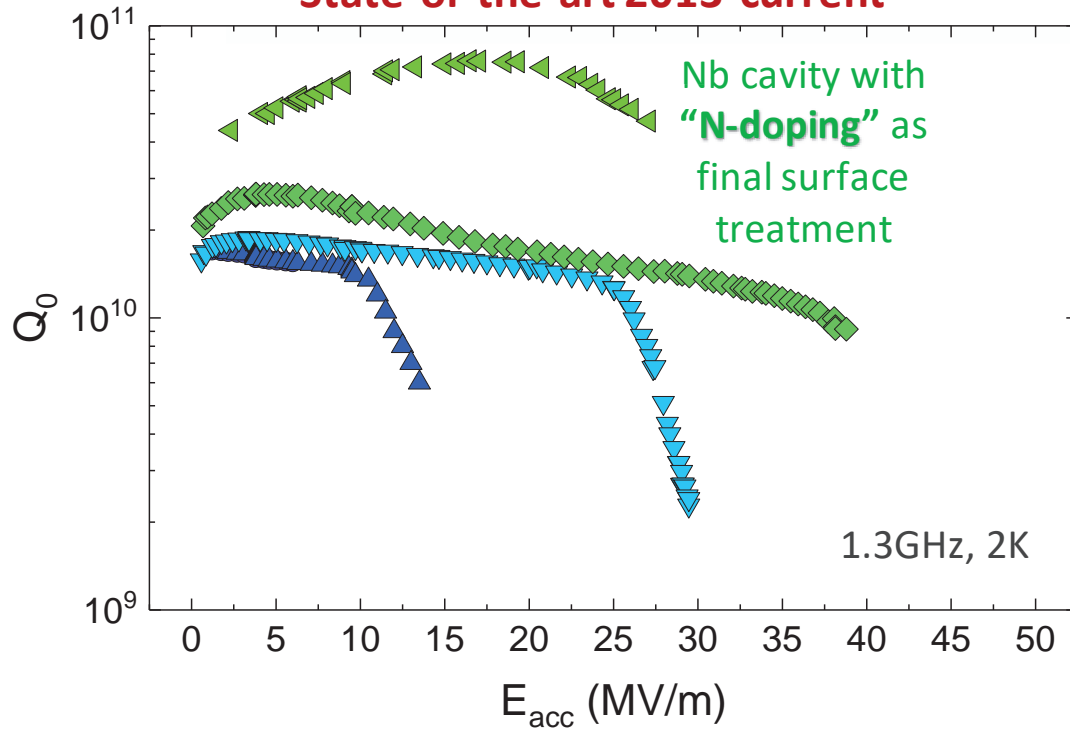


Enabled big SRF-based projects (LINAC):

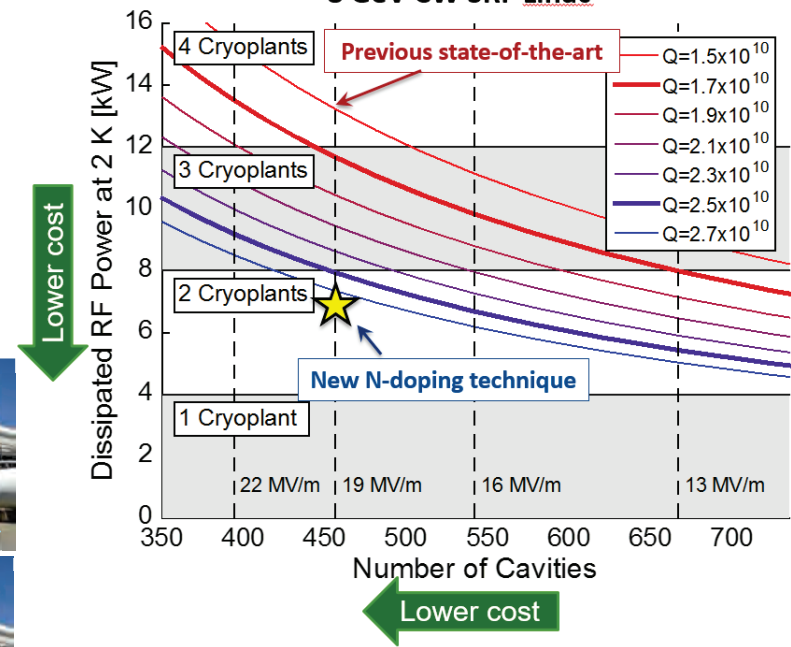
- **ILC – 500 GeV** (16,000 Nb cavities)
 $Q = 1 \cdot 10^{10}$ at 31.5 MV/m
- **E-XFEL – 17.5 GeV** (816 Nb cavities)
 $Q = 1 \cdot 10^{10}$ at 23.6 MV/m

Historical progress of SRF technology

State-of-the-art 2013-current



8 GeV CW SRF Linac

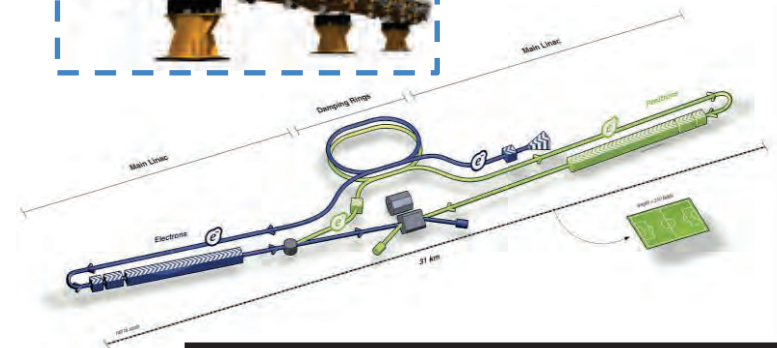
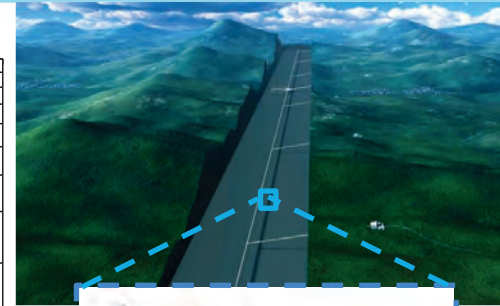
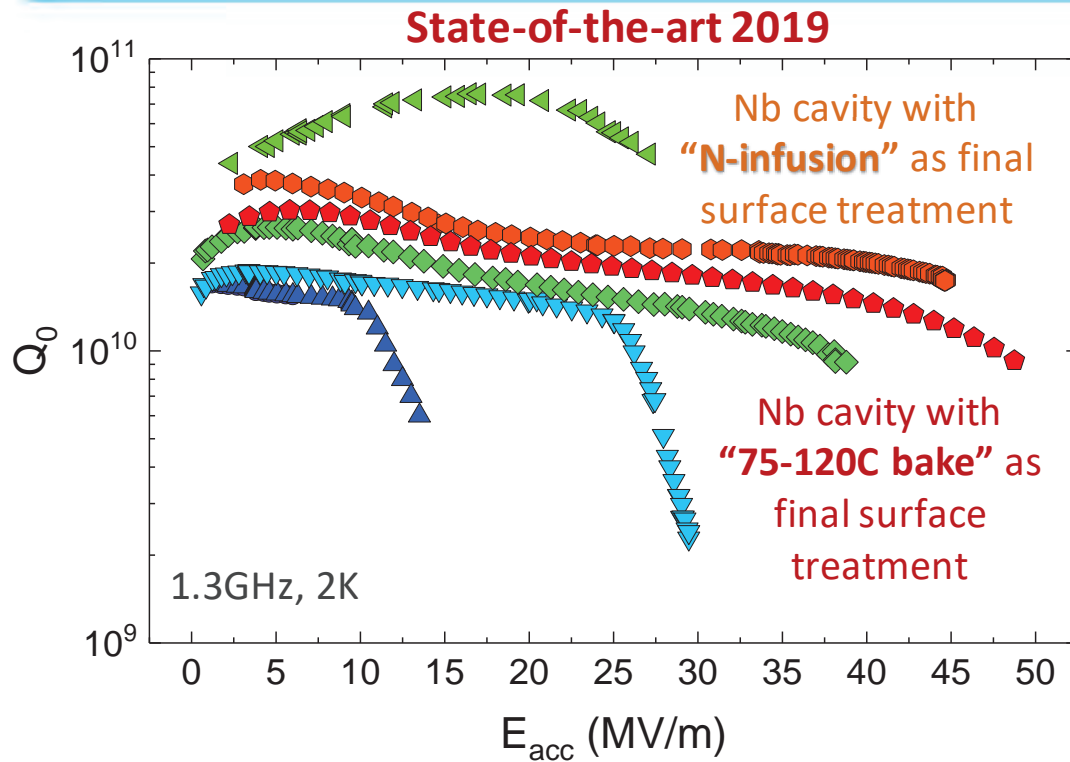


Enabled affordable CW operation:

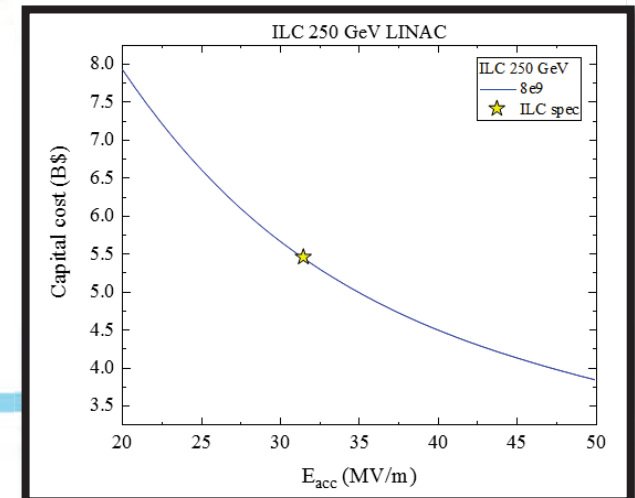
- LCLS-II
- LCLS-II HE
- PIP-II
- ...



Historical progress of SRF technology



N-infusion and *75-120C bake* enable significant cost reduction for high gradient machines like **ILC**, **E-XFEL upgrade**, etc

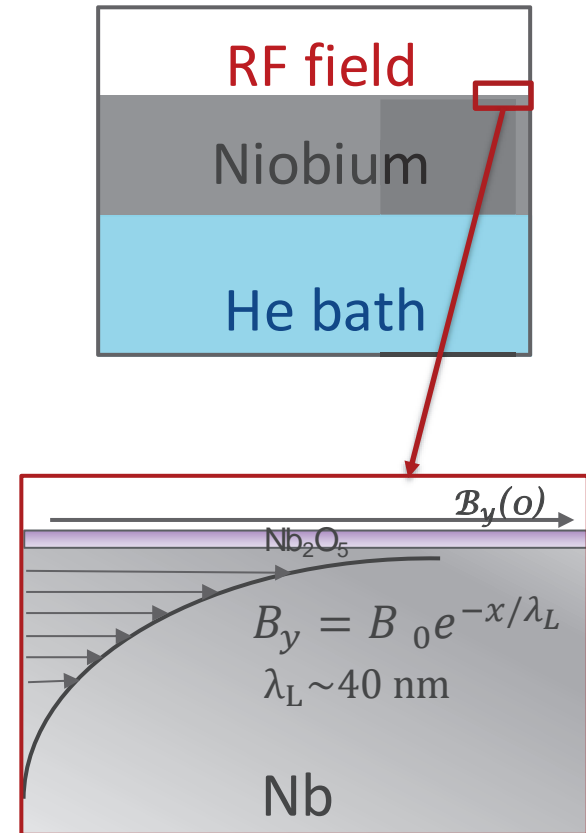
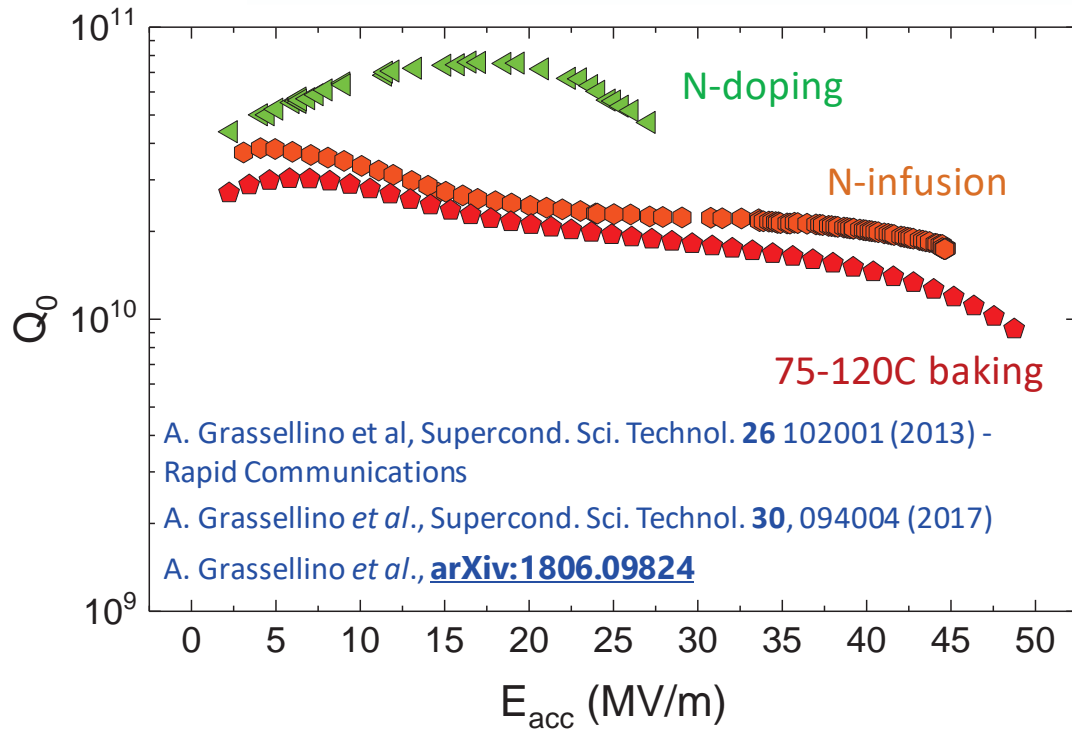


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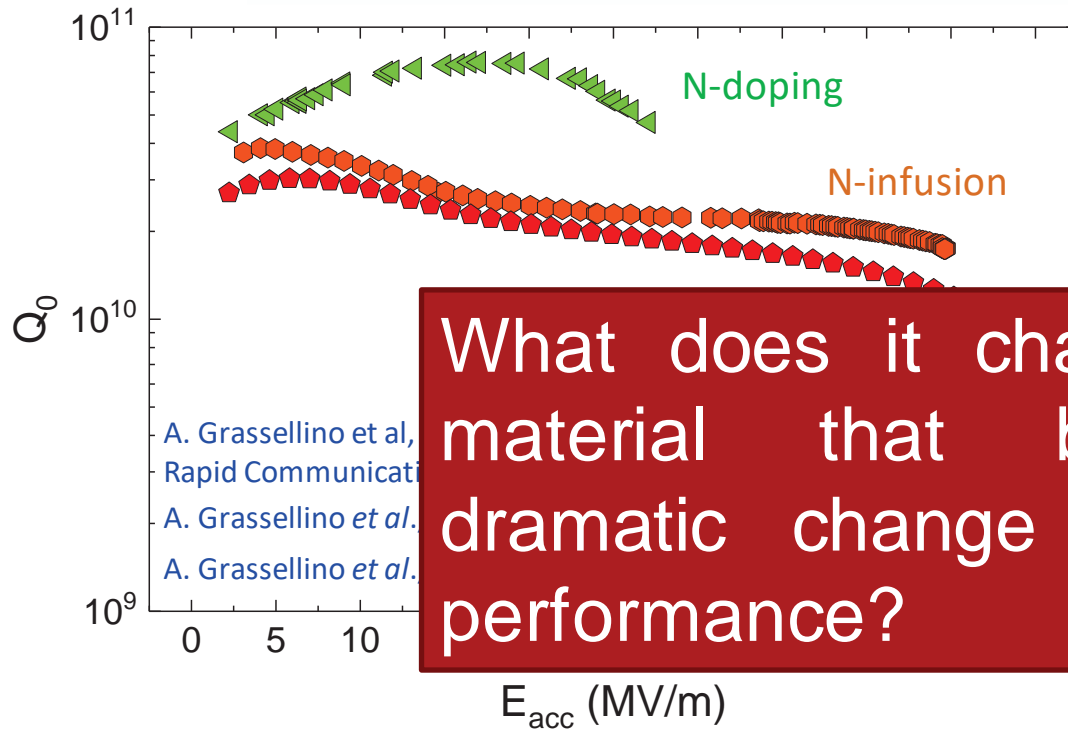
Historical progress of SRF technology

Current state-of-the-art

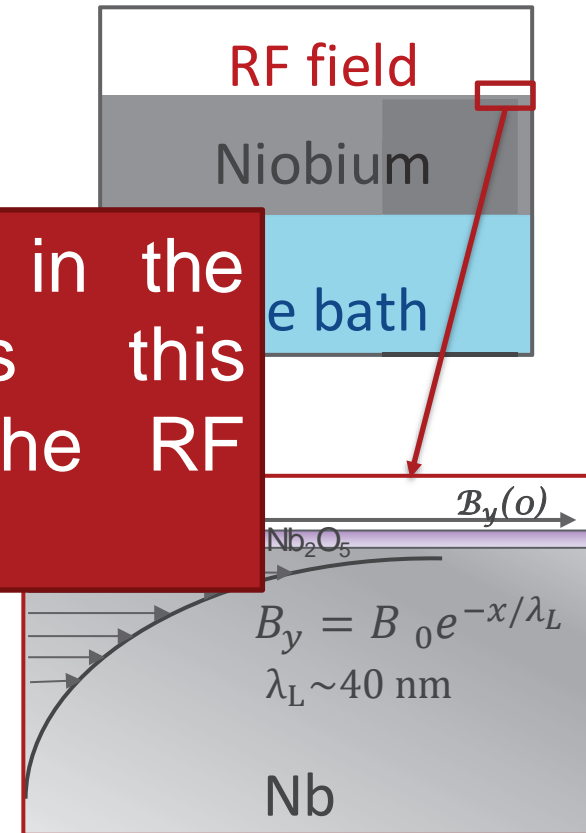


Historical progress of SRF technology

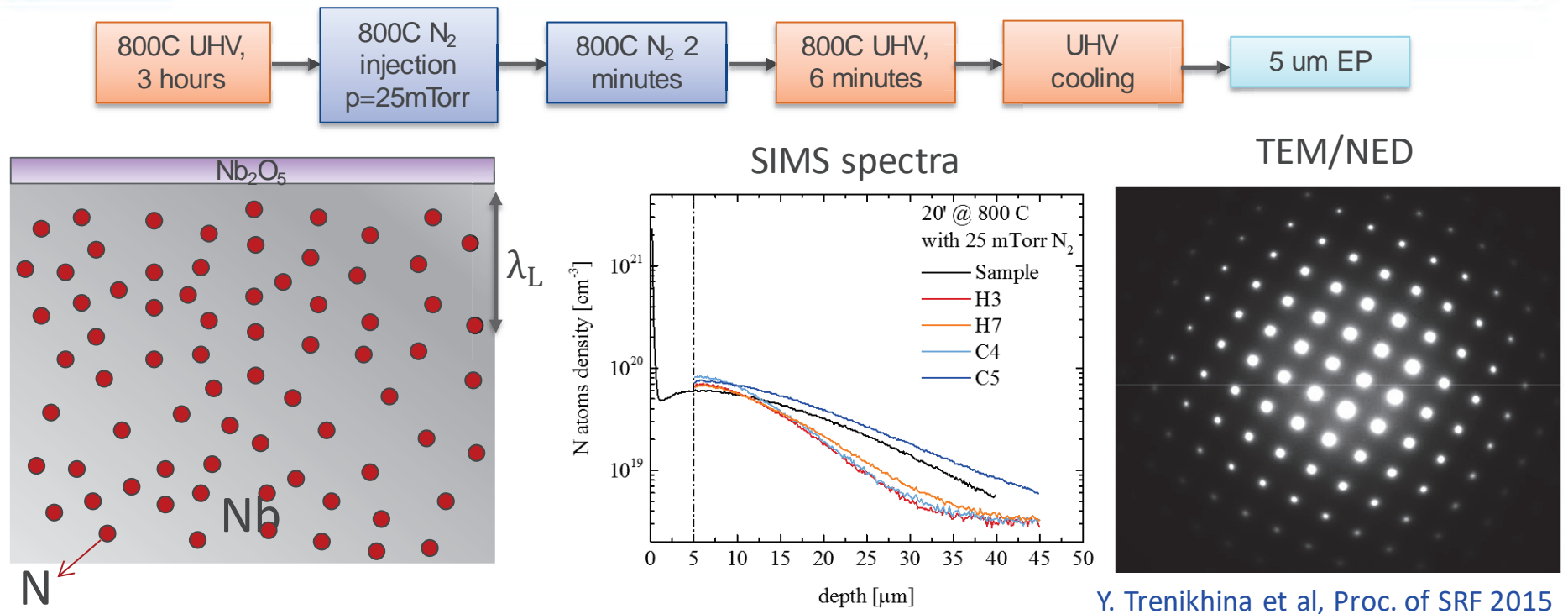
Current state-of-the-art



What does it change in the material that brings this dramatic change in the RF performance?

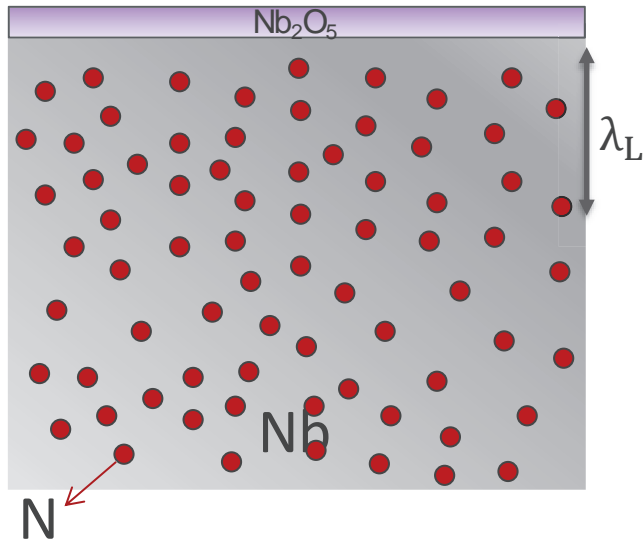
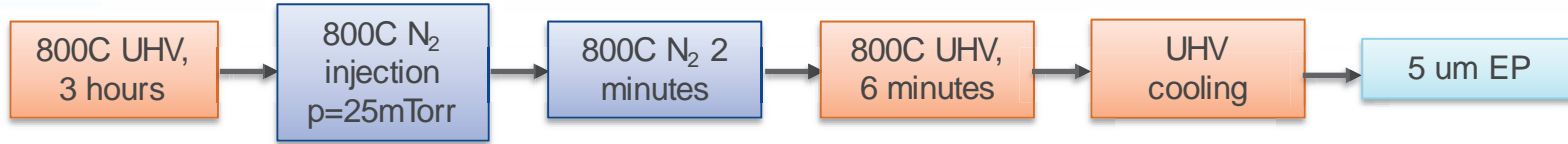


N-doping

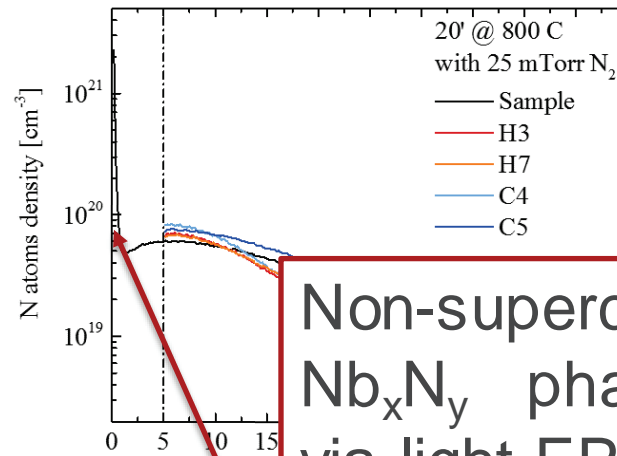


- SIMS spectra: nitrogen in the material for about tens of micrometers
- TEM/NED: only Nb signal from diffraction pattern → N is interstitial

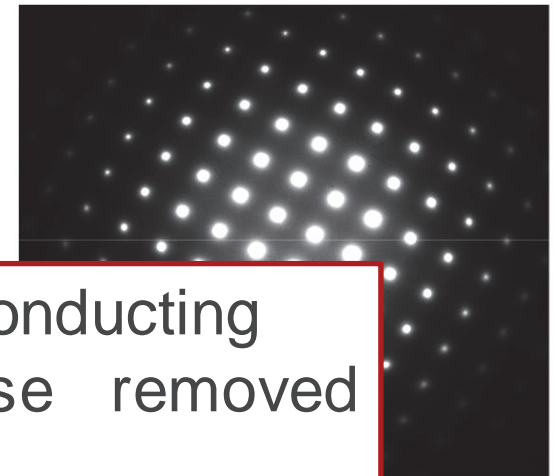
N-doping



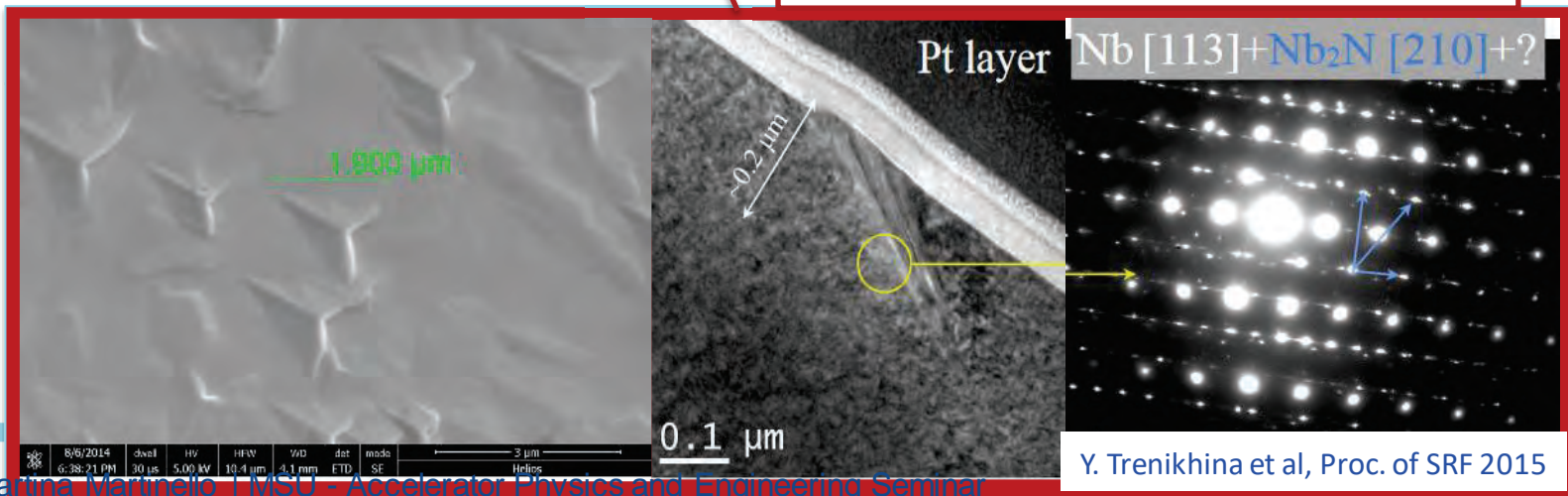
SIMS spectra



TEM/NED



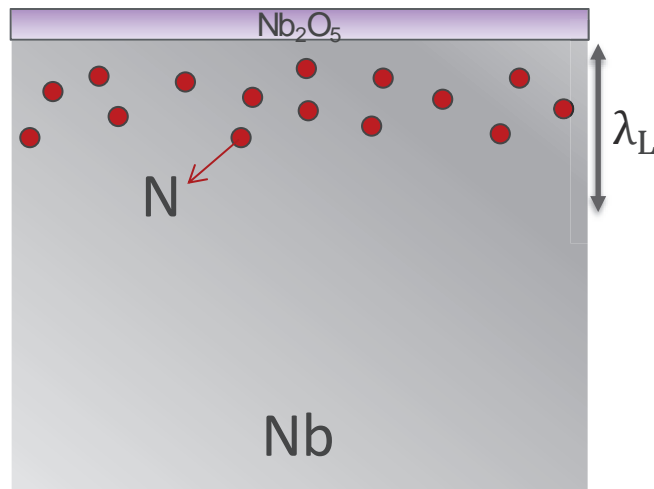
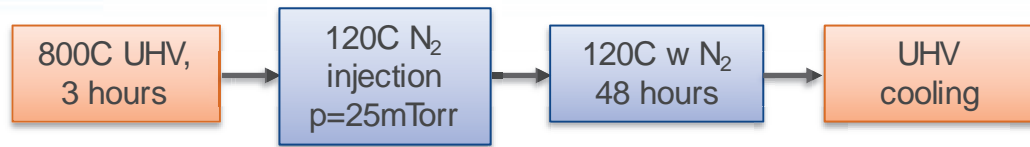
Non-superconducting Nb_xN_y phase removed via light EP



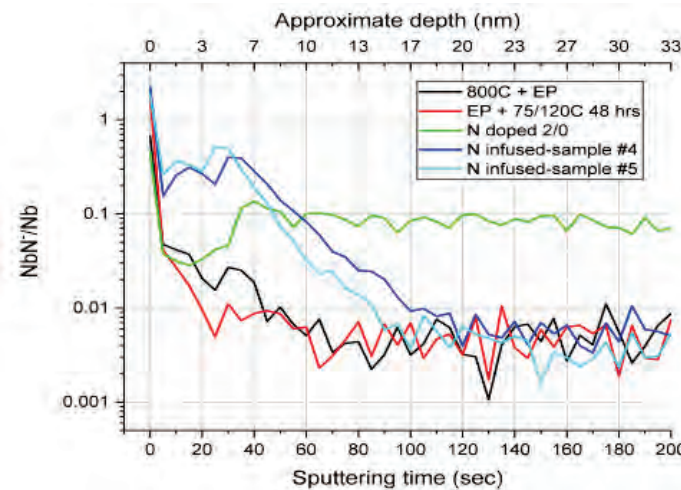
Y. Trenikhina et al, Proc. of SRF 2015

ilab

120C N-infusion



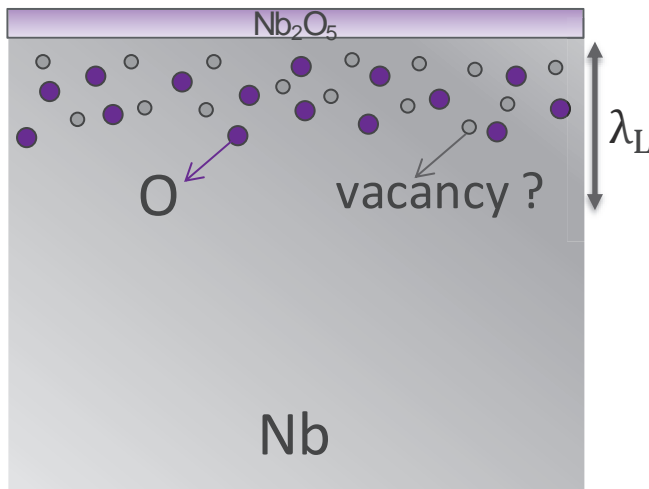
SIMS spectra of cavity cut-outs



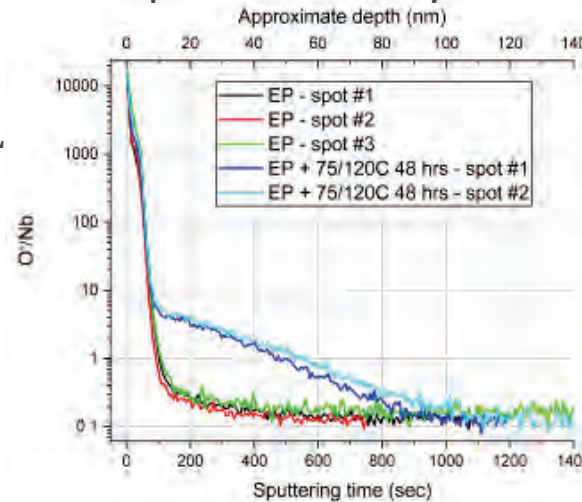
A. Romanenko *et al.*, Proc. of SRF 2019

- SIMS spectra: nitrogen in the material for about 10-15 nanometers

75-120C baking



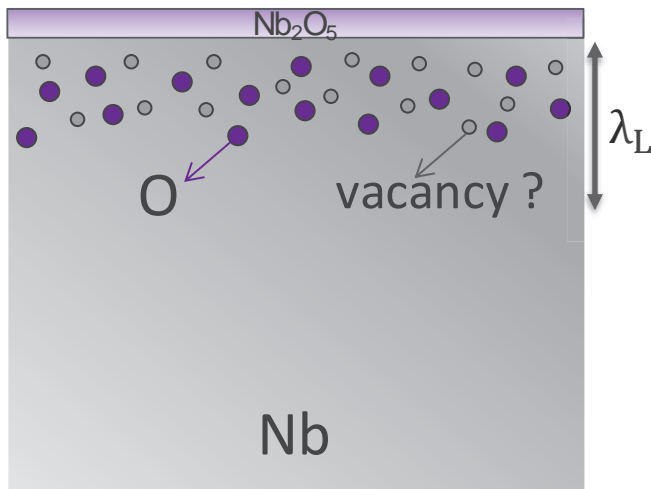
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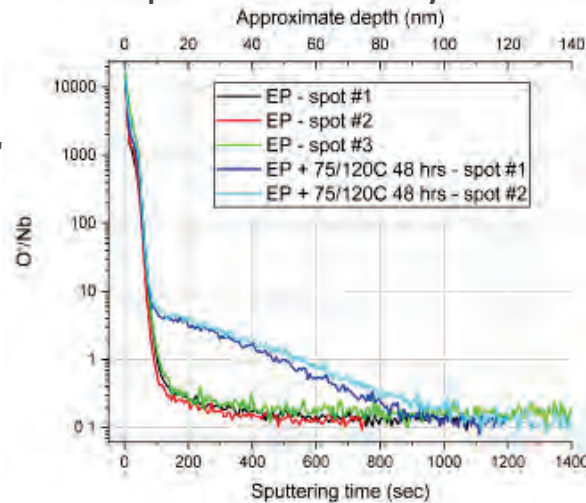
A. Romanenko *et al.*, Proc. of SRF 2019

- SIMS spectra: oxygen in the material for about tens of nanometers

75-120C baking

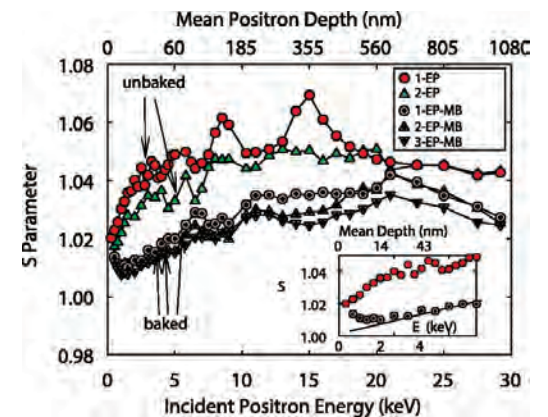


SIMS spectra of cavity cut-outs



A. Romanenko *et al.*, Proc. of SRF 2019

Positron annihilation in 120C baked samples

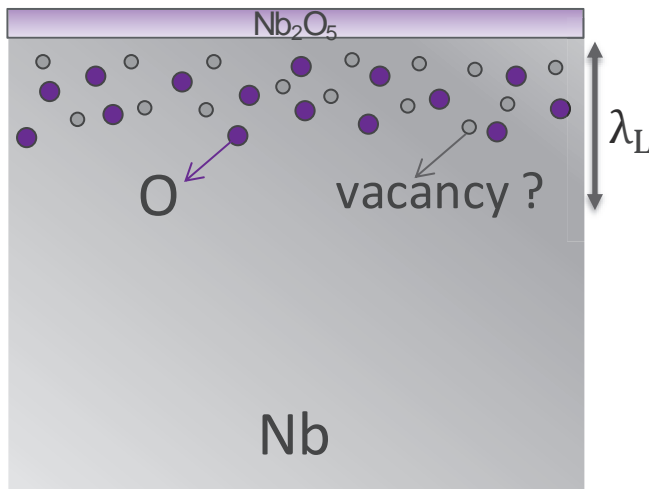


A. Romanenko *et al.*, Appl. Phys. Lett. 10, 232601 (2013)

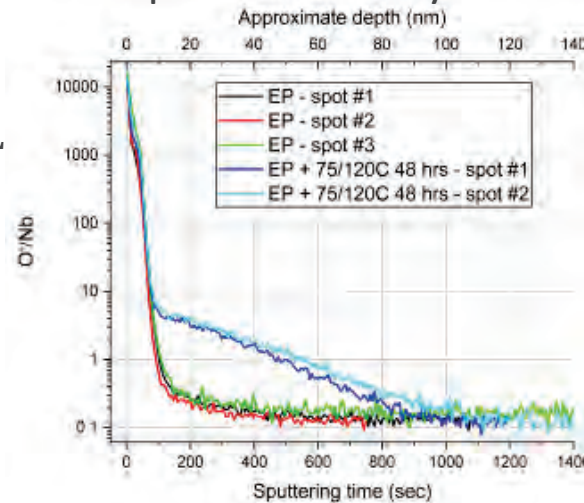
- SIMS spectra: oxygen in the material for nanometers

Positron annihilation studies show possible presence of vacancy-hydrogen complexes in 120C bake samples, in the first nanometers

75-120C baking

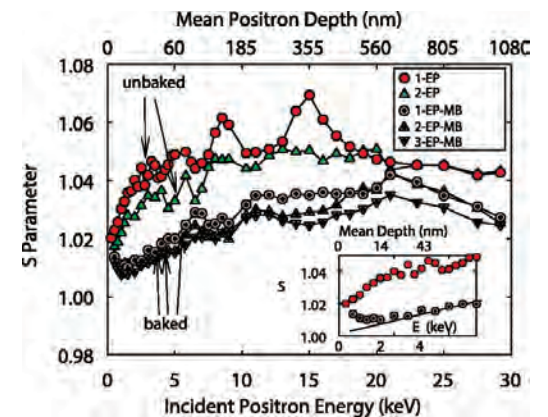


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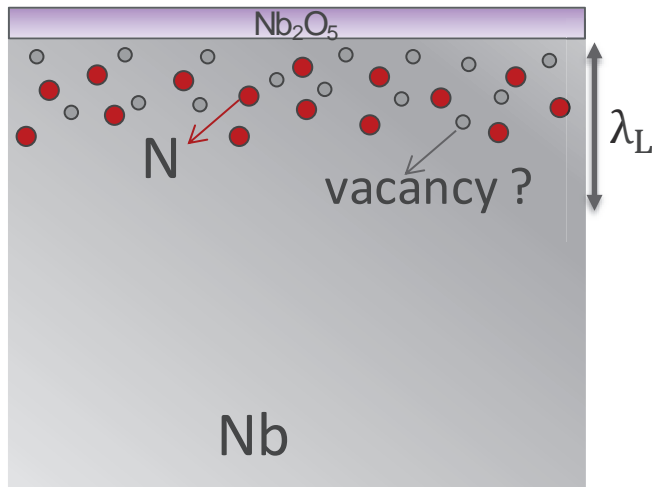
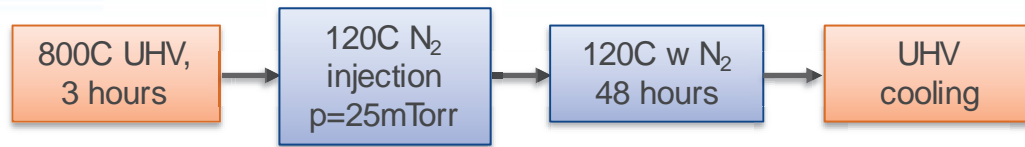


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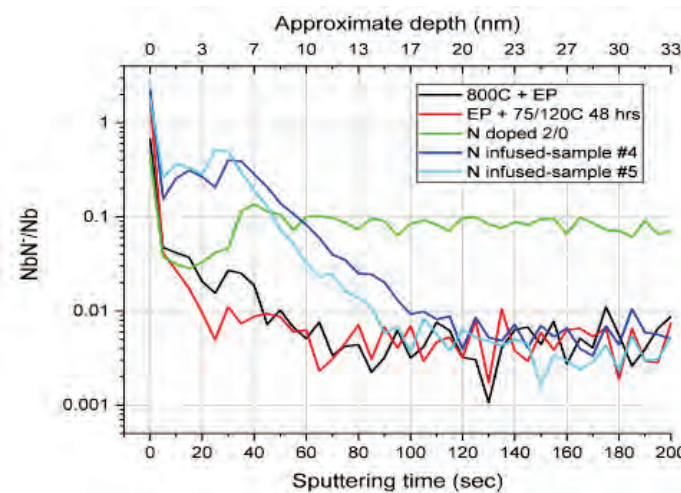
- SIMS spectra: oxygen in the material for nanometers
- Possibility of having vacancy-hydrogen complexes also after the 75-120C baking

Positron annihilation studies show possible presence of vacancy-hydrogen complexes in 120C bake samples, in the first nanometers

120C N-infusion



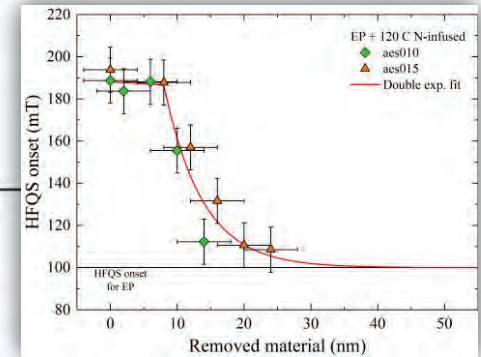
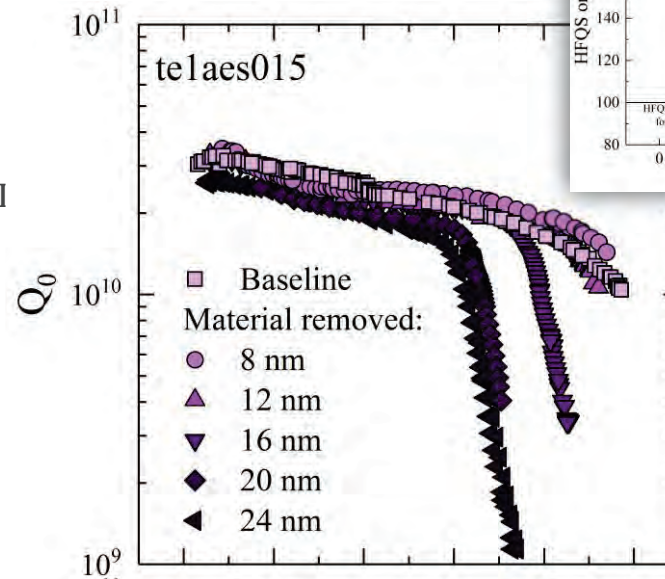
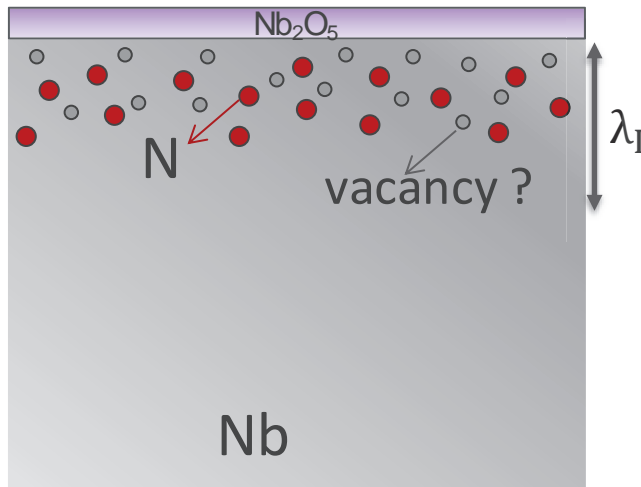
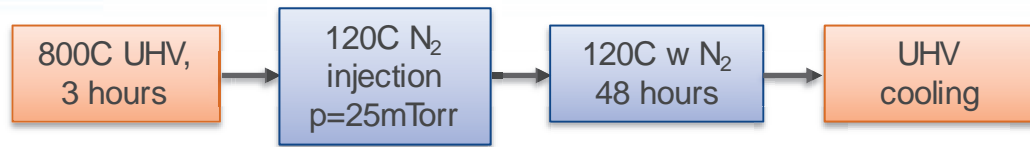
SIMS spectra of cavity cut-outs



A. Romanenko *et al.*, Proc. of SRF 2019

- SIMS spectra: nitrogen in the material for about 10-15 nanometers
- Vacancy-hydrogen complexes present also after 120C N-infusion?

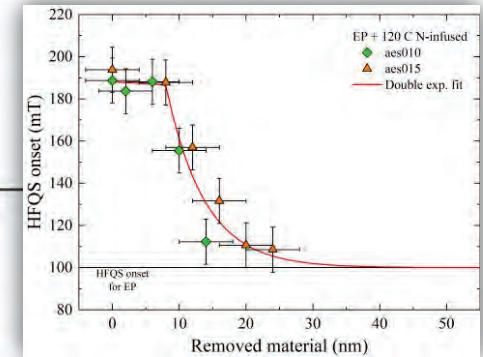
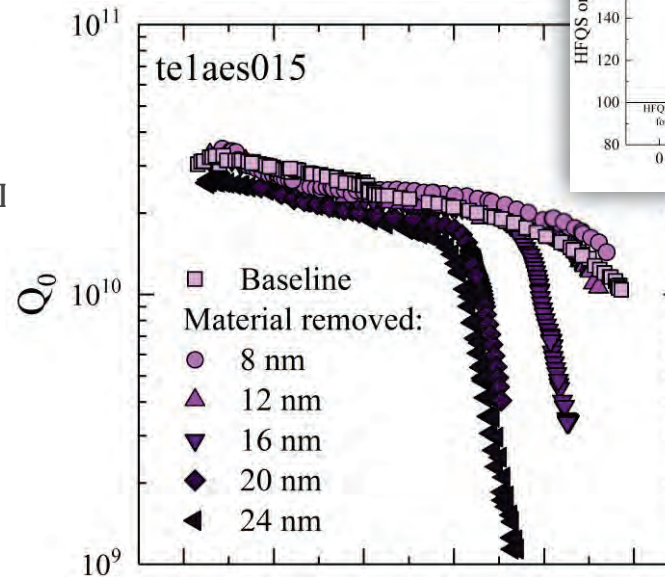
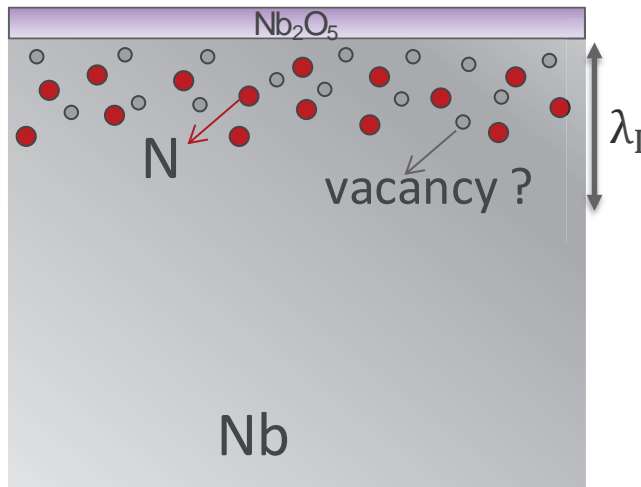
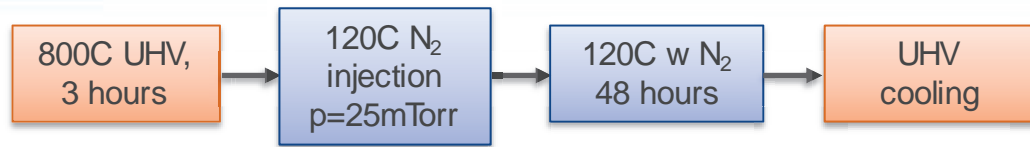
120C N-infusion



M. Checchin, A. Grassellino *Appl. Phys. Lett.* 117, 032601 (2020)

- HF rinses studies shows that the N-infusion process modifies only the surface of the material
- HFQS behavior is totally re-established after removing ~ 20 nm

120C N-infusion



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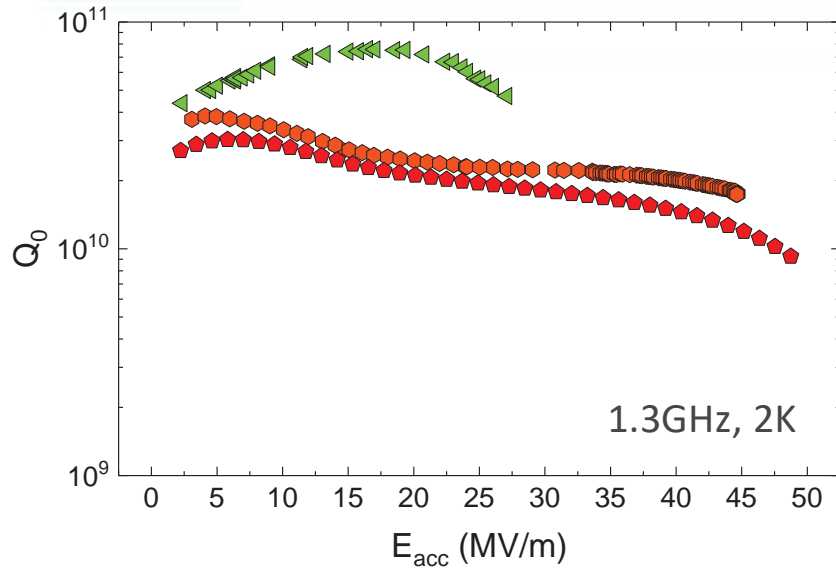
- HF rinses studies shows that the N-infusion process modifies only the surface of the material
- HFQS behavior is totally re-established after removing ~ 20 nm

The performance improvement of the N-infusion treatment must be connected to impurities within the first 15-20 nm

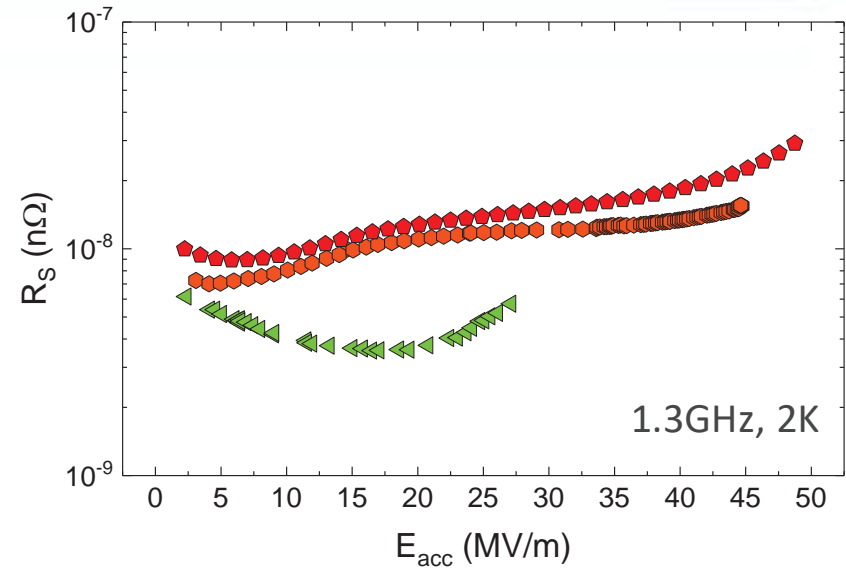
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The surface resistance



$$Q_0 = \frac{G}{R_S}$$

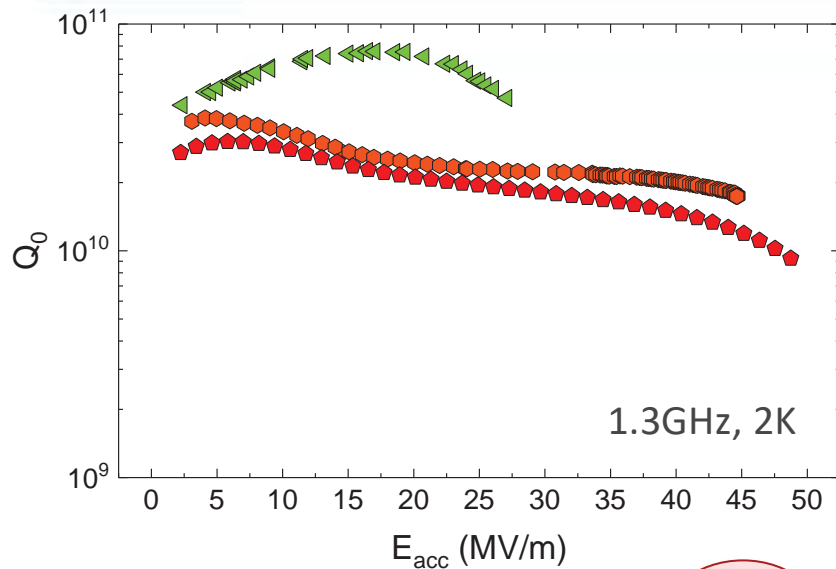


$$R_S = R_{BCS} + R_{fl} + R_0$$

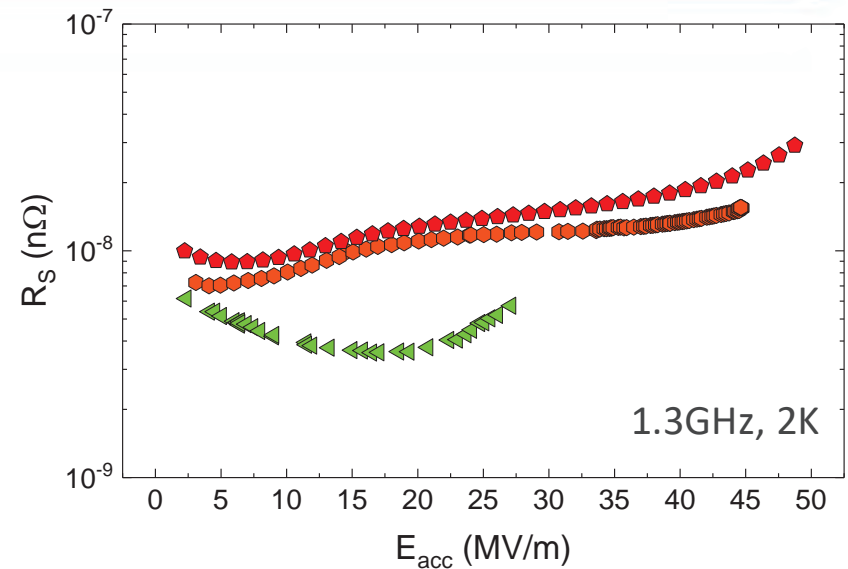
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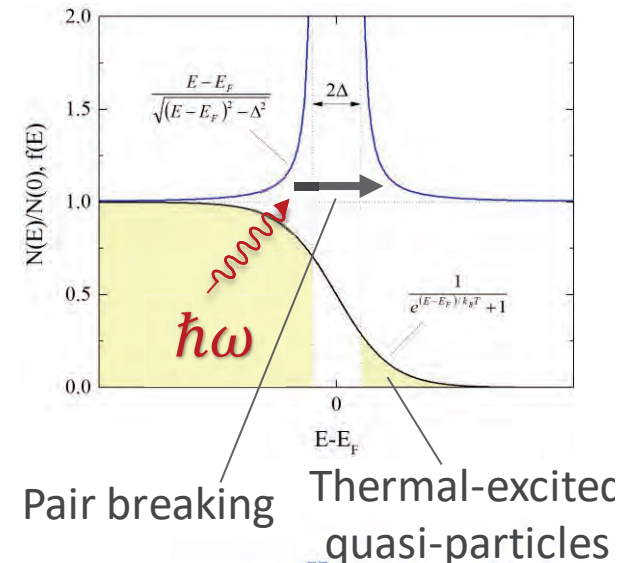
$$Q_0 = \frac{G}{R_S}$$



$$R_S = R_{BCS} + R_{fl} + R_0$$

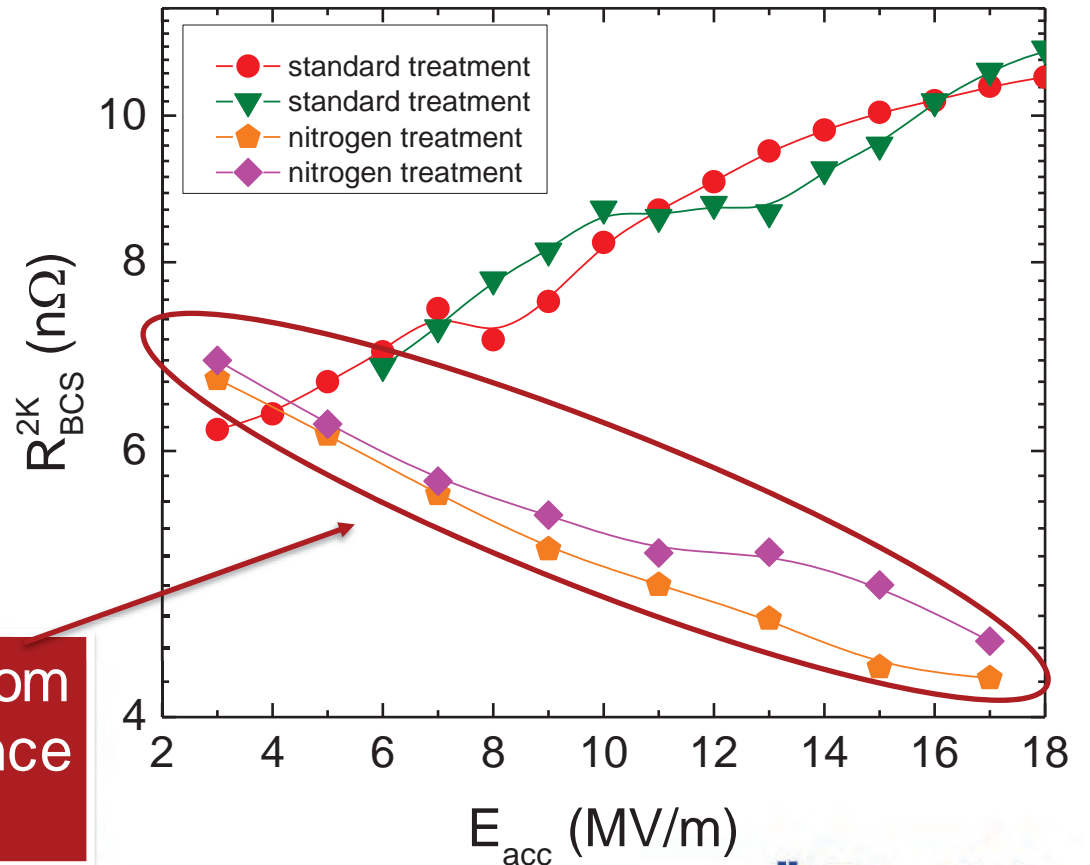
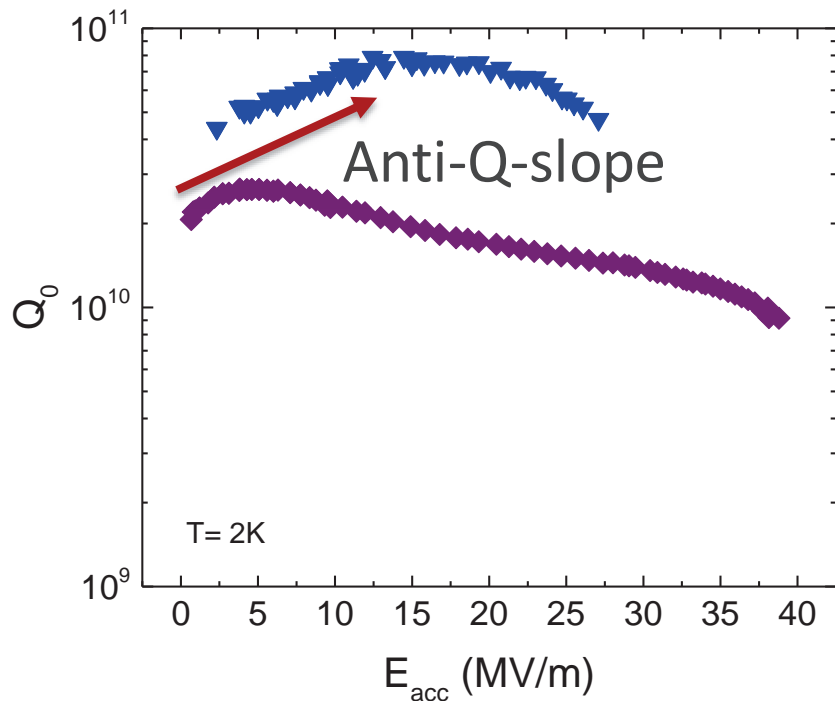
BCS surface resistance

- Dissipation due to thermally-excited quasi-particles (“normal-electrons”)
- Depends on temperature, normal-conducting and superconducting material parameters (mean-free-path, energy gap, penetration depth, etc)



First observation of R_{BCS} decreasing with field

$$R_S(2 K) = R_{BCS}(2 K) + R_0 + R_{fl}$$

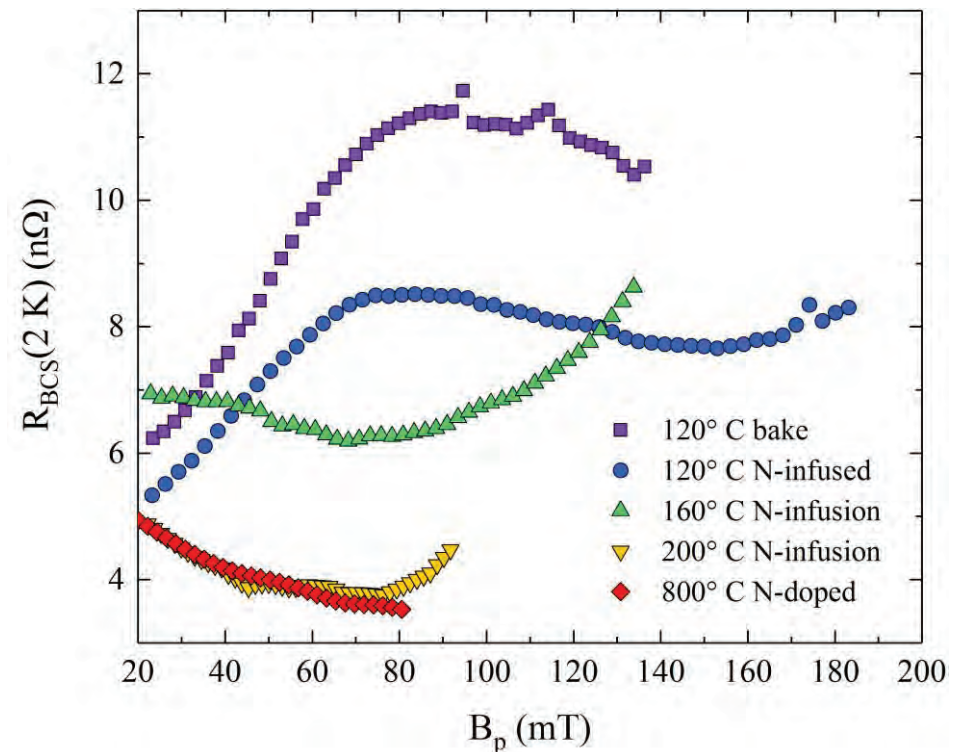
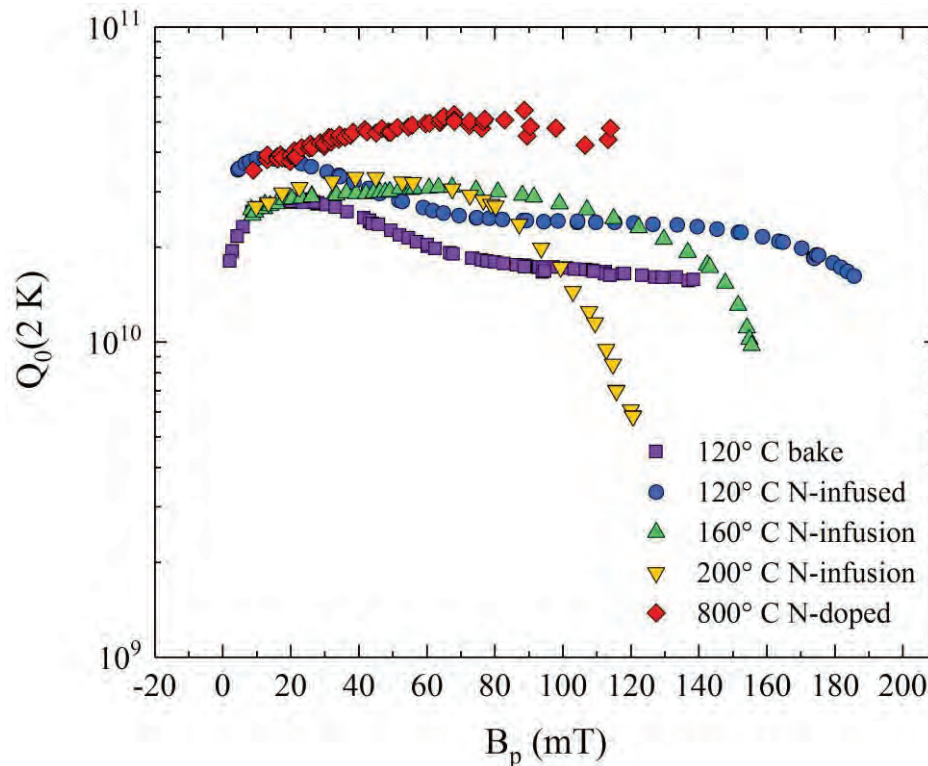


Anti-Q-slope emerges from the BCS surface resistance decreasing with field

R_{BCS} tuning with N-infusion at different T

By N-doping Nb cavities at lower temperatures (N-infusion) we can tune the Q-factor:

⇒ strong effects on the BCS and residual resistance



A. Grassellino *et al.*, Supercond. Sci. Technol. **30**, 094004 (2017)

Study of Cavities at Different Frequency

650 MHz



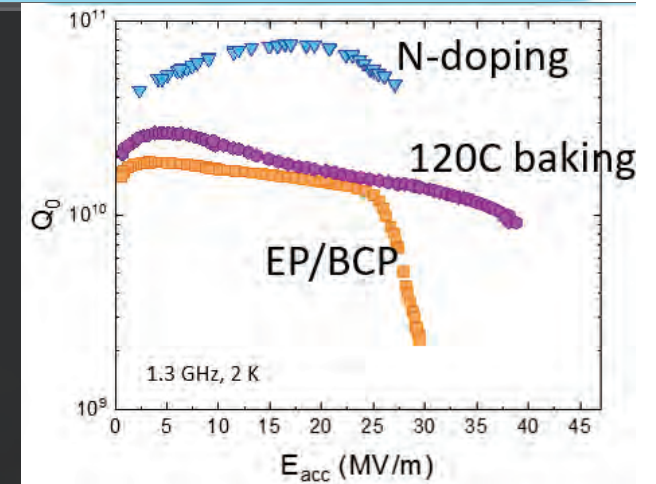
1.3 GHz



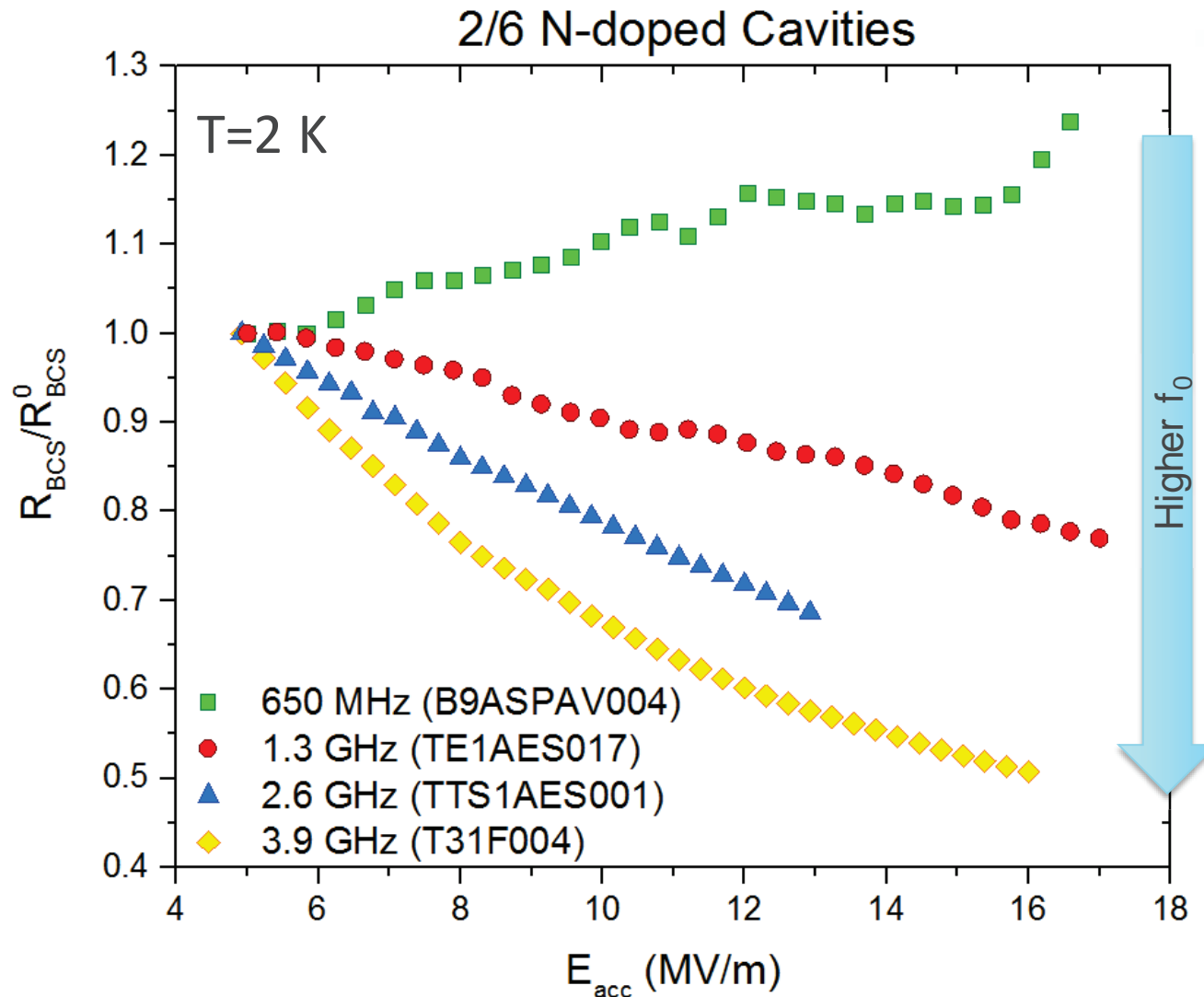
2.6 GHz



3.9 GHz

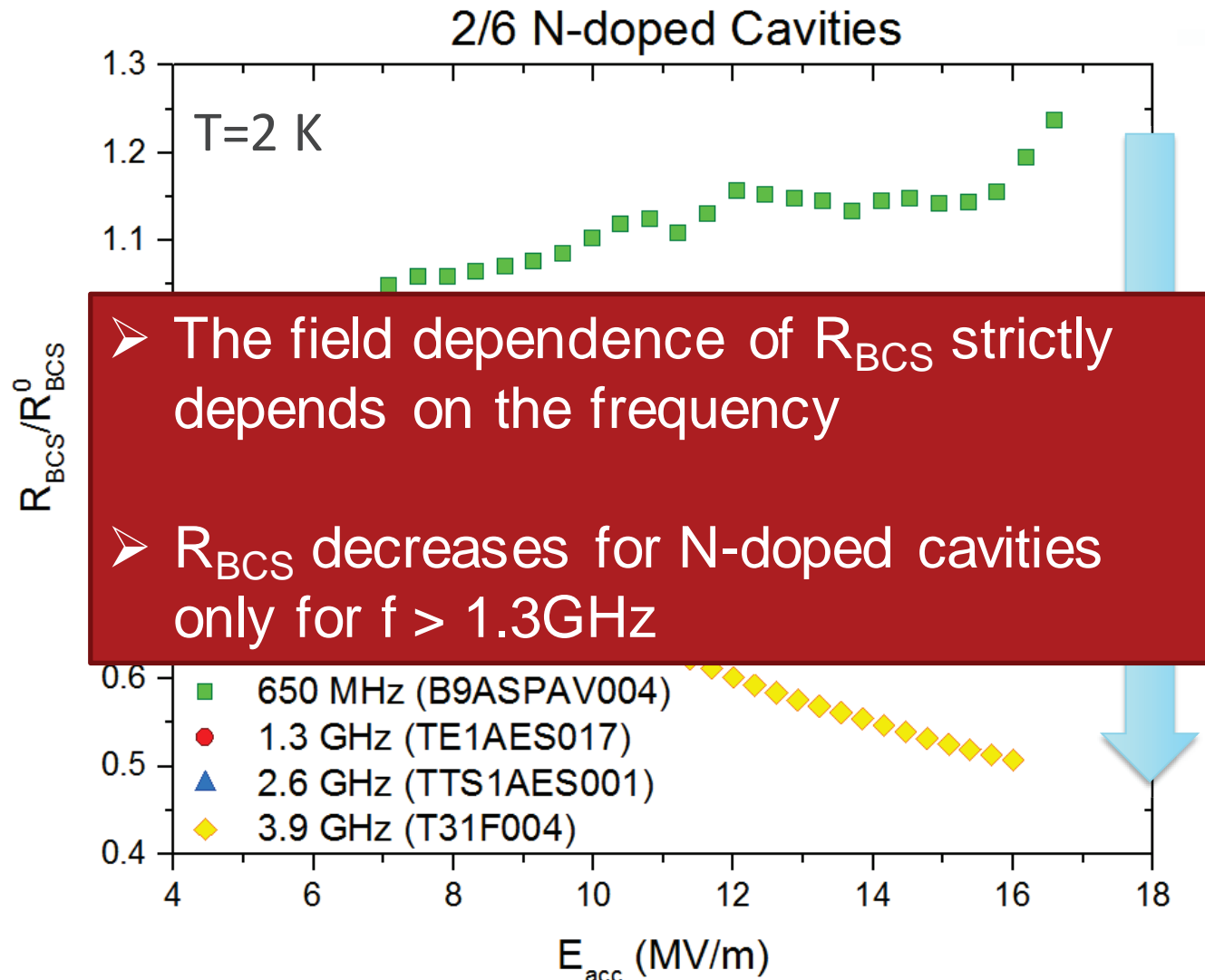


Frequency dependence of $R_{BCS}(E_{acc})$: toward a better understanding



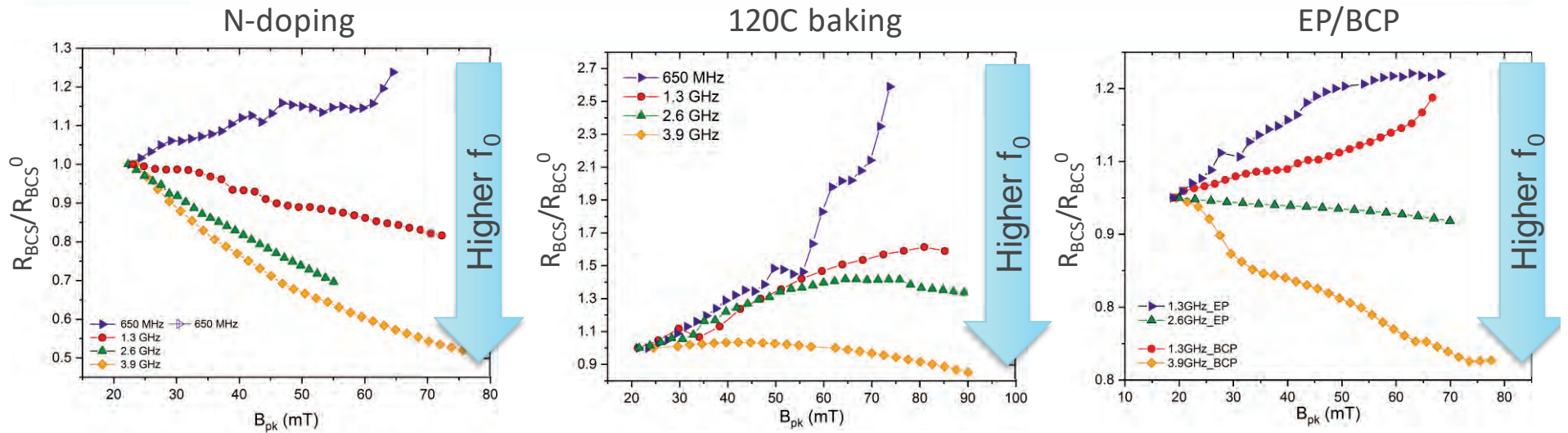
M. Martinello *et al.*, Phys. Rev. Lett. 121, 224801 (2018)

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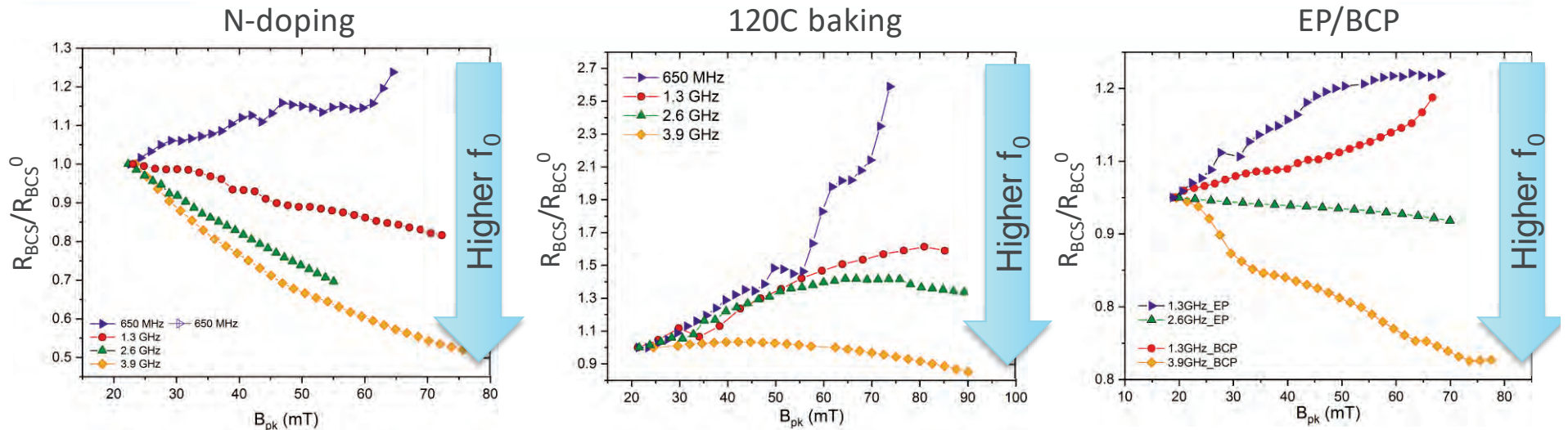
Summary of the Frequency Dependence of $R_{BCS}(E_{acc})$



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Summary of the Frequency Dependence of $R_{BCS}(E_{acc})$

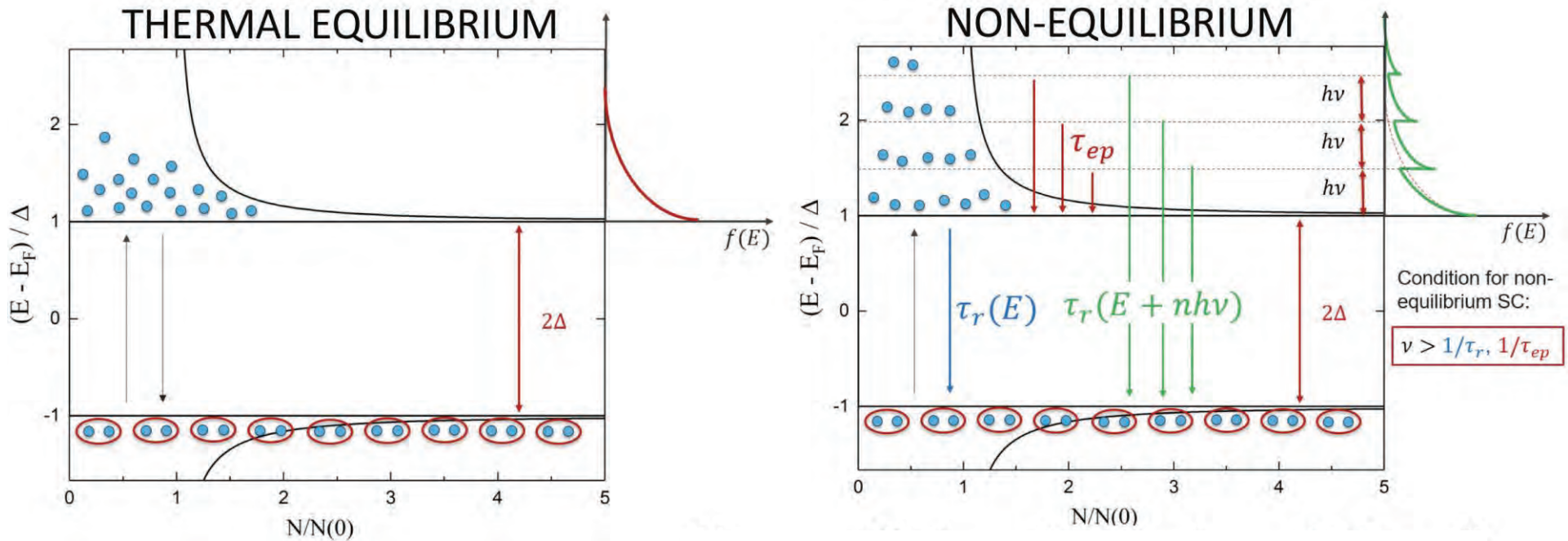


The R_{BCS} decreasing with the field, that has been considered the signature of the N-doped treatment, is actually visible also in clean Nb (EP/BCP) but at higher frequency (> 1.3 GHz)

M. Martinello *et al.*, Phys. Rev. Lett. 121, 224801 (2018)

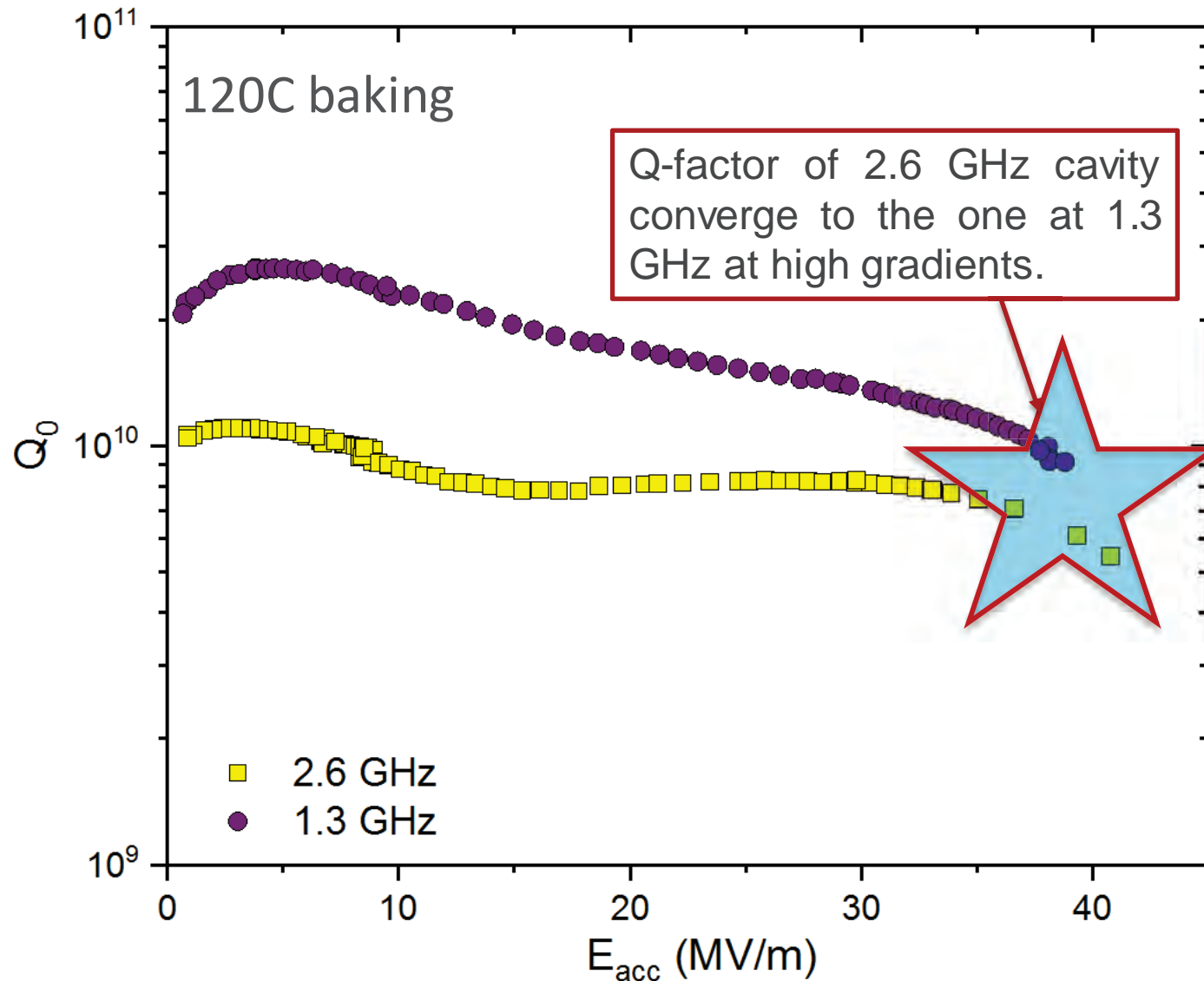


Doping and Frequency Effect on the QPs Distribution

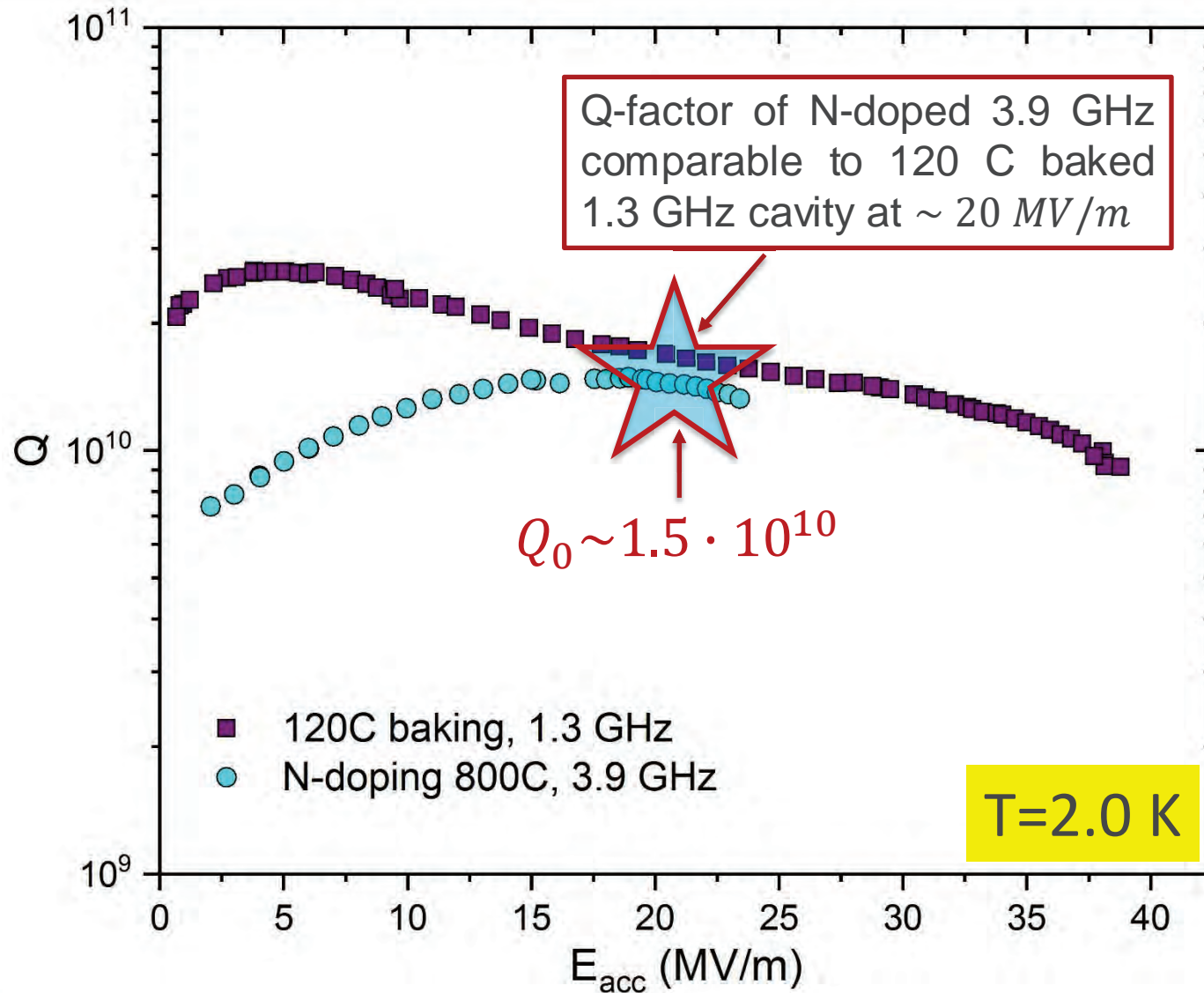


- Higher the frequency, higher the probability to match the condition for non-equilibrium
- N-doping (surface treatments in general) may modify τ_{ep} and τ_r enhancing non-equilibrium effects

High Frequency Cavities Favorable at High Field



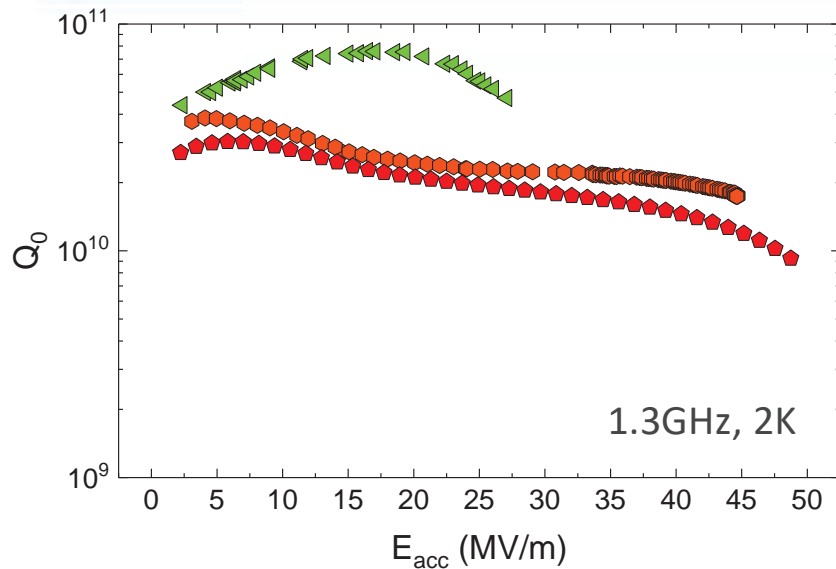
Unprecedented Medium Field Q_0 at 3.9 GHz



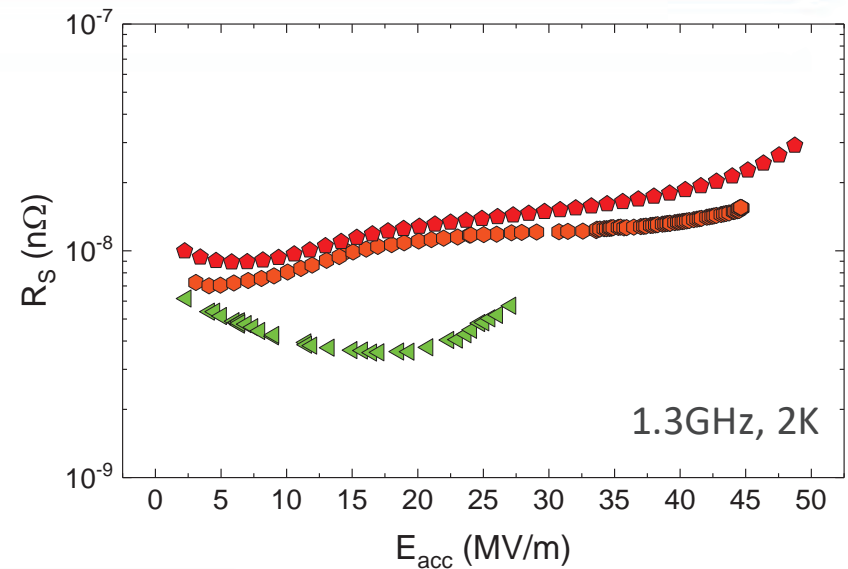
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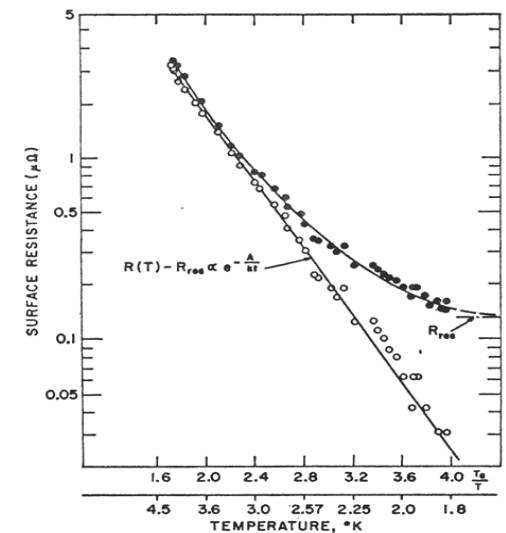
$$Q_0 = \frac{G}{R_S}$$



$$R_S = R_{BCS} + R_{fl} + R_0$$

Residual resistance

- Constant contribution which does not depend on temperature
- R_{fl} (trapped flux surface resistance) depends on magnetic field trapped during SC transition
- R_0 depends on defects, precipitates, etc..



Trapped flux surface resistance

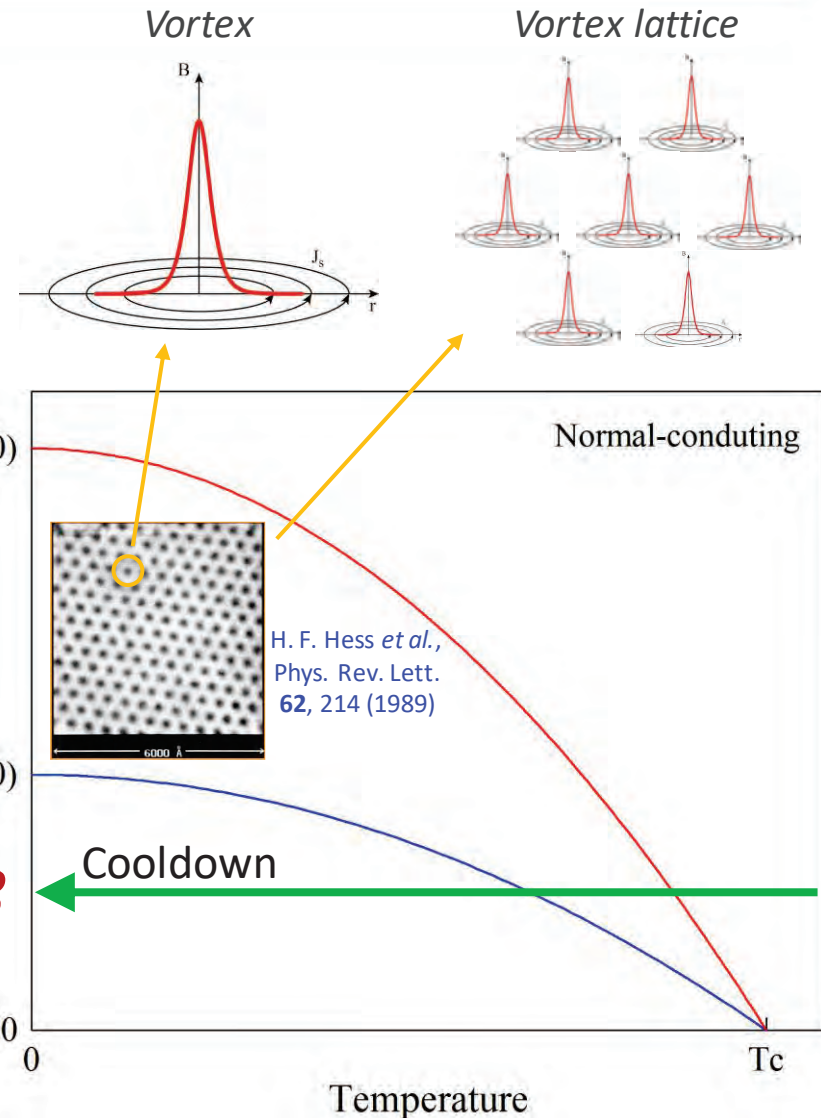
If pinned, vortices may survive in the Meissner state introducing dissipation:

$$R_{fl} = \eta_t \cdot S \cdot B$$

η_t —flux trapping efficiency

S —trapped flux sensitivity

B —external magnetic field



Trapped flux surface resistance

If pinned, vortices may survive in the Meissner state introducing dissipation:

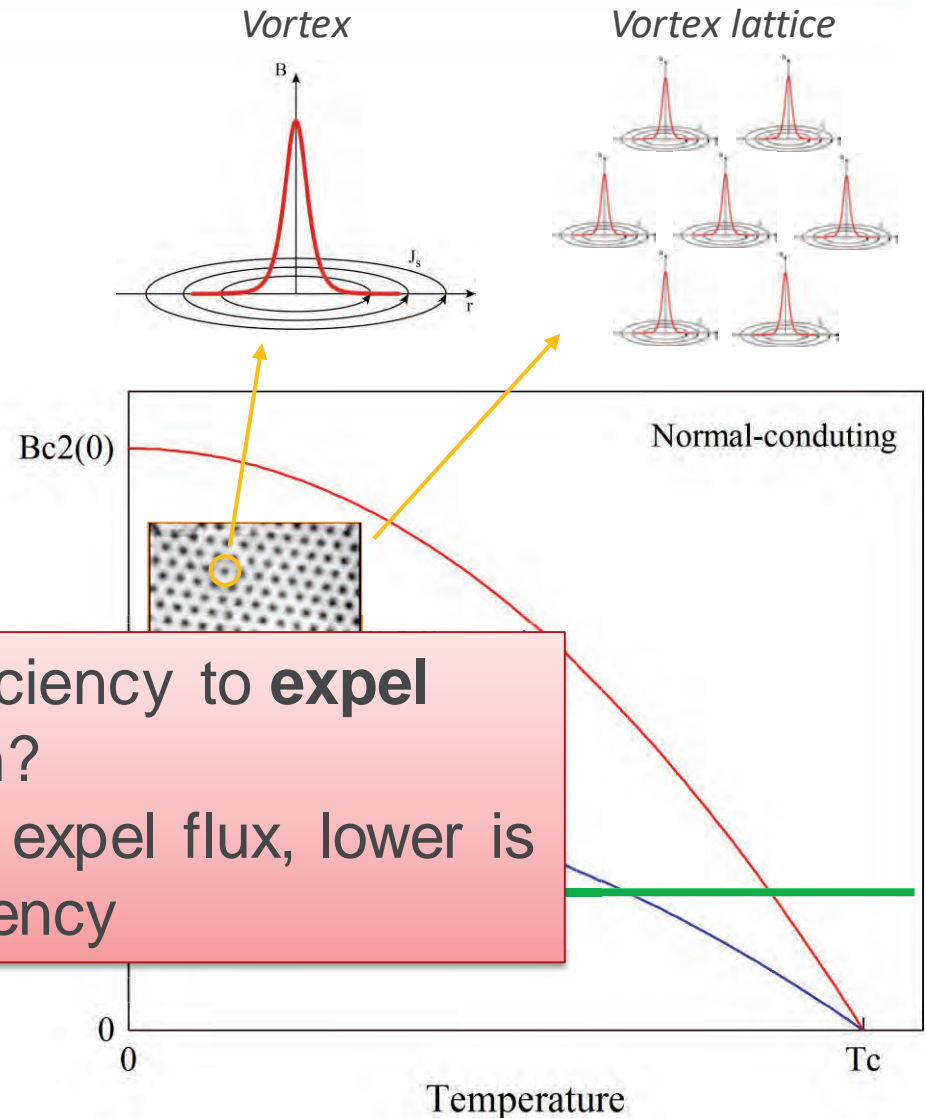
$$R_{fl} = \eta_t \cdot S \cdot B$$

η_t —flux trapping efficiency

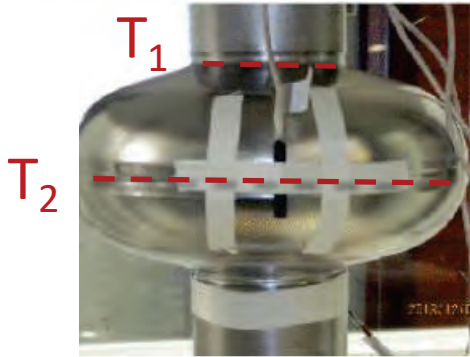
S —
 B —

What is the cavity ability/efficiency to **expel flux** during the SC transition?

- Higher is the ability to expel flux, lower is the flux trapping efficiency



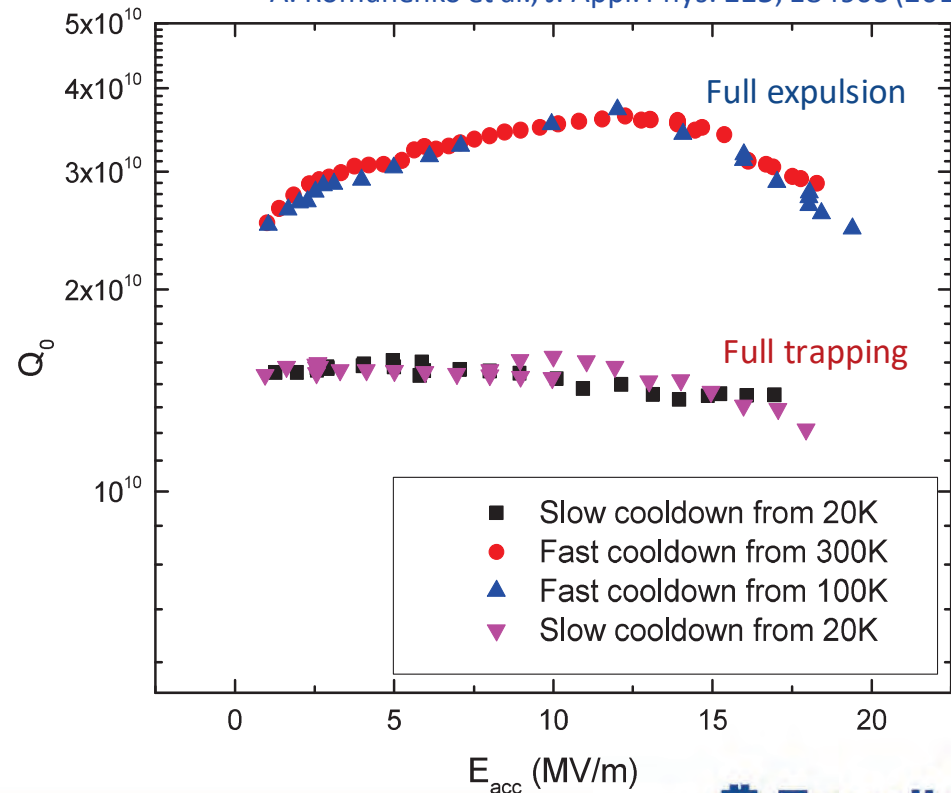
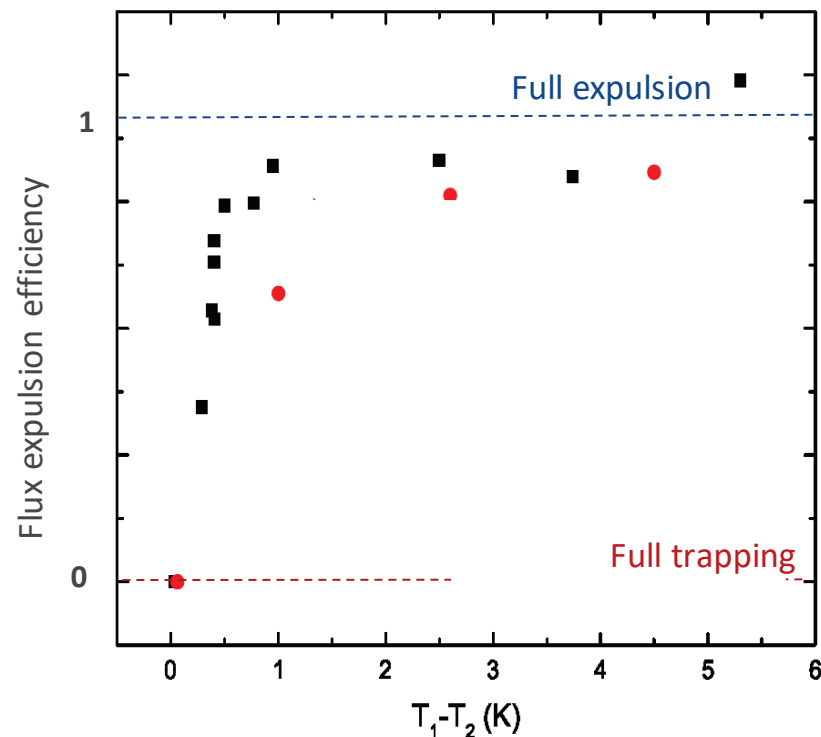
Fast cooldown helps flux expulsion



- **Fast cool-down** leads to large thermal gradients → efficient flux expulsion
- **Slow cool-down** leads to small thermal gradients → poor flux expulsion

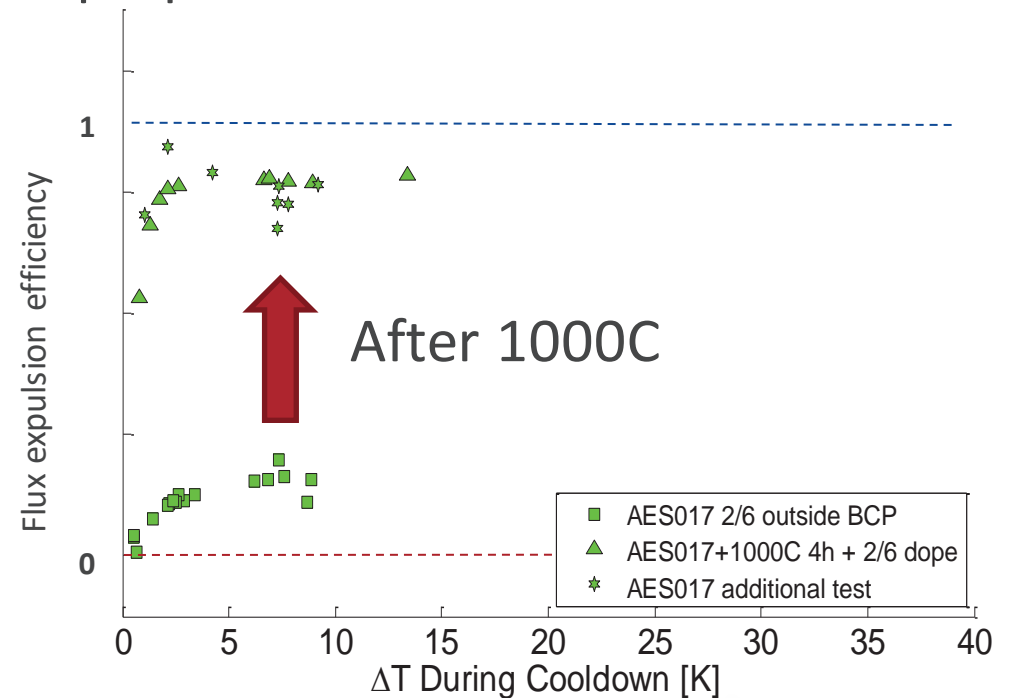
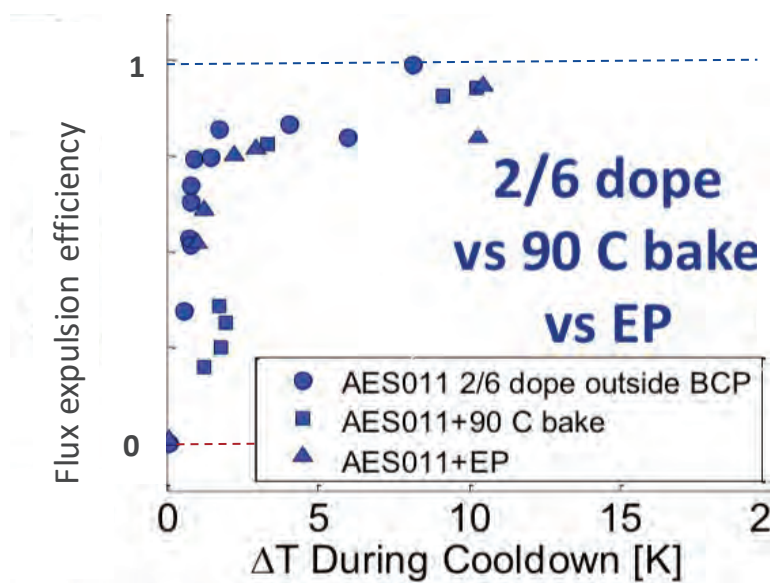
A. Romanenko et al., Appl. Phys. Lett. **105**, 234103 (2014)

A. Romanenko et al., J. Appl. Phys. **115**, 184903 (2014)



Flux expulsion depends on bulk properties of Nb cavities

- Flux expulsion is a bulk property → does not depend on surface treatment
- Not all materials show good flux expulsion, even with large thermal gradient during the SC transition → high T treatments allow to improve materials flux expulsion properties

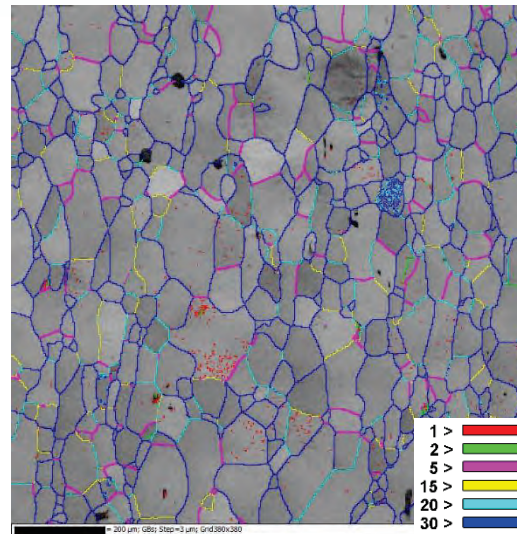


S. Posen et al., J. Appl. Phys. **119**, 213903 (2016)

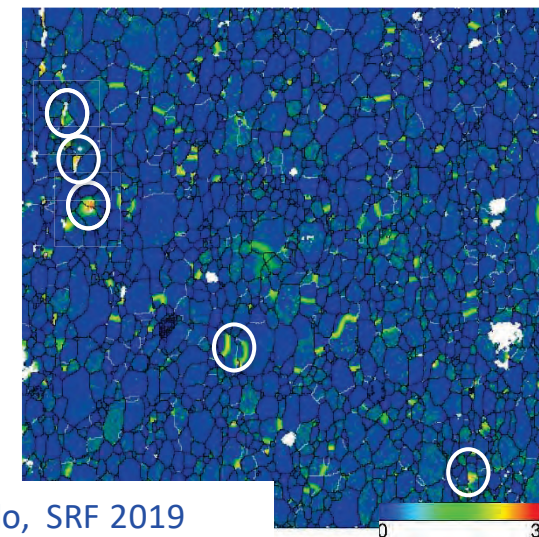
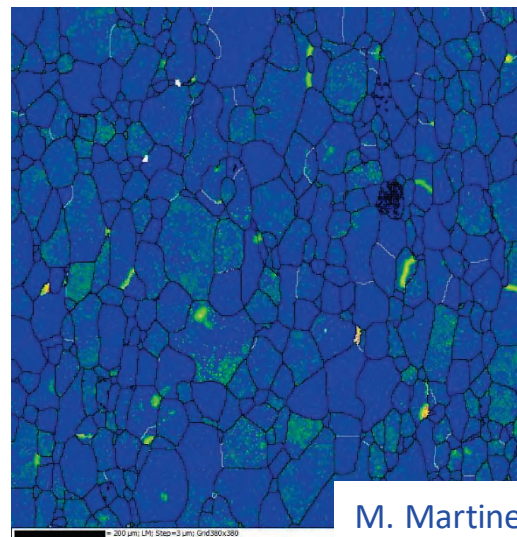
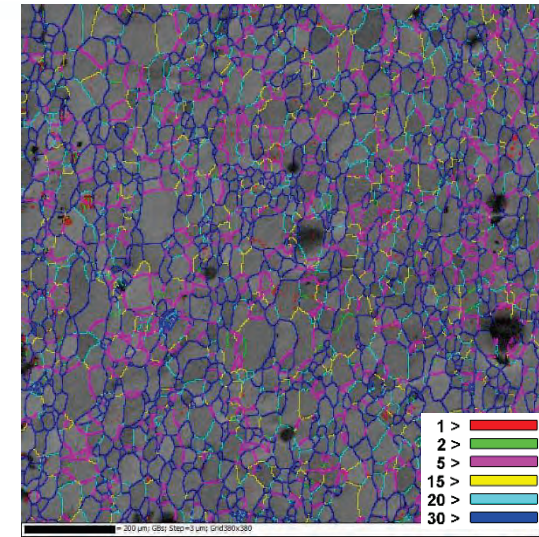
Analysis of “as received” materials

- Material that shows **good flux expulsion** properties after annealing at 800C has bigger grain size in the “as received” condition
- Material with **bad flux expulsion** properties shows larger density of low-angle GBs (misorientation $< 15^\circ$)
- Material with bad flux expulsion properties shows larger density of regions with very high local misorientation

ATI -good flux expulsion-



Ningxia -bad flux expulsion-



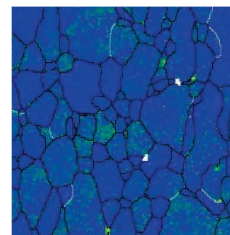
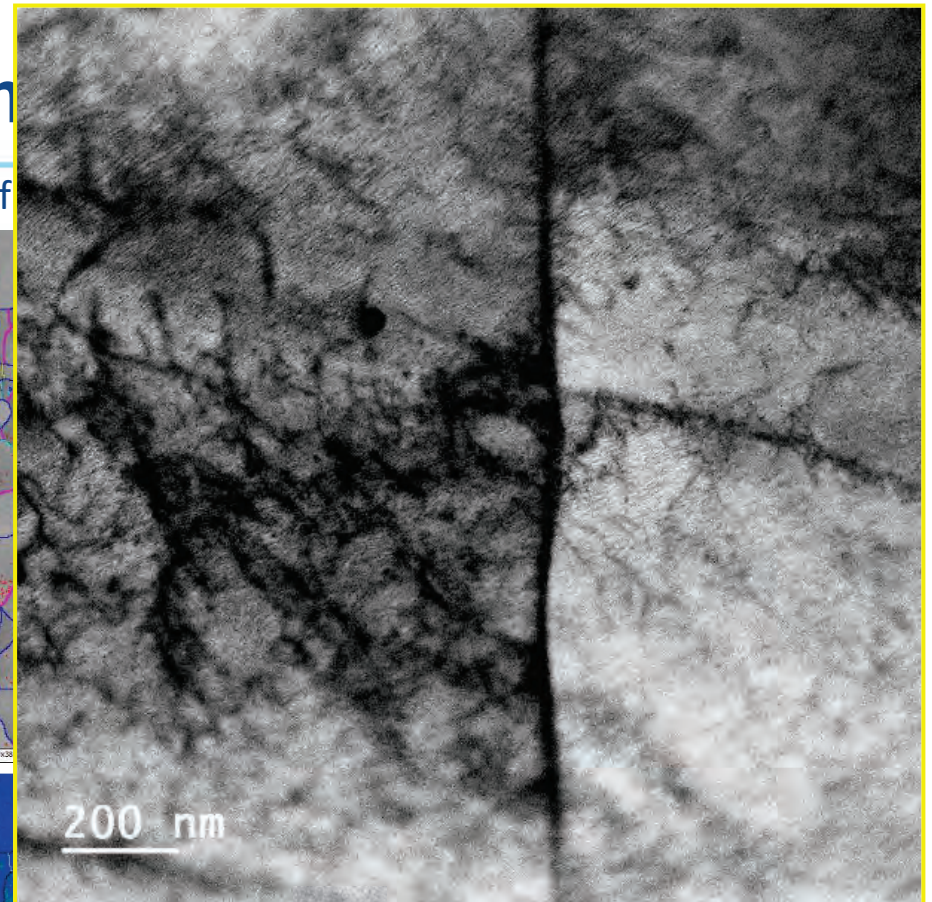
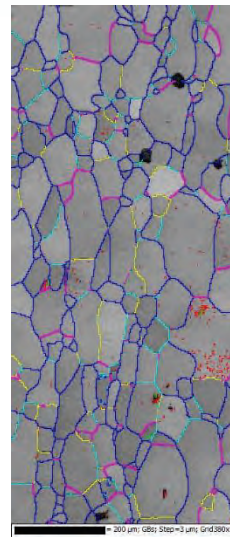
M. Martinello, SRF 2019



Analysis of “as received” material

- Material that shows **good flux expulsion** properties after annealing at 800C has bigger grain size in the “as received” condition
- Material with **bad flux expulsion** properties shows larger density of low-angle GBs (misorientation $< 15^\circ$)
- Material with bad flux expulsion properties shows larger density of regions with very high local misorientation
-> **dislocations tangles**

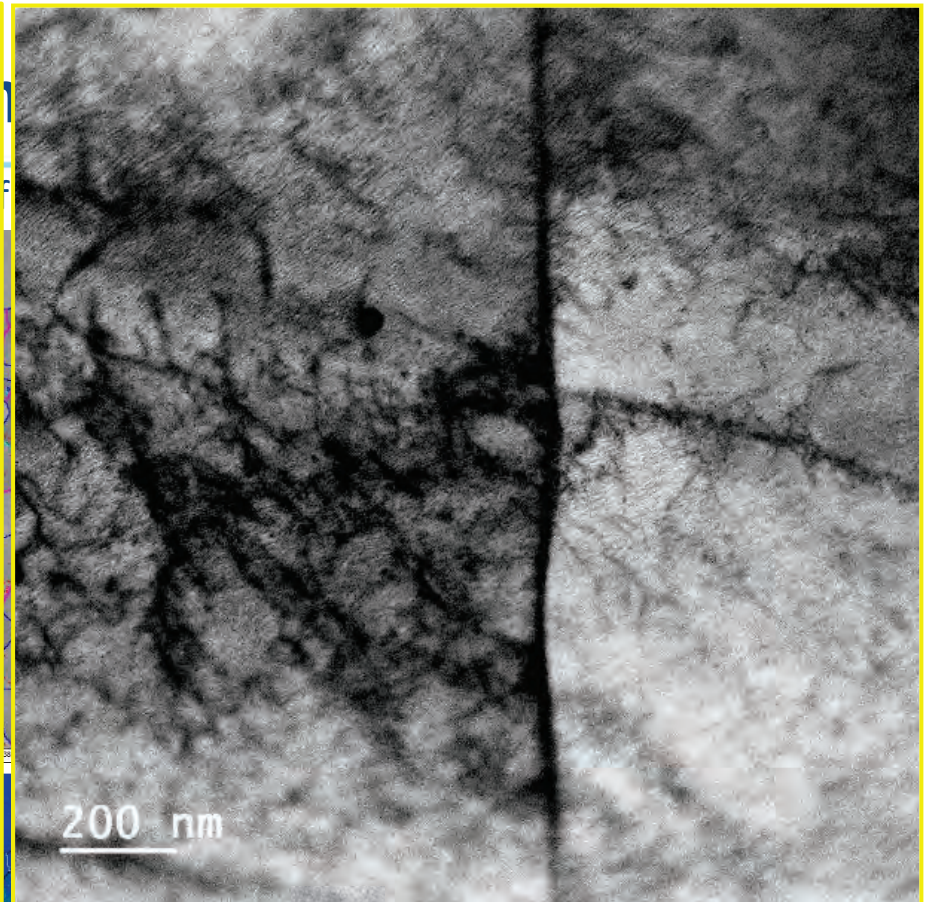
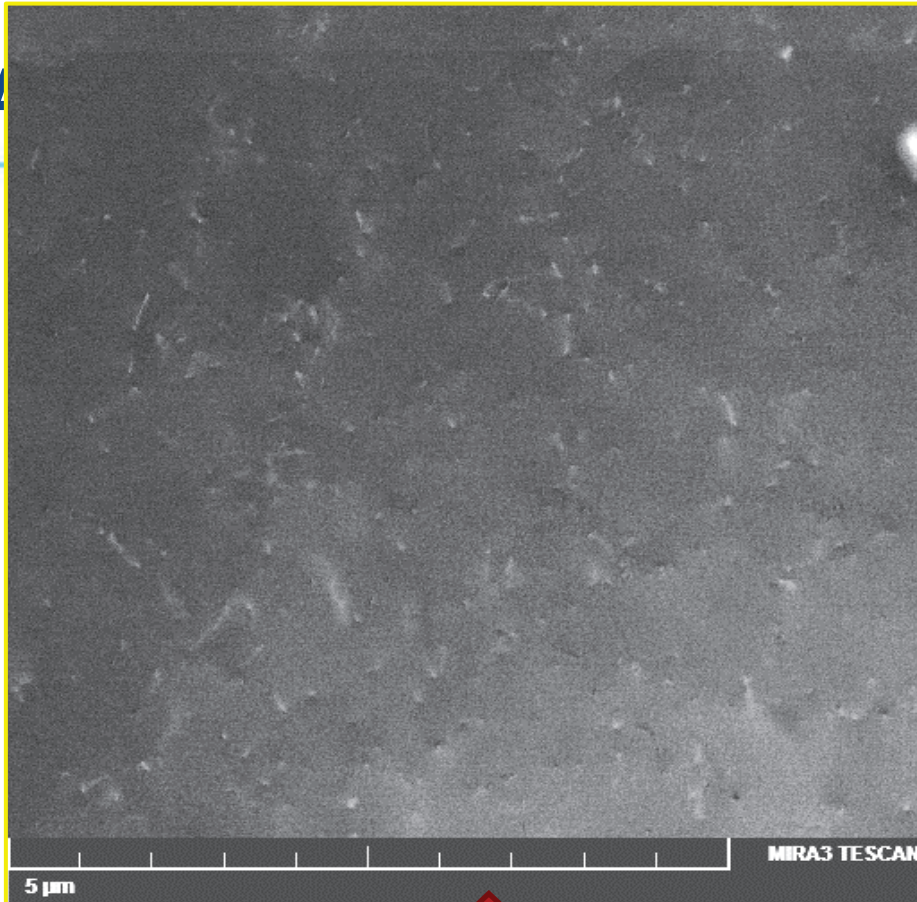
ATI -good flux



Dislocations tangles observed in highly defective regions of as-received material with bad flux expulsion

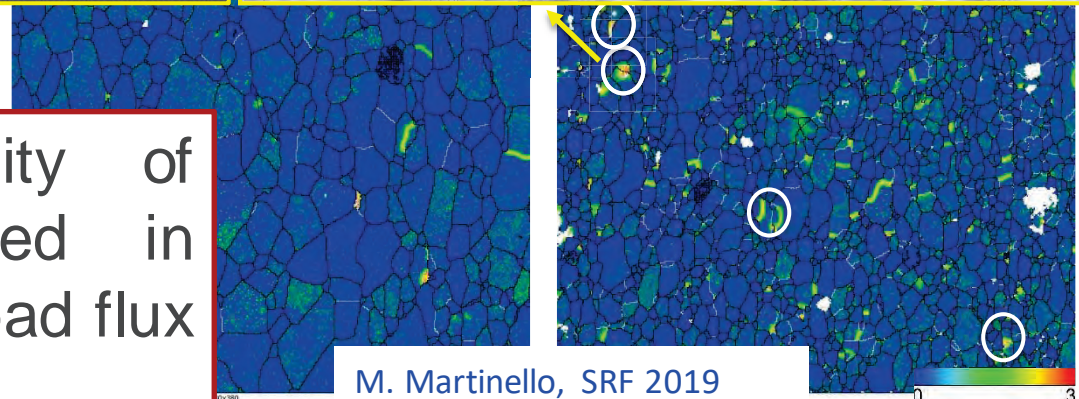
M. Martinello, SRF 2019





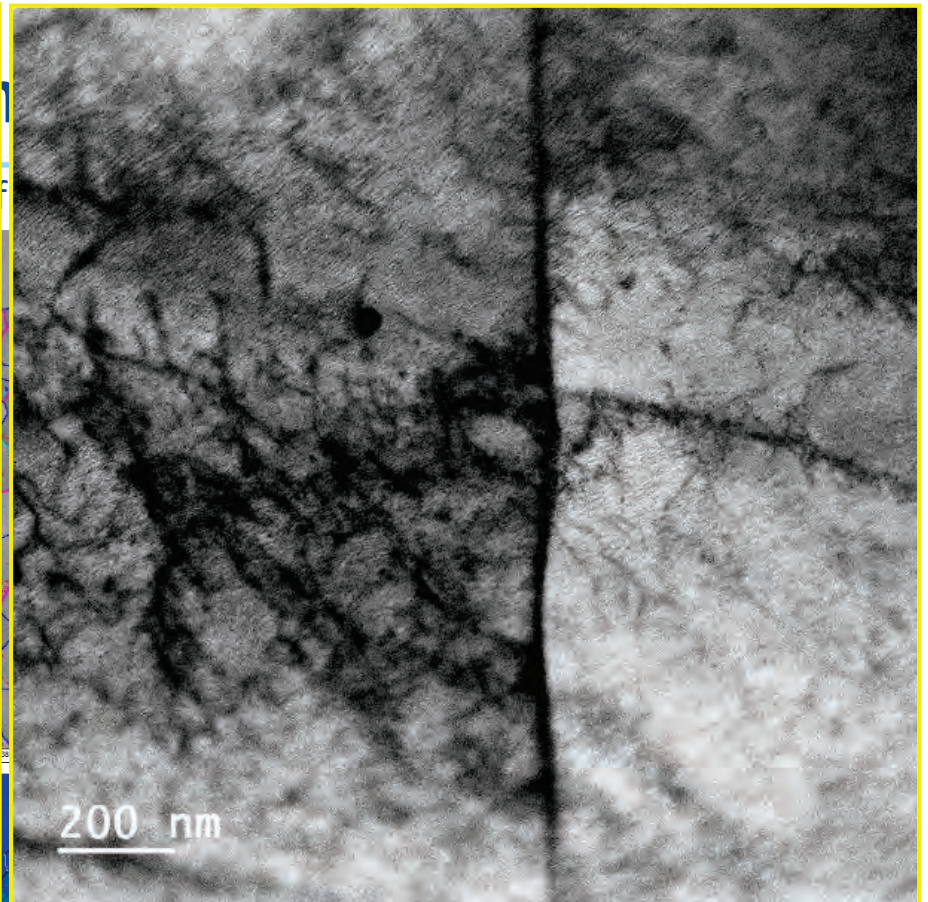
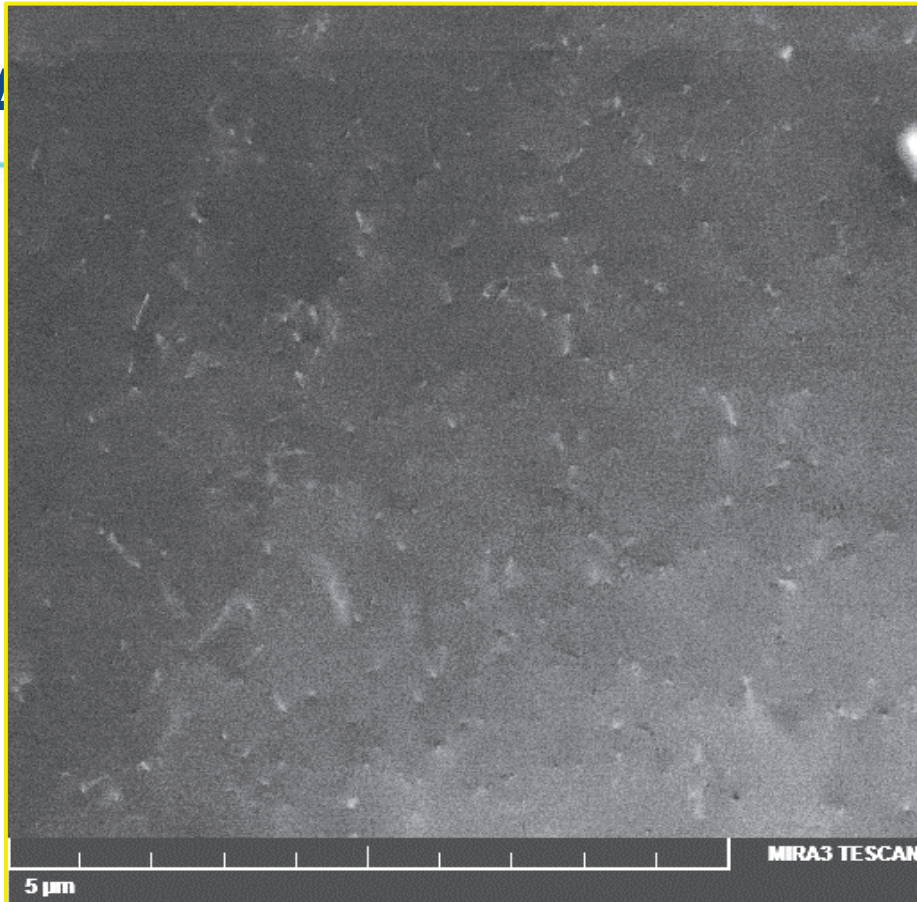
- Material with low flux expulsion properties shows

Lots of high-density of dislocations observed in **cavity cut-out** with bad flux expulsion



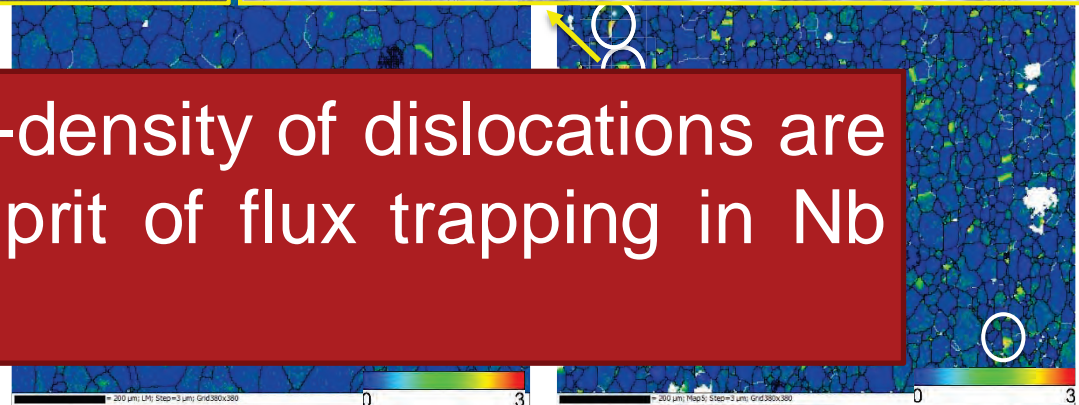
M. Martinello, SRF 2019





- Material with bad flux
- expulsion
- larger
- with
- misorientation
- > dislocation tangles

Area with high-density of dislocations are most likely culprit of flux trapping in Nb cavities



Trapped flux surface resistance

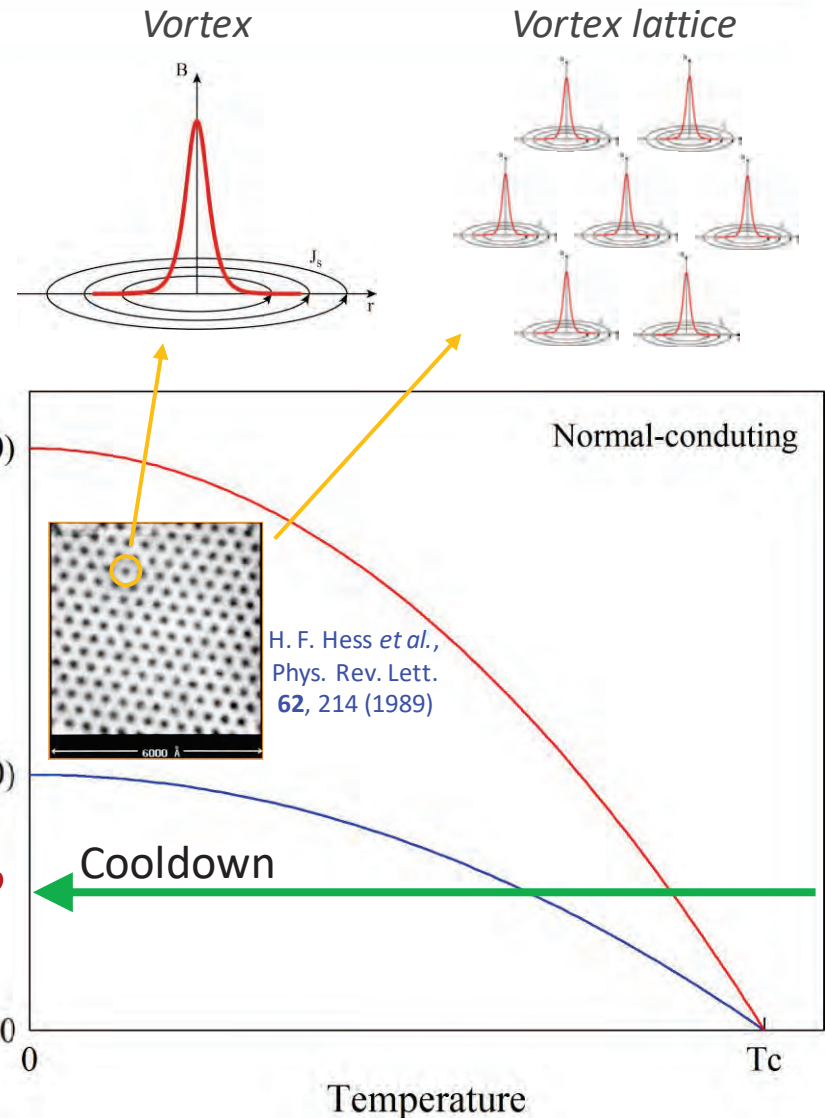
If pinned, vortices may survive in the Meissner state introducing dissipation:

$$R_{fl} = \eta_t \cdot S \cdot B$$

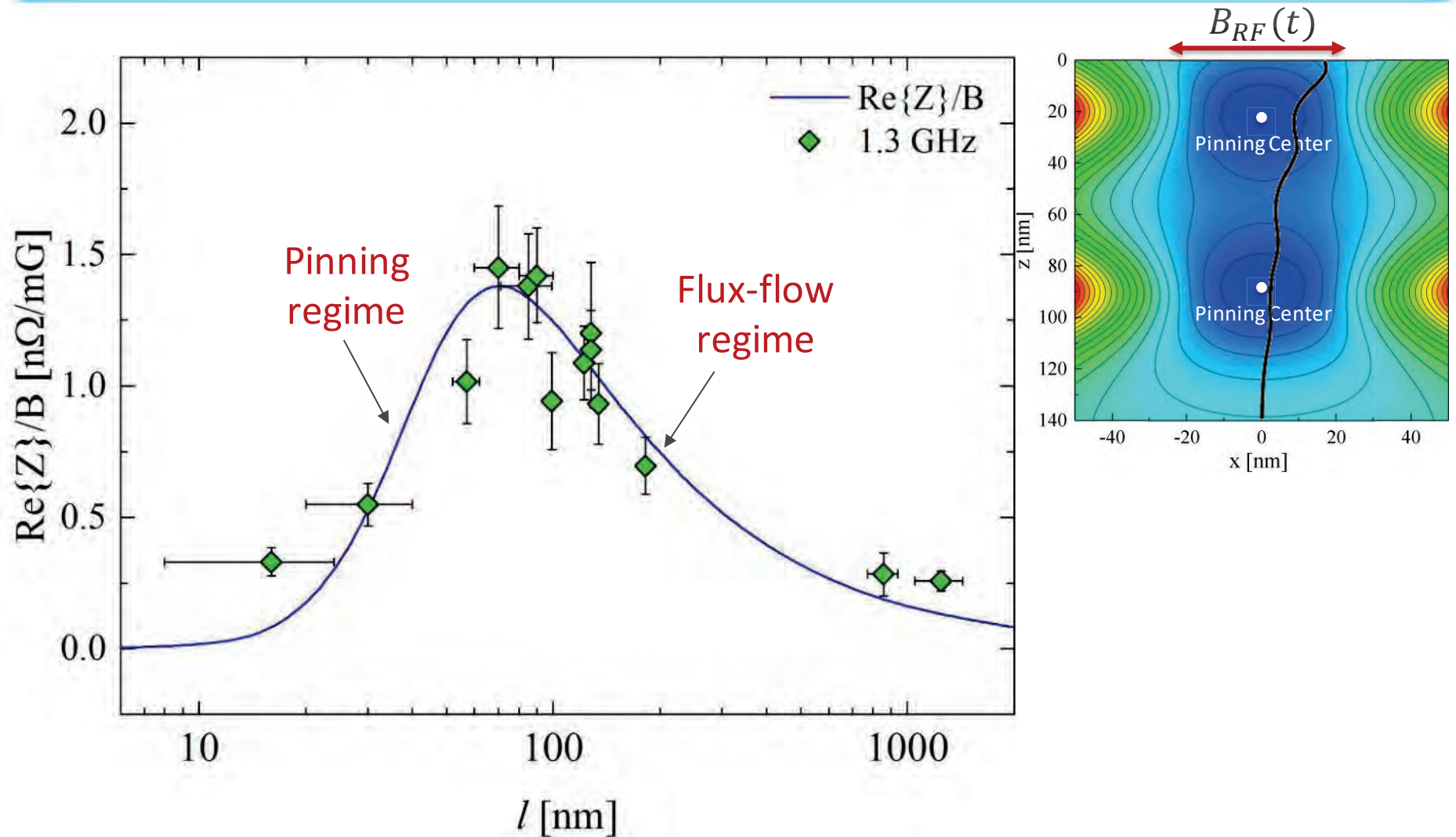
η_t —flux trapping efficiency

S —trapped flux sensitivity

Trapped flux sensitivity = losses (due to trapped flux) normalized for the amount of flux trapped



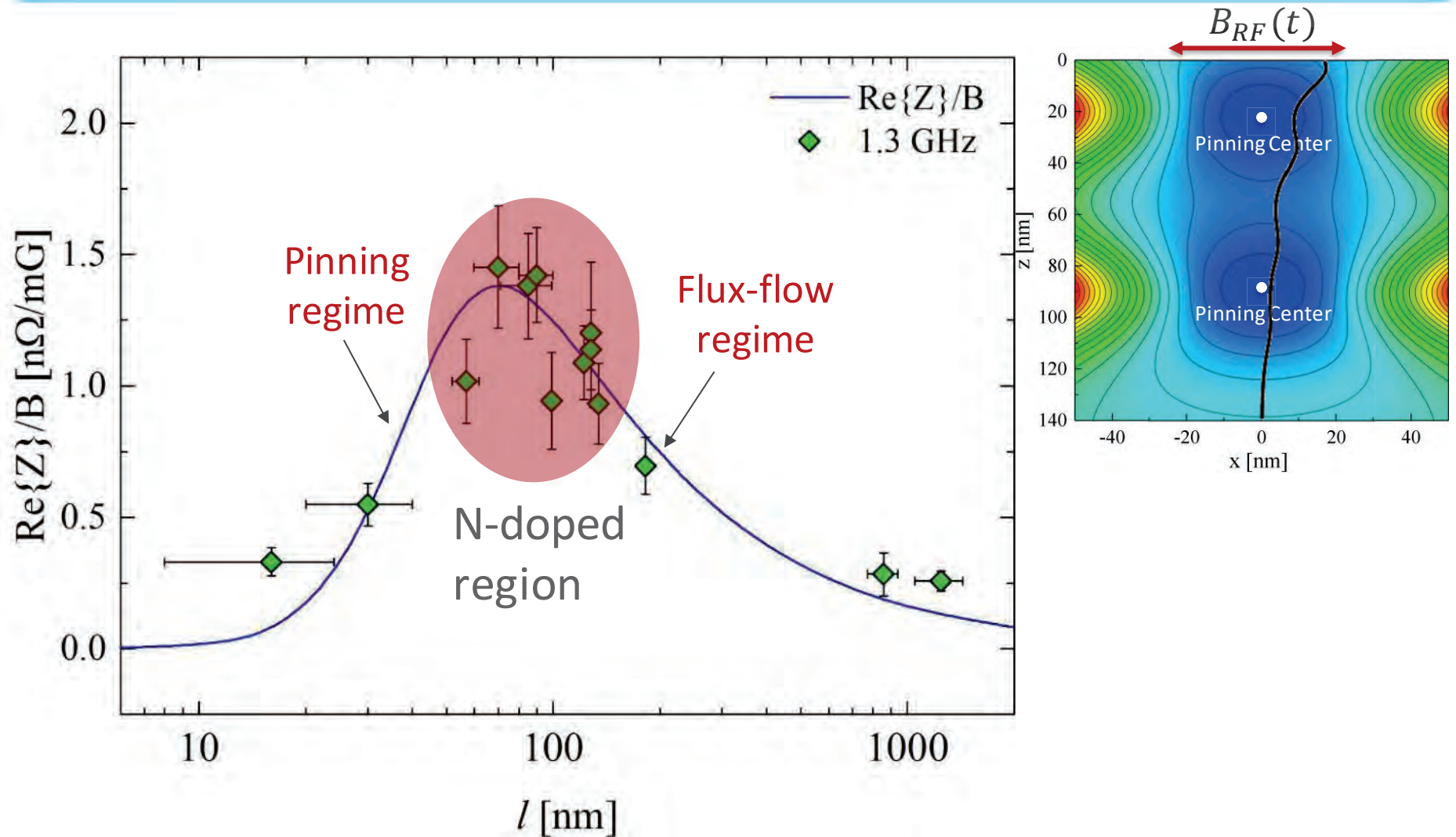
Sensitivity versus mean-free-path



M. Martinello et al., App. Phys. Lett. **109**, 062601 (2016)

M. Checchin et al., Supercond. Sci. Technol. **30**, 034003 (2017)

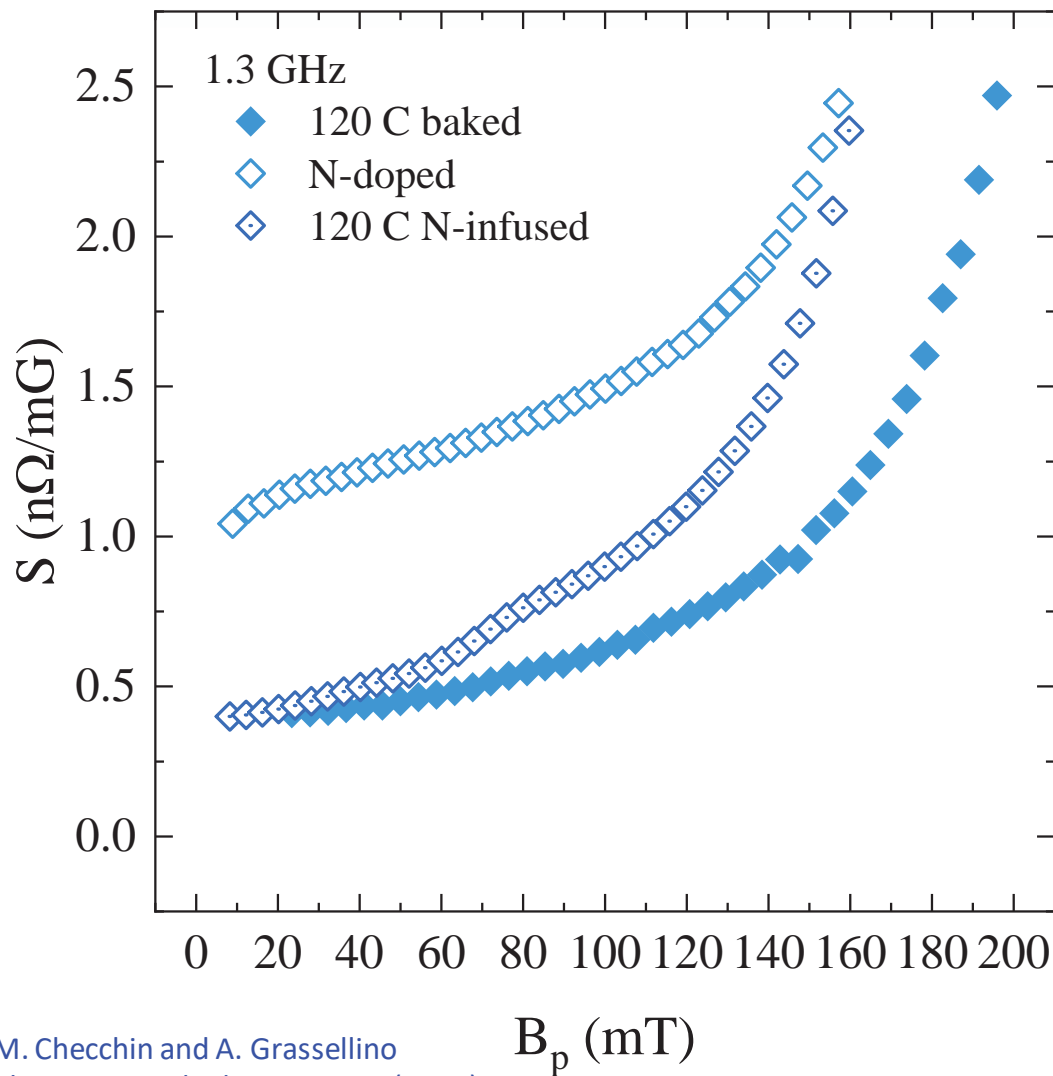
Sensitivity versus mean-free-path



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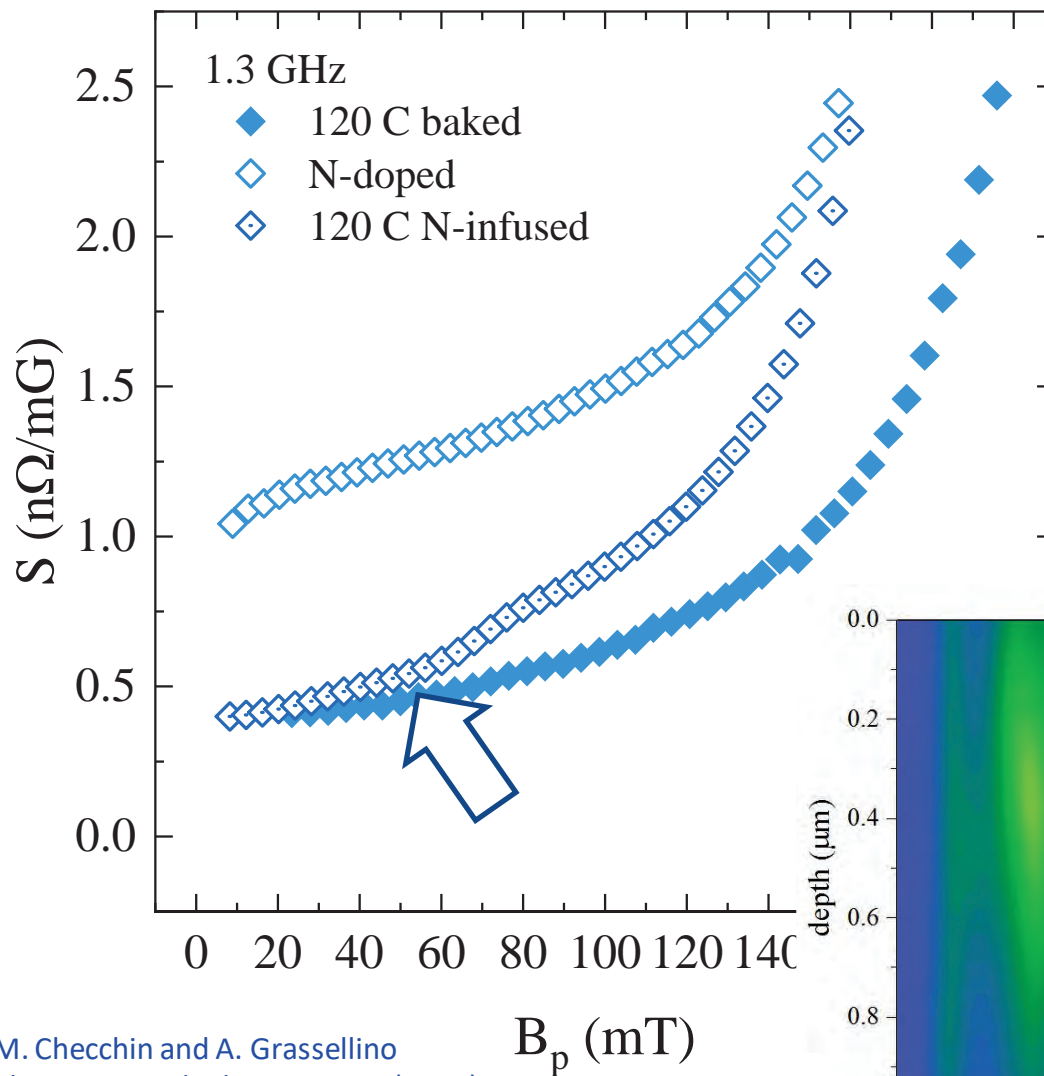
Sensitivity as a function of the field



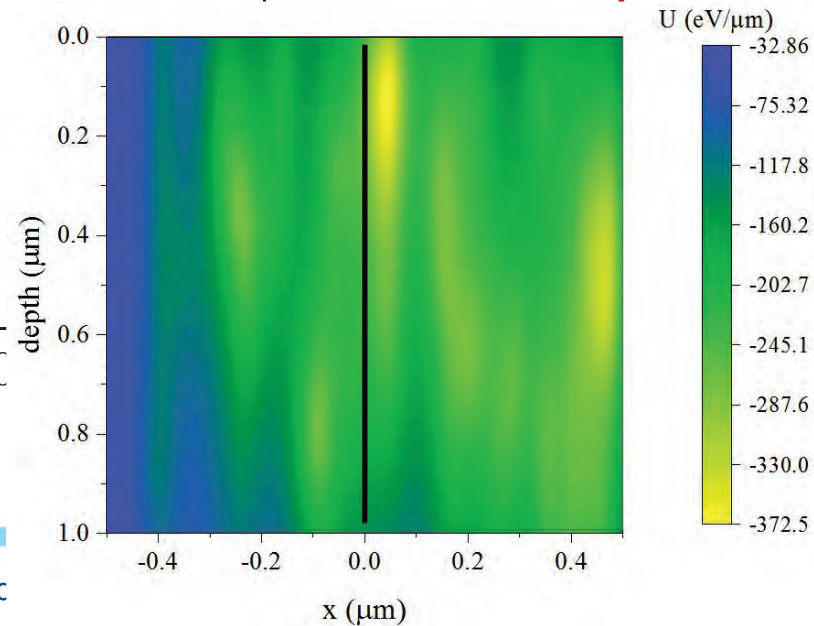
M. Checchin and A. Grassellino
Phys. Rev. Applied 14, 044018 (2020)

- At high field the trapped flux sensitivity increases exponentially
- 120C baked and N-infused cavities have very low values of sensitivity at low field but significantly high at high field
- Important to take that into account for high-field applications!

Sensitivity as a function of the field



- At high field the trapped flux sensitivity increases exponentially
- 120C baked and N-infused cavities have very low values of sensitivity at low field



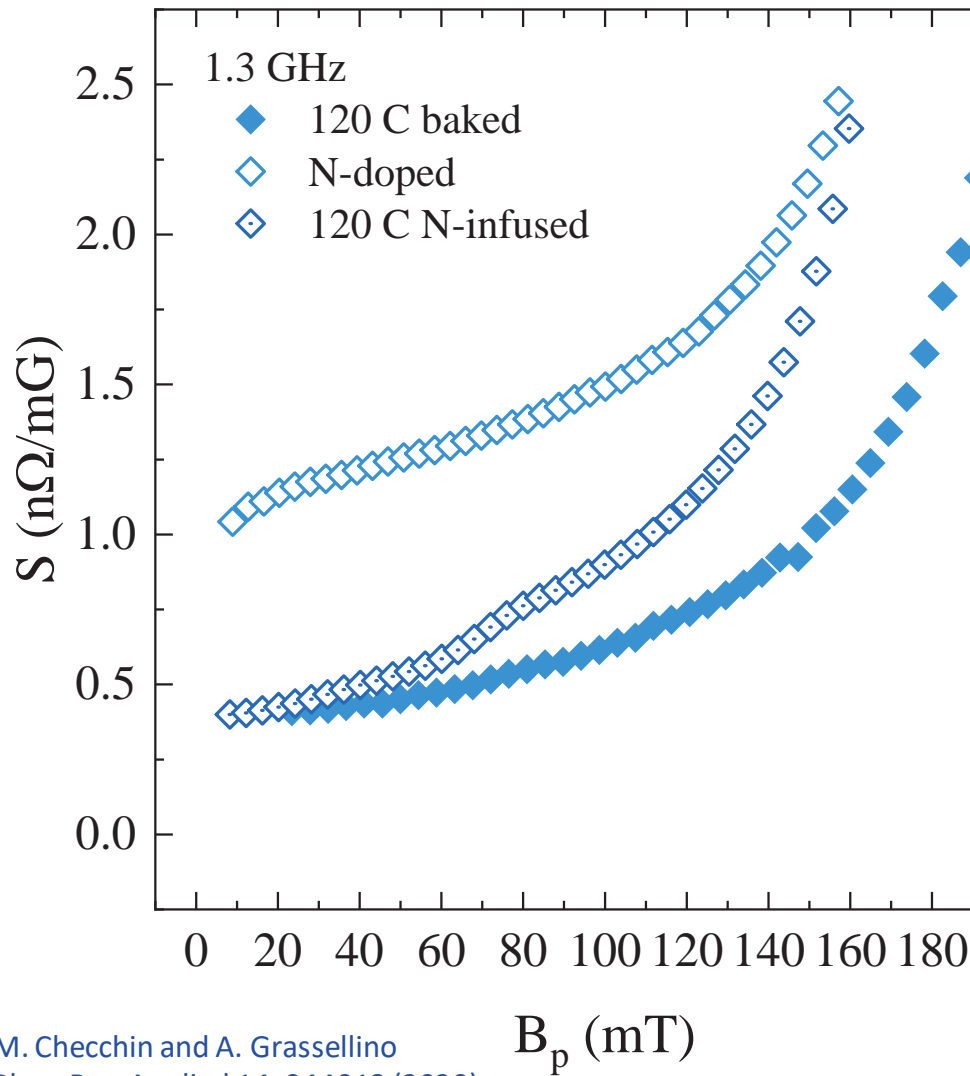
gh at

that high-

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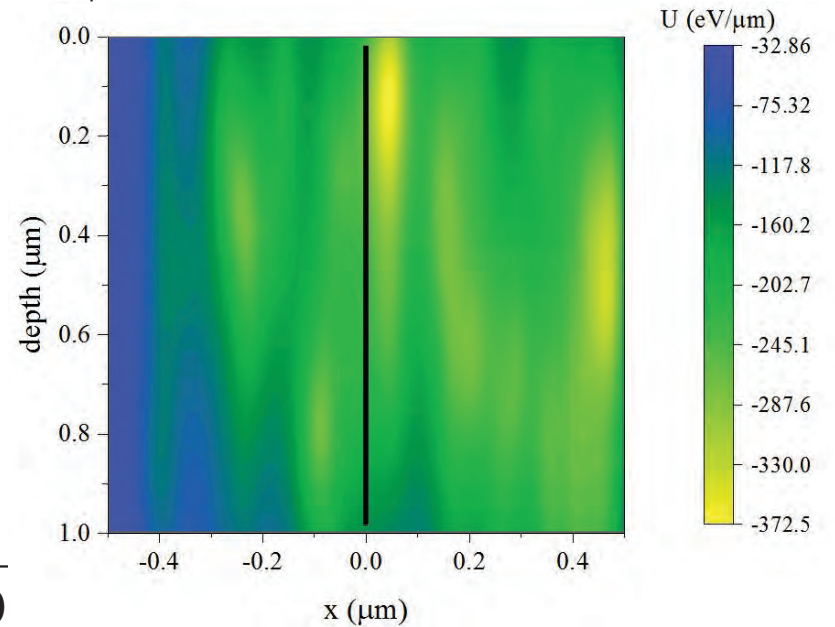
M. Checchin and A. Grassellino
Phys. Rev. Applied 14, 044018 (2020)

Sensitivity as a function of the field



- At high field the trapped flux sensitivity increases exponentially

- 120C baked and N-



field applications!

M. Checchin and A. Grassellino
Phys. Rev. Applied 14, 044018 (2020)

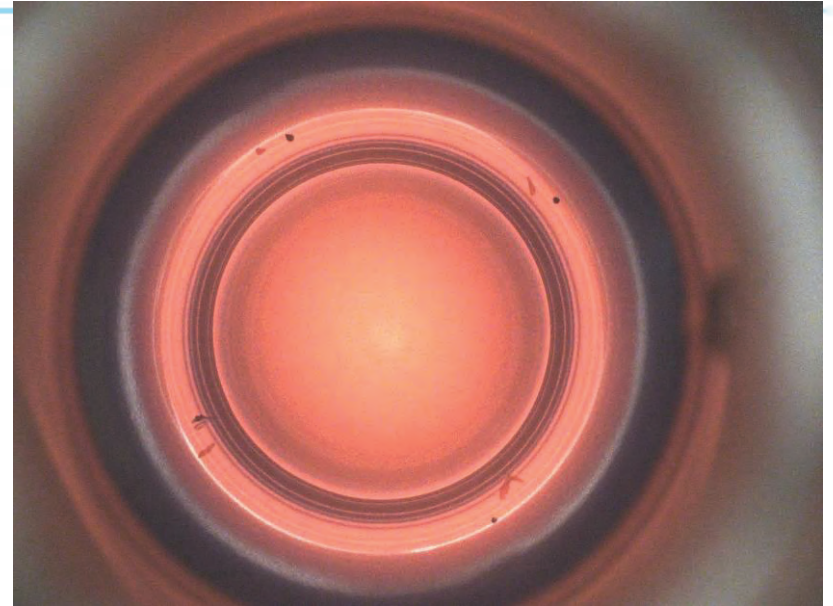


Outline

- Introduction to SRF cavities
- State-of-the art surface treatments
- Toward a better understanding
 - BCS surface resistance: new insights and optimization
 - Residual resistance: understanding and minimizing degradation due to trapped flux
- **Technology improvement for SRF-based projects**
- Conclusions

Plasma processing R&D for LCLS-II cavities

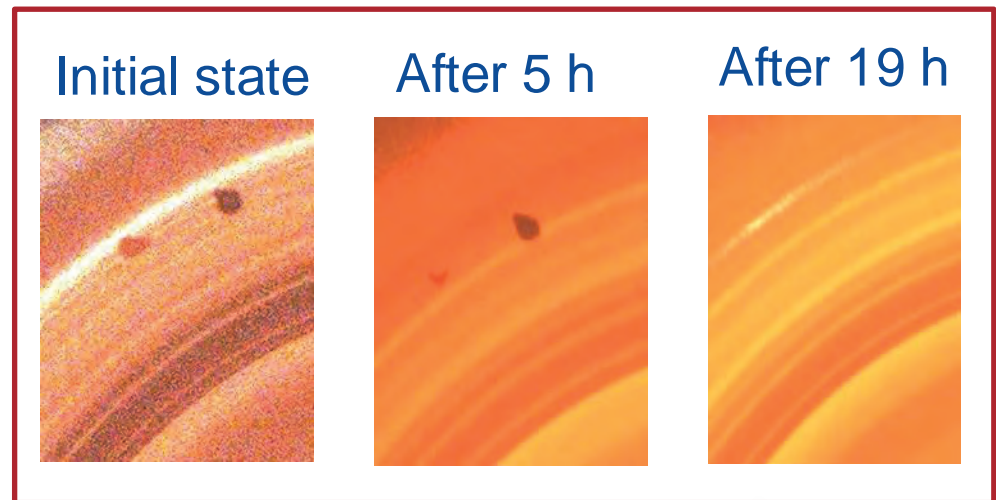
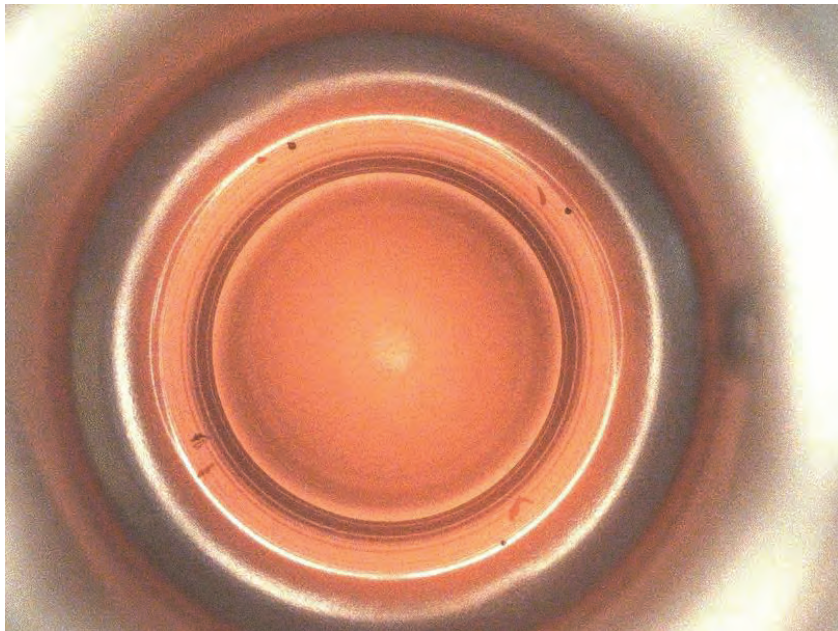
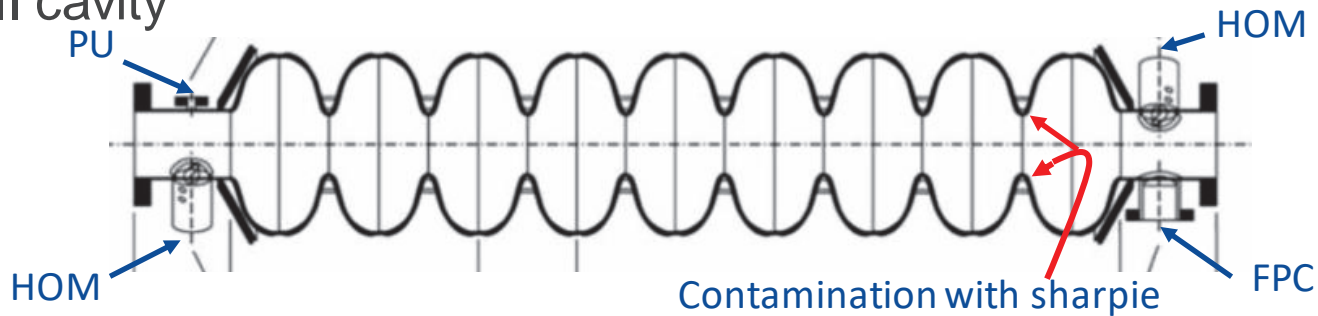
Work performed in collaboration with SLAC and ORNL



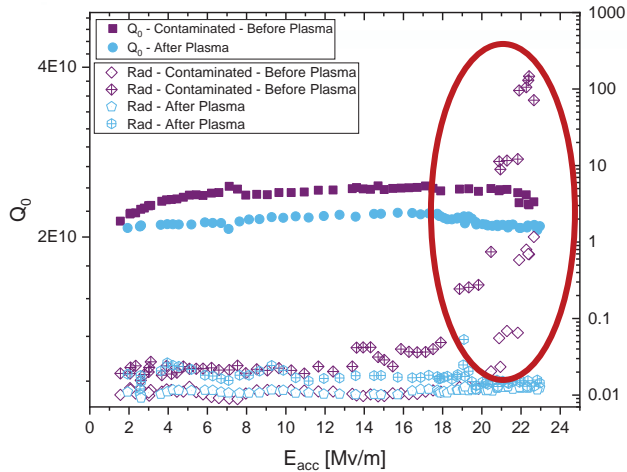
- Ne-O gas mixture (few % of O₂, mostly Ne) at p ~ 75-150mTorr
- Plasma ignition sequentially cell-by-cell using HOMs antenna (10-50 W for ignition, few W for sustain plasma) [P. Berrutti et al., J. App. Phys. 126, 023302 \(2019\)](#)
- Approximately 1-2 hours processing per cell
- On-going R&D to optimize parameters and understand rate of success in naturally FE cavities [B. Giaccone, M. Martinello et al. Phys. Rev. Accel. Beams 24, 022002\(2021\)](#)

First study in a 9-cell contaminated cavity

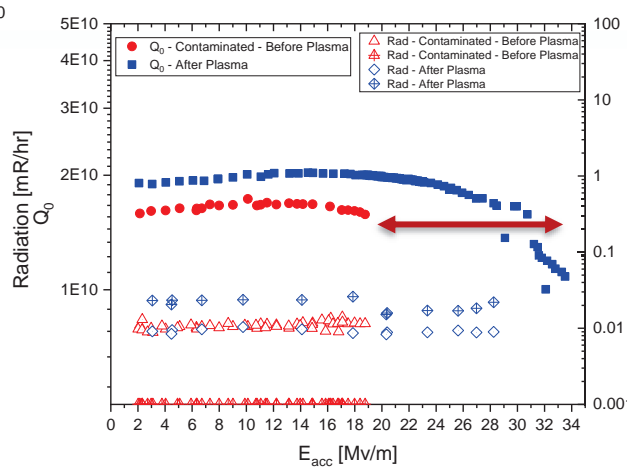
- **Scope:** remove hydrocarbon contamination with plasma cleaning
- 8 “dots” of permanent markers around the iris of the first cell of a 9-cell LCLS-II cavity



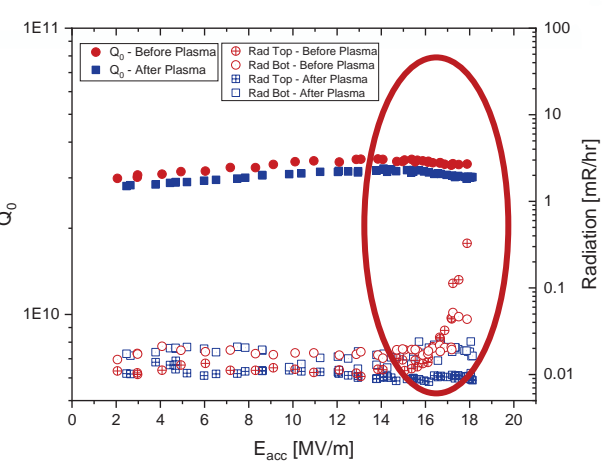
Summary results of plasma processed cavities



No radiation detected after plasma processing!

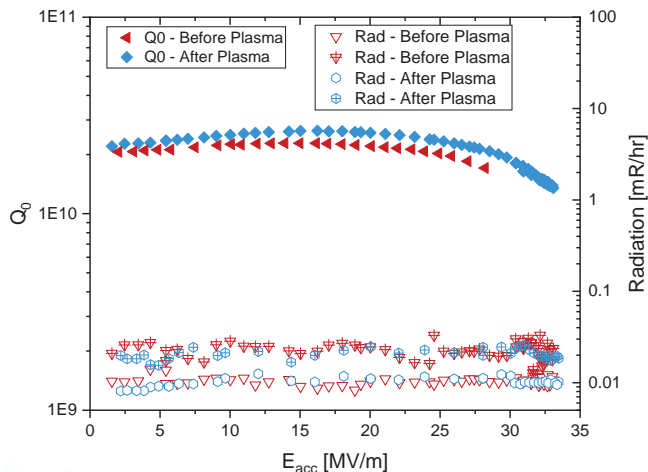


15 MV/m gained after plasma processing!



No radiation detected after plasma processing!

B. Giaccone, M. Martinello et al. Phys. Rev. Accel. Beams 24, 022002(2021)



➤ No Q degradation observed in N-doped cavities after plasma process

➤ Increasing of performance observed in 3 cavities treated with plasma processing

LCLS-II CM production at FNAL



LCLS-II production at FNAL is completed,
now time to look at what's next

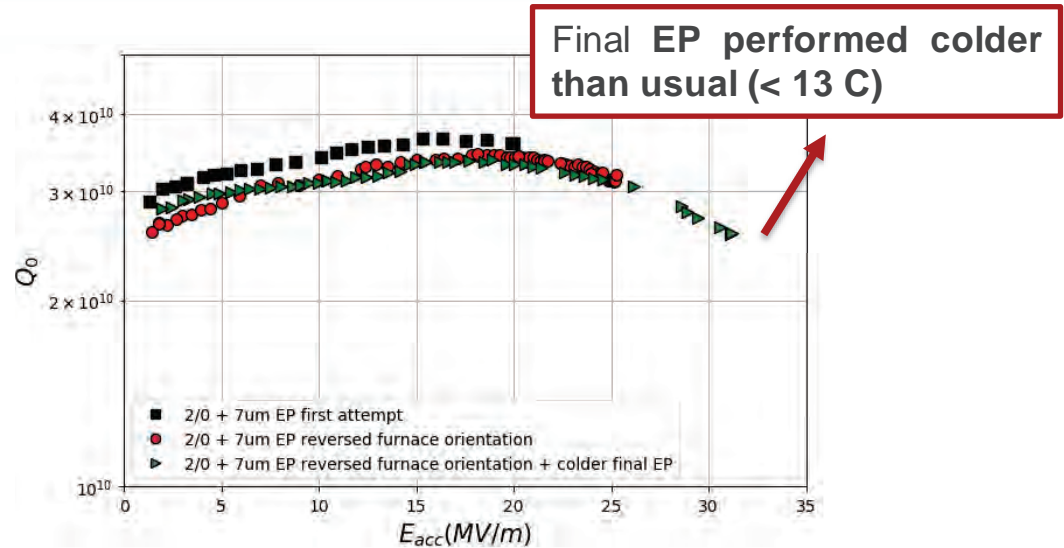
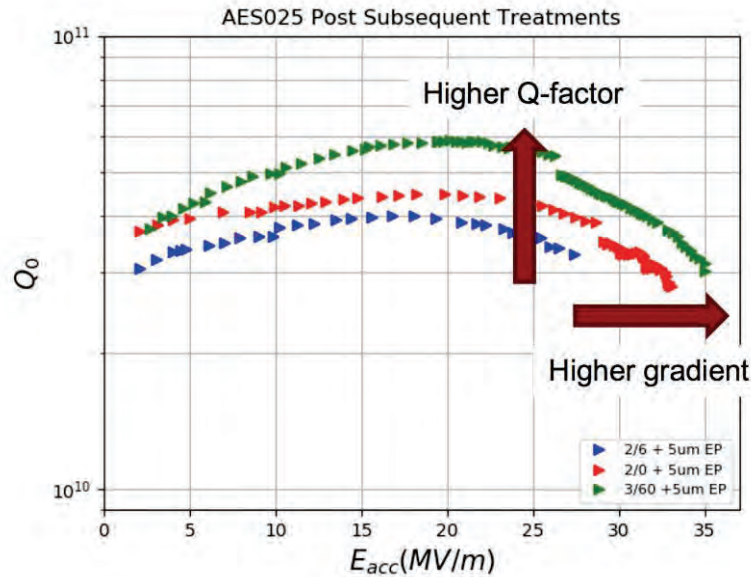
LCLS-II - High Energy



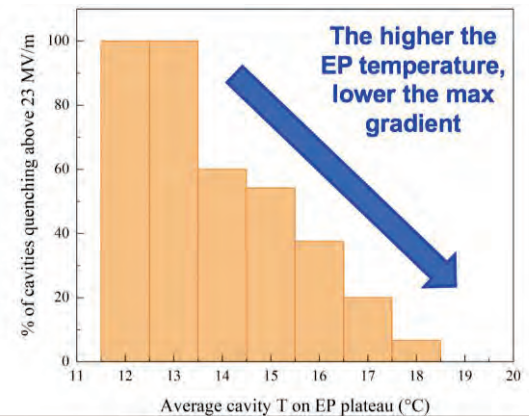
The LCLS-II-HE project will **increase the energy of the CW-SCRF linac to 8 GeV**, enabling the photon energy range to be extended to at least 13 keV and potentially up to 20 keV at 1 MHz repetition rates

Parameter	LCLS-II	LCLS-II HE
# 1.3 GHz CMs	35	20
Operating Gradient	16 MV/m	20.8 MV/m for new CMs 18 MV/m for old CMs
Required Q_0 at Operating Gradient	2.7×10^{10}	2.7×10^{10}

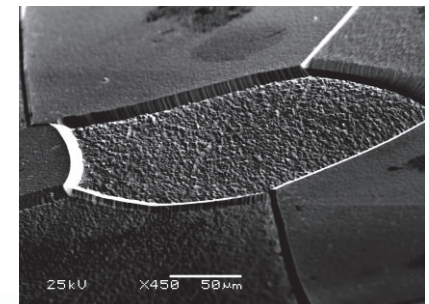
N-doping optimization for LCLS-II HE



- 2N0 recipe studied in single-cell cavities showed higher gradient compare to 2N6 (LCLS-II) recipe
- Cold EP prevents preferential etching around nitrides leading to a smoother surface finishing -> higher gradients

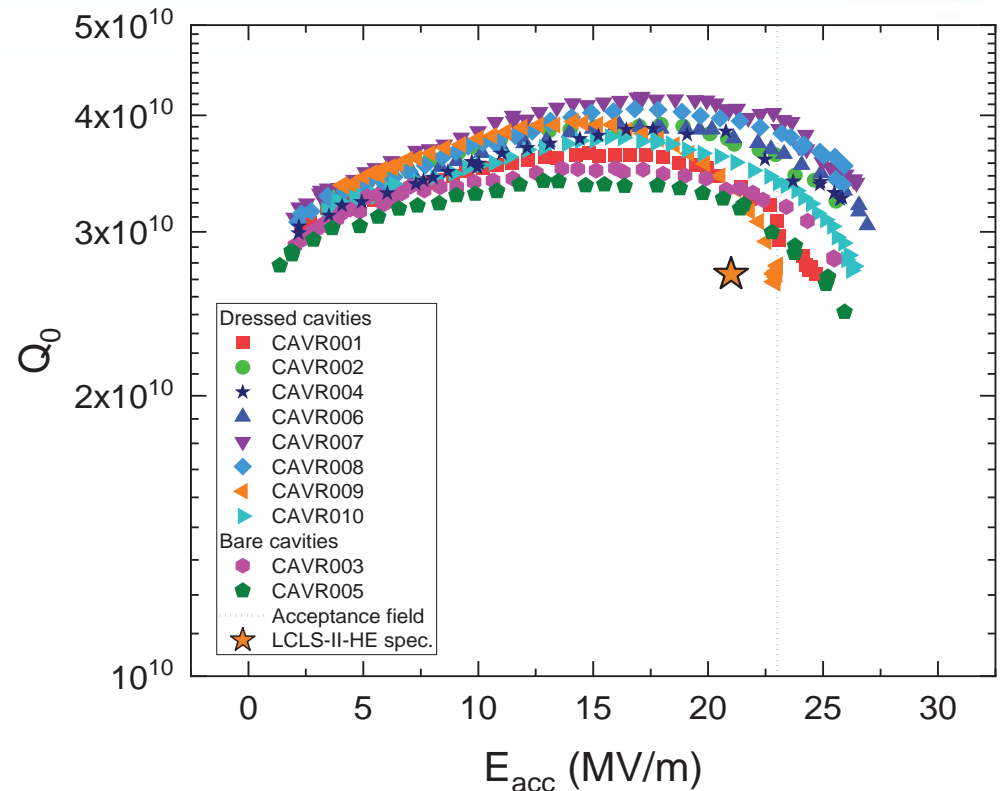


Higher density of Nb_xN_y precipitates promote preferential etching during EP forming recessed grains



Verification-CM 9-cell cavities results

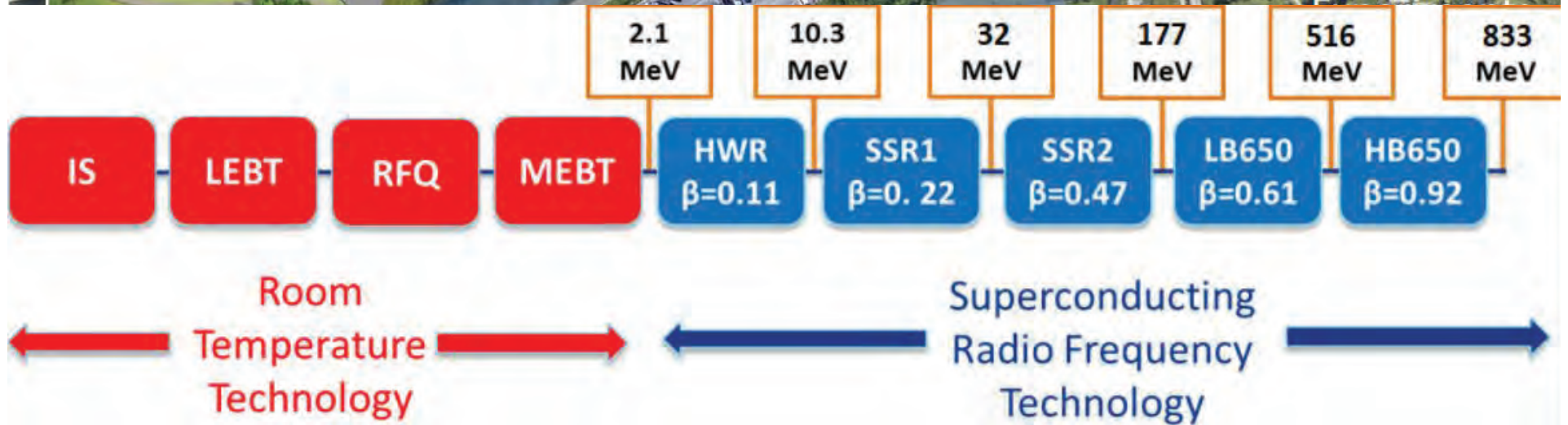
- New N-doping protocol (2N0 + cold EP) successfully transferred to industry
- Vertical test completed for 10 vCM cavities fabricated and processed in industry
- Performance exceeded specification with average $Q_0=3.6 \times 10^{10}$ and $E_{acc}=25.6$ MV/m
- 8 fully dressed cavities have been assembled in the vCM



Proton Improvement Plan at FNAL: PIP-II

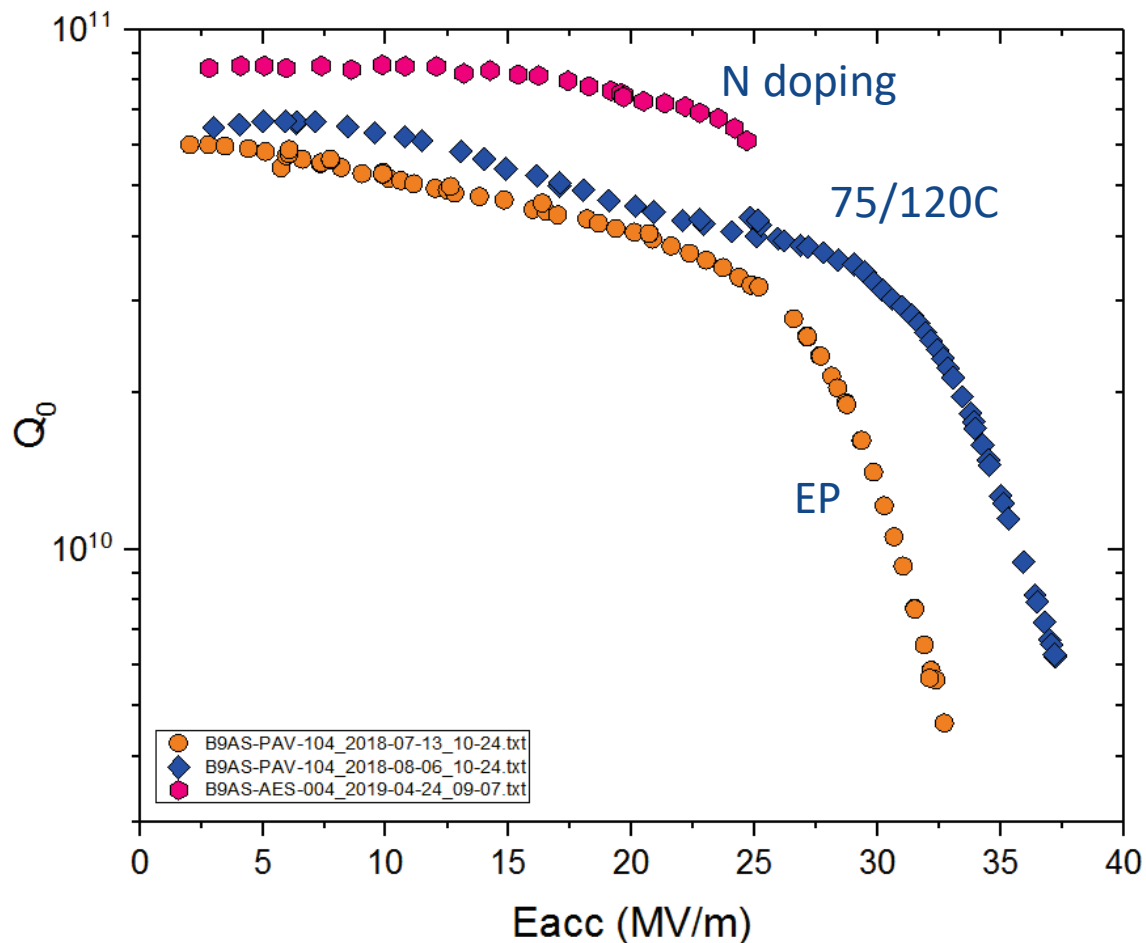


Proton Improvement Plan at FNAL: PIP-II



State-of-the-art surface treatments in HB650 cavities

Single-cell HB650, 2K



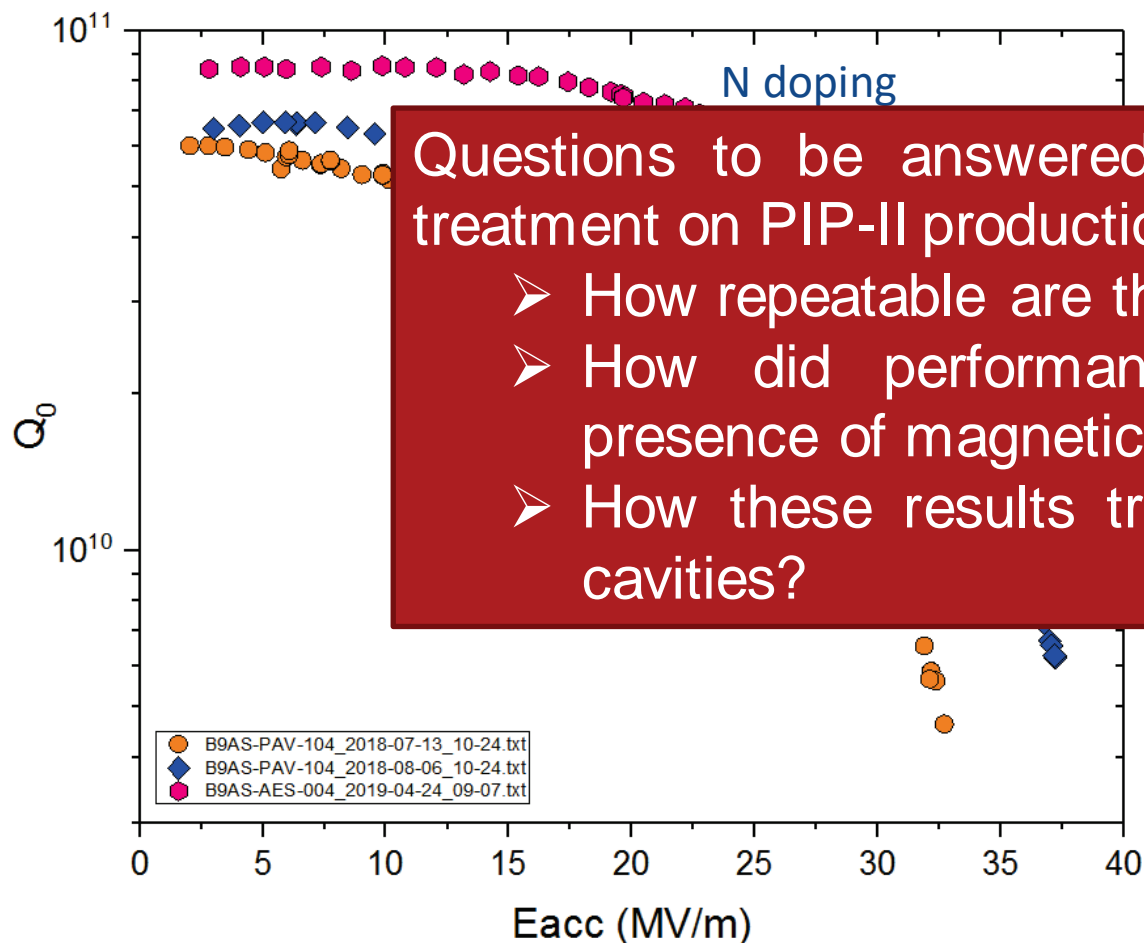
- All these treatments may exceed PIP-II specs:

N-doping	7.6e10
75-120C baking	4.7e10
EP	4e10

- N-doping gives the highest Q-factor at medium field

State-of-the-art surface treatments in 650 MHz cavities

Single-cell HB650, 2K



Questions to be answered to finalize the treatment on PIP-II production cavities:

- How repeatable are these results?
- How did performance degrade in presence of magnetic field?
- How these results translate in 5-cell cavities?

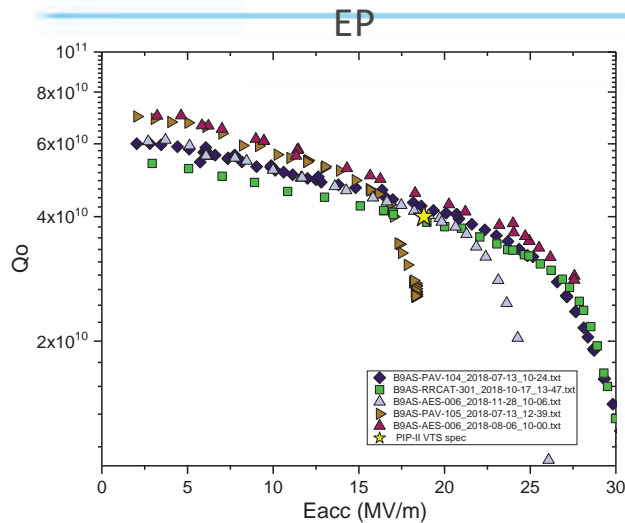
- All these treatments may exceed PIP-II specs:

- 7.6e10
- 4.7e10
- 4e10

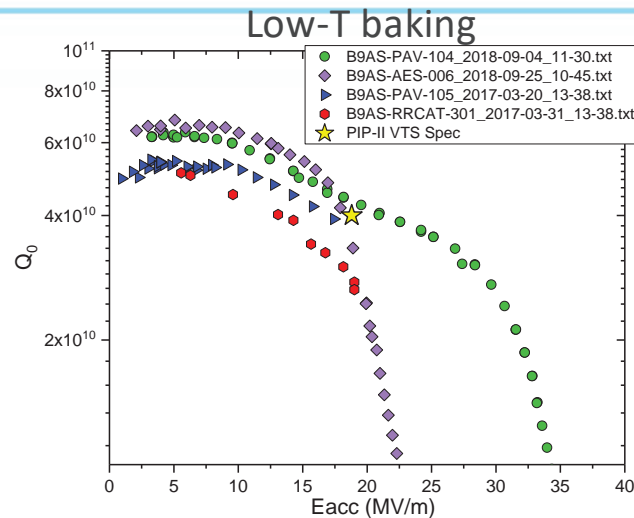
gives the highest Q-factor at medium field

All test done in compensated magnetic field

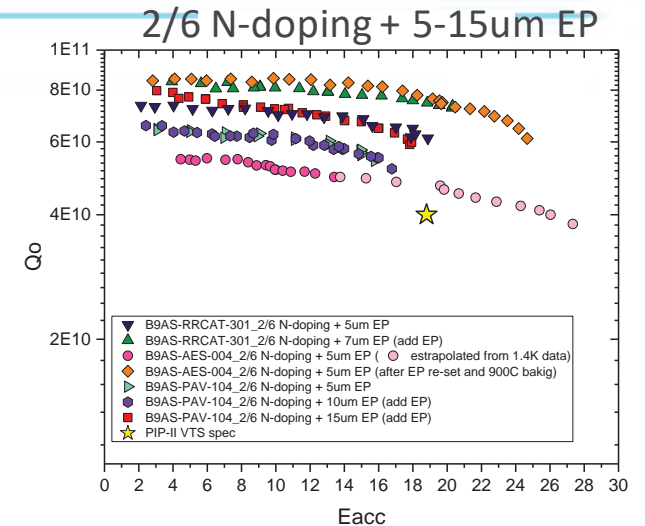
Surface treatments comparison in single-cell cavities



Q_0 at 20 MV/m between
 $\sim 3.7e10$ and $4.2e10$



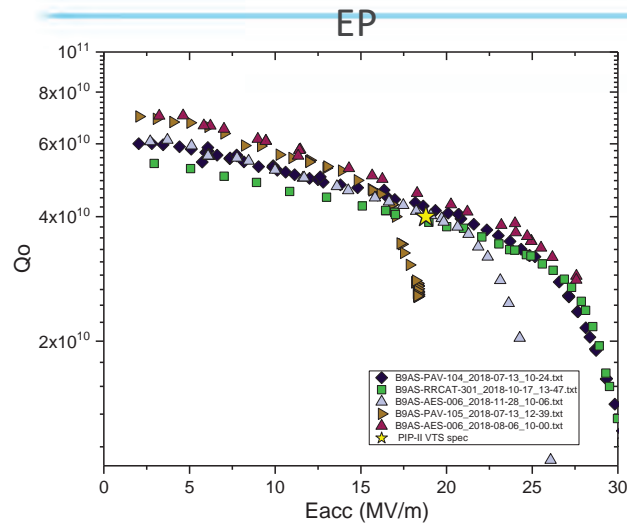
Q_0 at 20 MV/m between
 $\sim 2.5e10$ and $4.2e10$



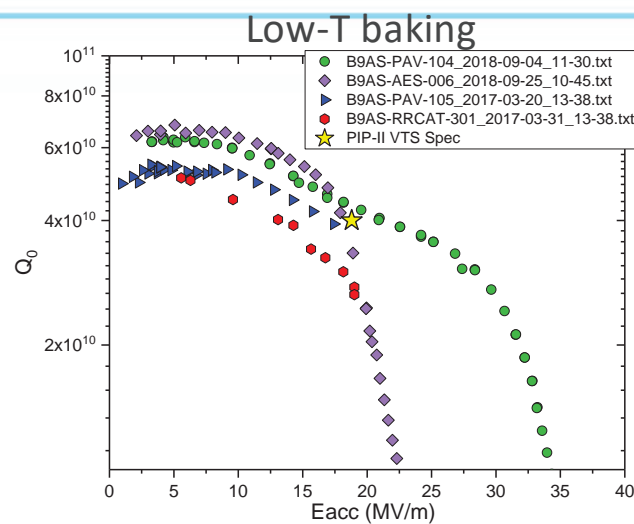
Q_0 at 20 MV/m between
 $\sim 4.5e10$ and $7.5e10$

All test done in compensated magnetic field

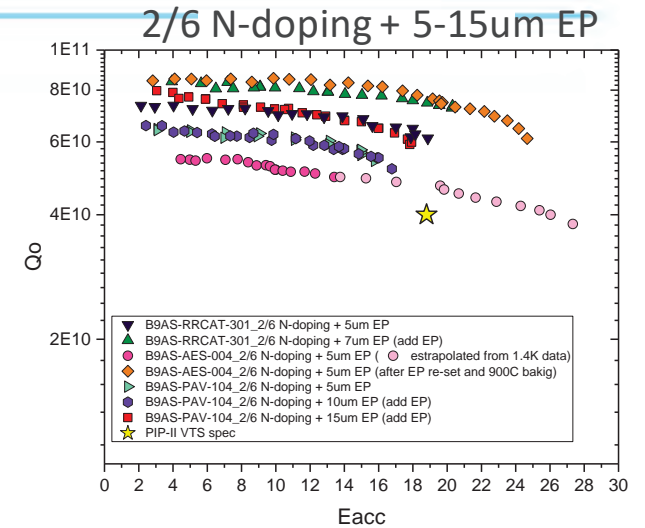
Surface treatments comparison in single-cell cavities



Q_0 at 20 MV/m between
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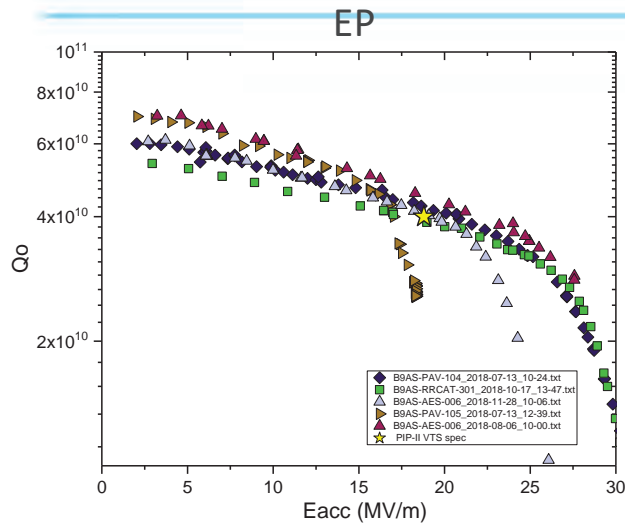
Q_0 at 20 MV/m between
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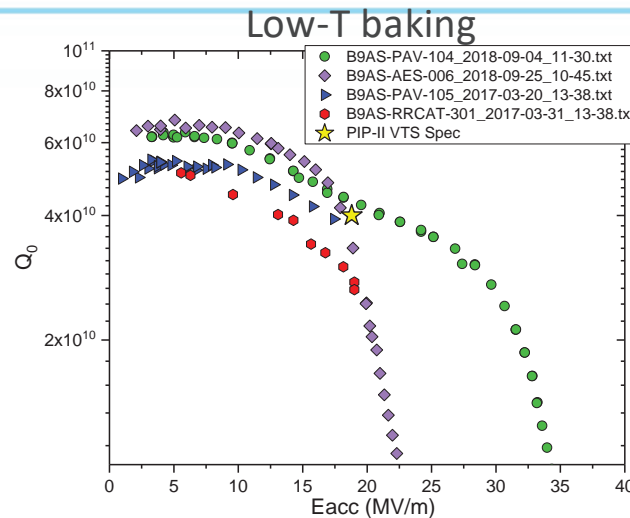
Q_0 at 20 MV/m between
 $\sim 4.5e10$ and $7.5e10$

Only N-doped cavities consistently exceed Q_0 spec

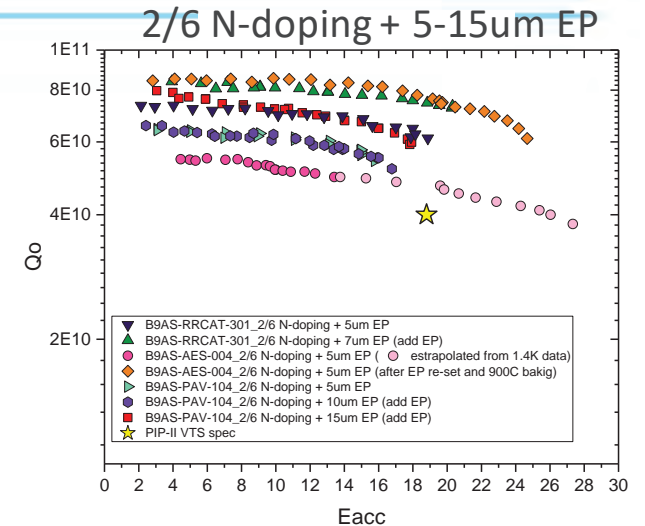
Surface treatments comparison in single-cell cavities



- Quench field between 17 and 30 MV/m
- HFQS onset starting from 16 MV/m to 27 MV/m

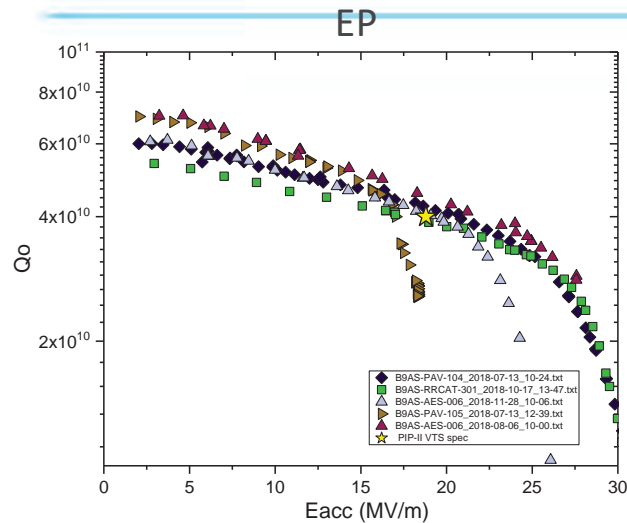


- Quench field between 18 and 35 MV/m
- HFQS onset starting from 17 MV/m to 27 MV/m

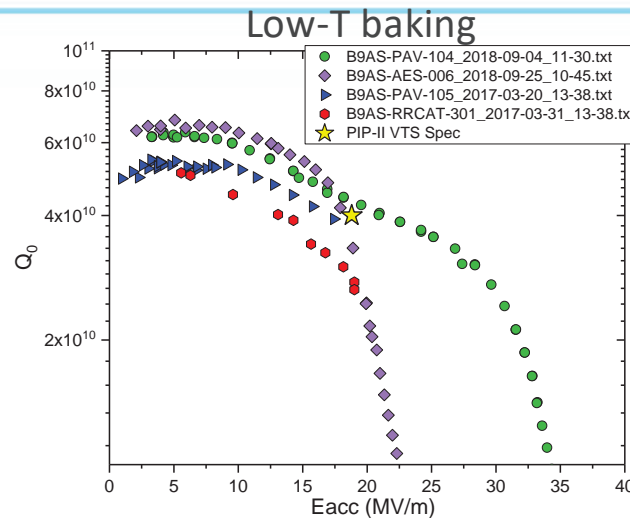


- Quench field between 16 and 28 MV/m
- No HFQS

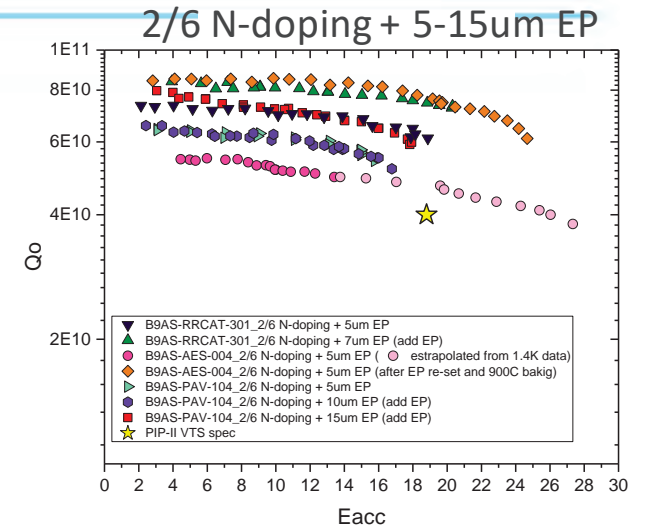
Surface treatments comparison in single-cell cavities



- Quench field between 17 and 30 MV/m
- HFQS onset starting from 16 MV/m to 27 MV/m



- Quench field between 18 and 35 MV/m
- HFQS onset starting from 17 MV/m to 27 MV/m

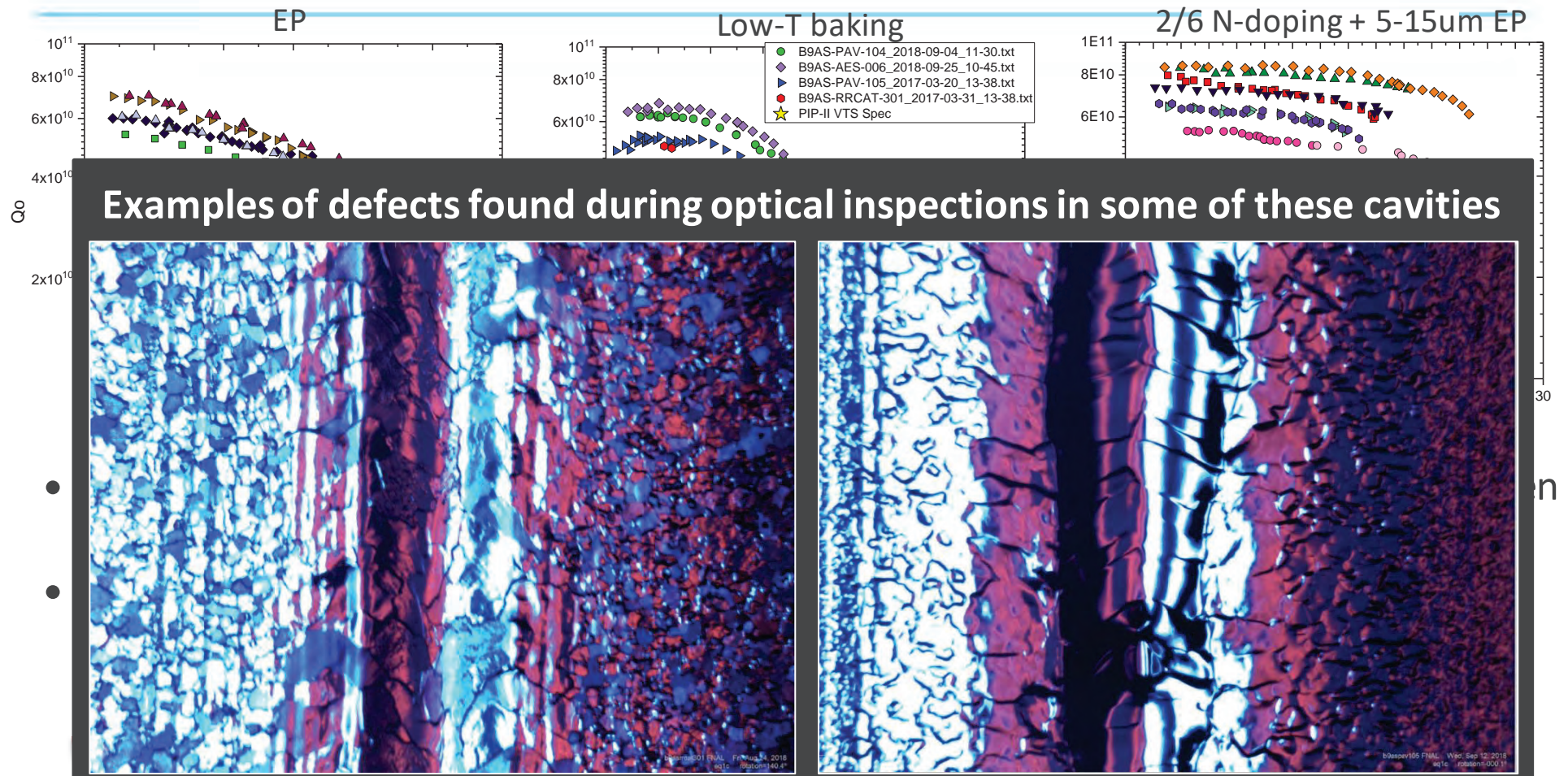


- Quench field between 16 and 28 MV/m
- No HFQS

In most cases early quench and/or HFQS onset is related to defects in the cavity, it is not a fundamental limitation

All test done in compensated magnetic field

Surface treatments comparison in single-cell cavities

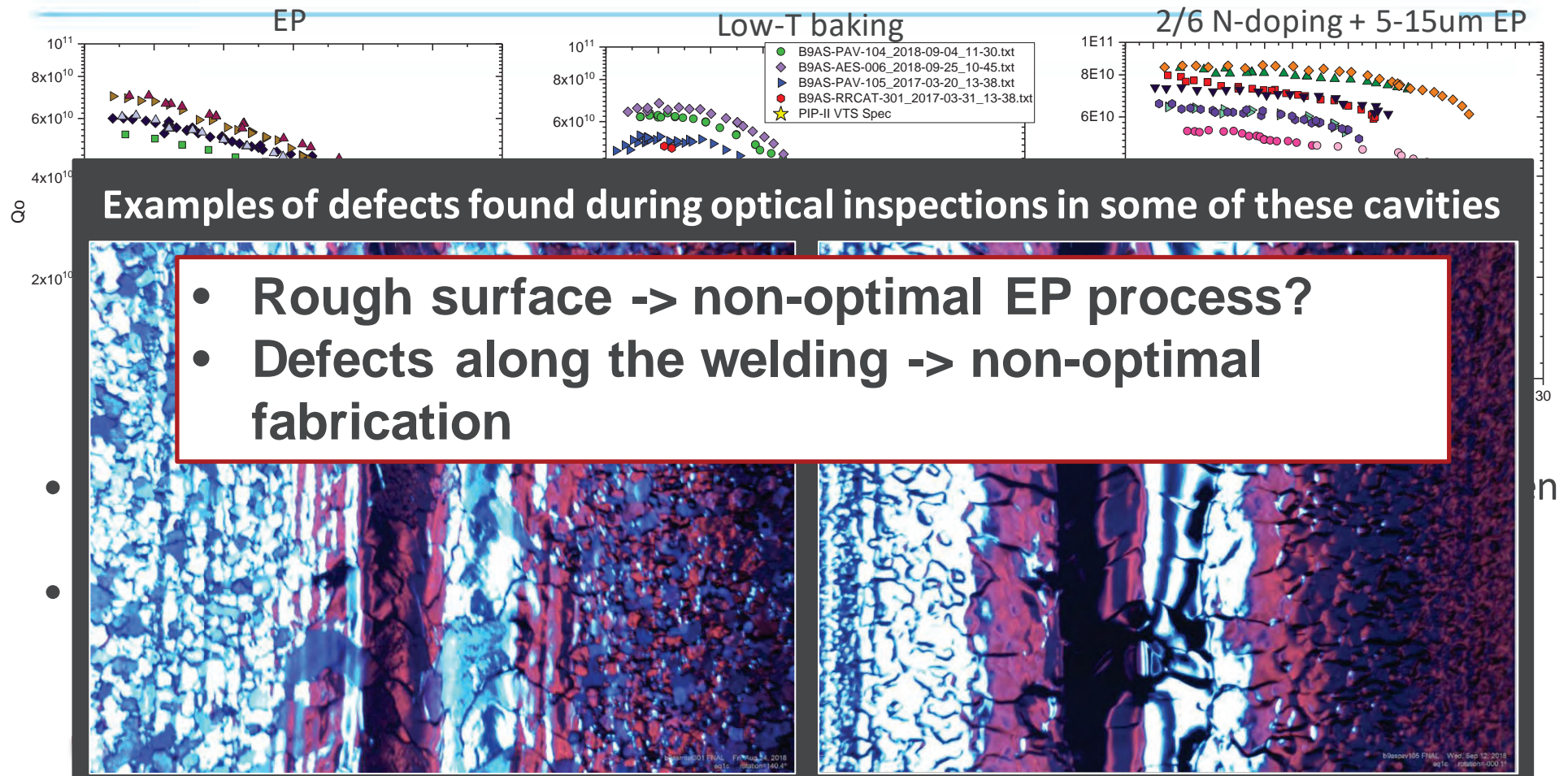


Examples of defects found during optical inspections in some of these cavities

In most cases early quench and/or HFQS onset is related to defects in the cavity, it is not a fundamental limitation

All test done in compensated magnetic field

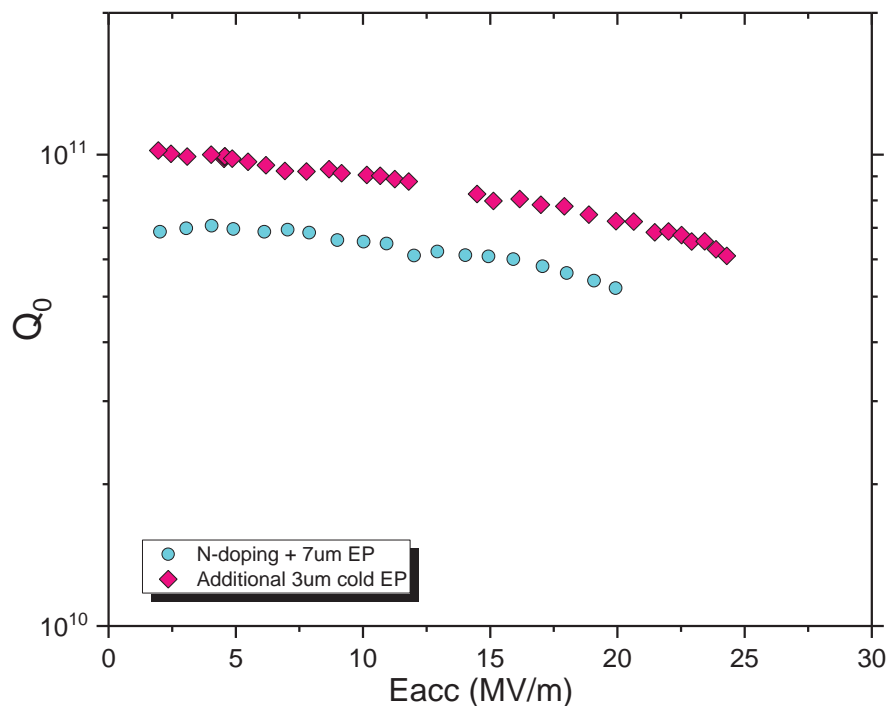
Surface treatments comparison in single-cell cavities



In most cases early quench and/or HFQS onset is related to defects in the cavity, it is not a fundamental limitation

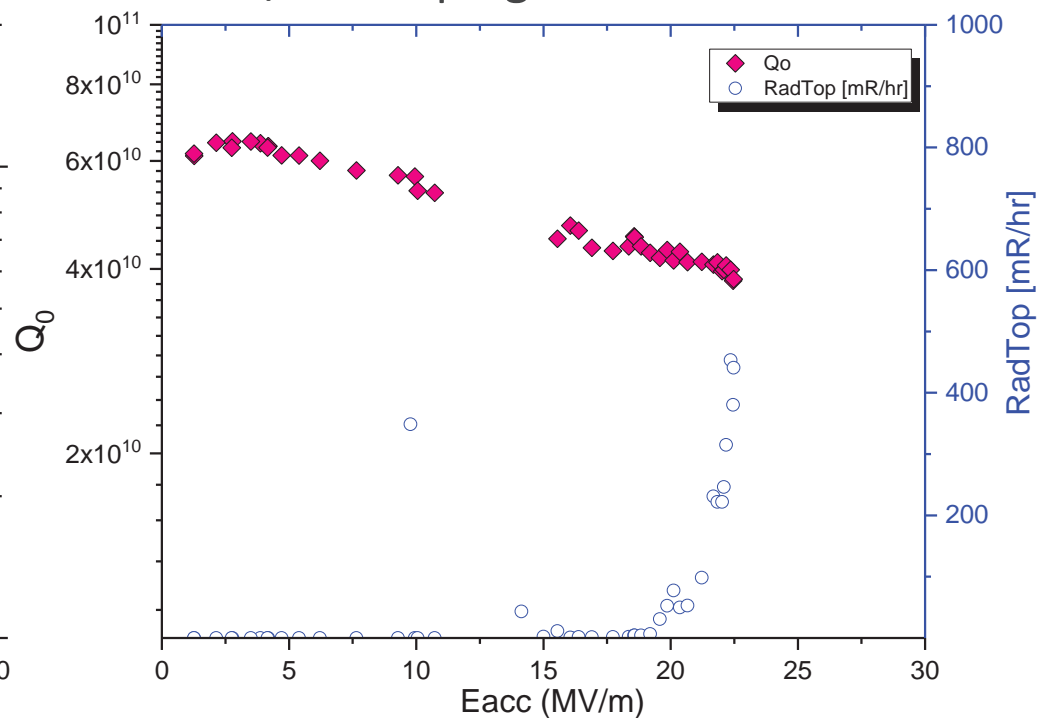
Cold EP on HB650 5-cell cavities

2/6 N-doping + 7um EP + 3um cold EP



Quench field improved from 20 to 24 MV/m by adding 3um of cold EP

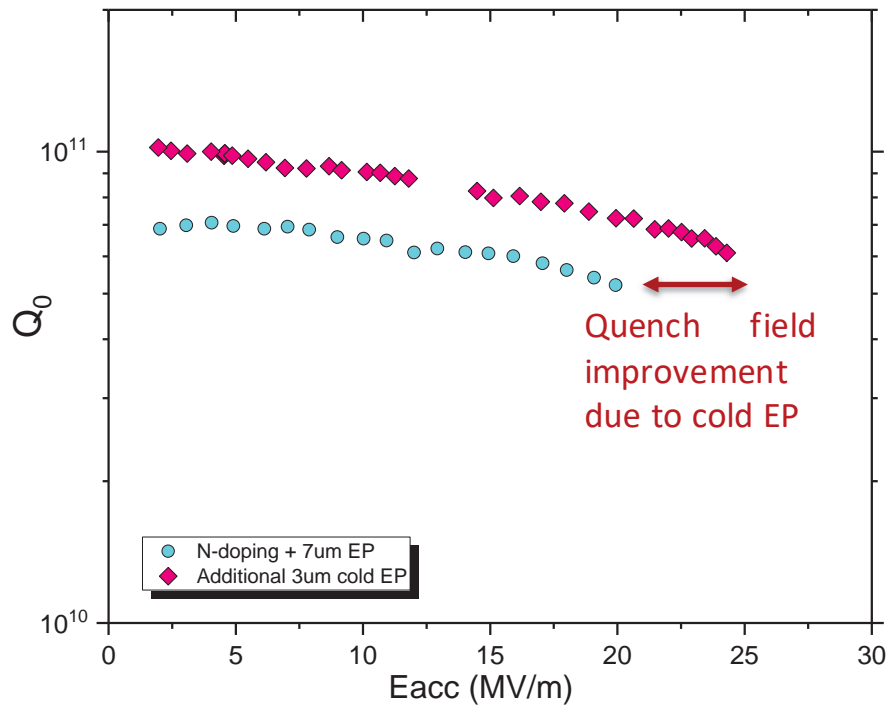
2/6 N-doping + 7um cold EP



Cavity limited by FE at ~ 23 MV/m, quench field may be higher

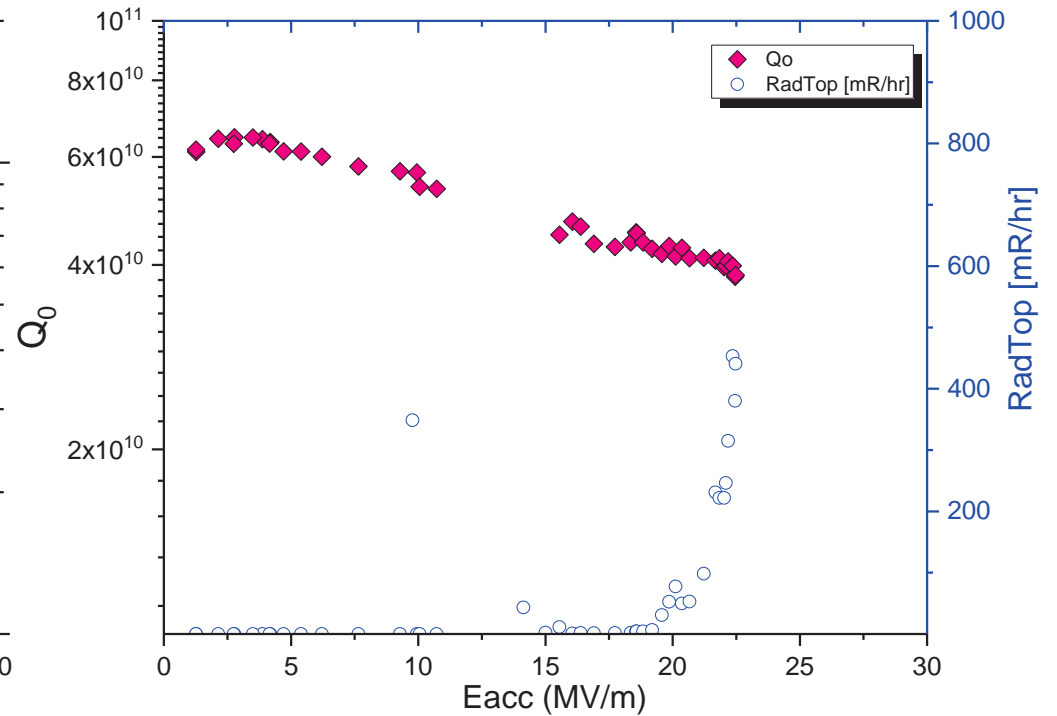
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2/6 N-doping + 7um EP + 3um **cold** EP



Quench field improved from 20 to 24 MV/m by adding 3um of cold EP

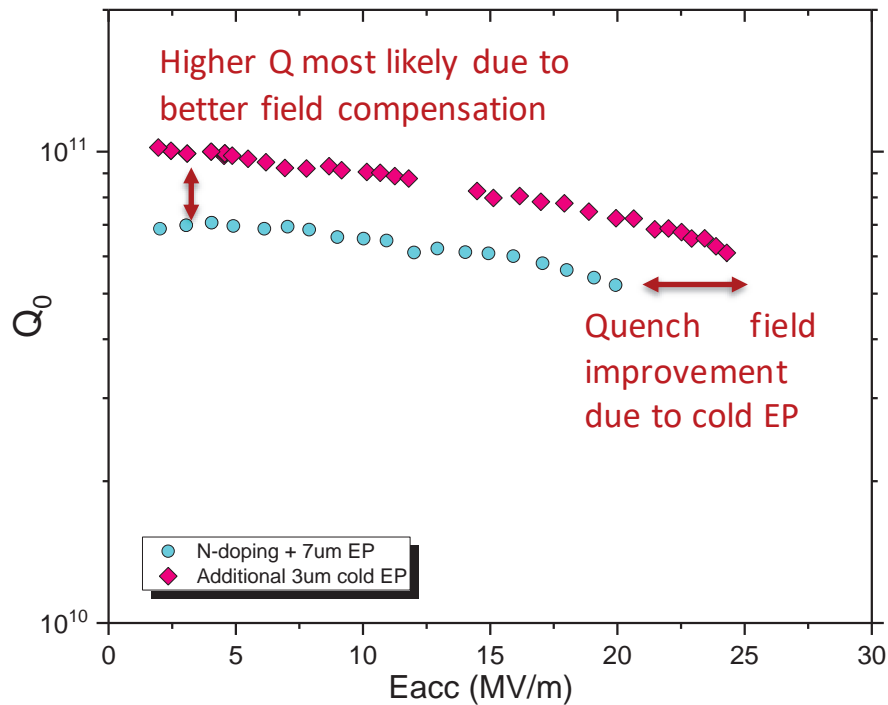
2/6 N-doping + 7um **cold** EP



Cavity limited by FE at ~23MV/m, quench field may be higher

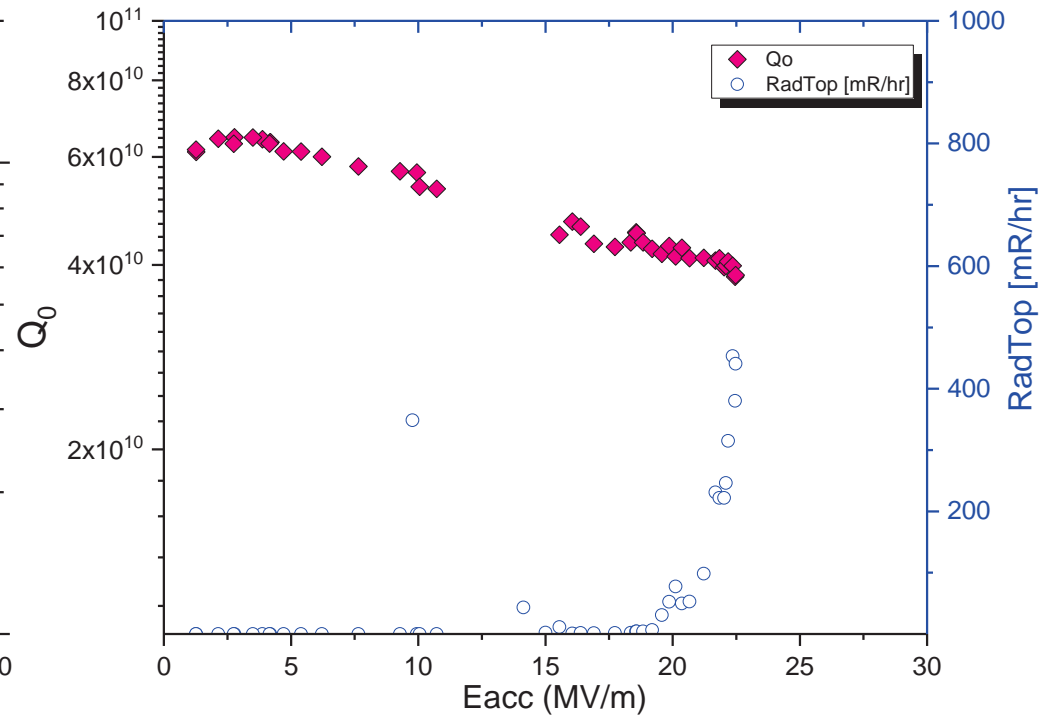
Cold EP on HB650 5-cell cavities

2/6 N-doping + 7um EP + 3um **cold** EP



Quench field improved from 20 to 24 MV/m by adding 3um of cold EP

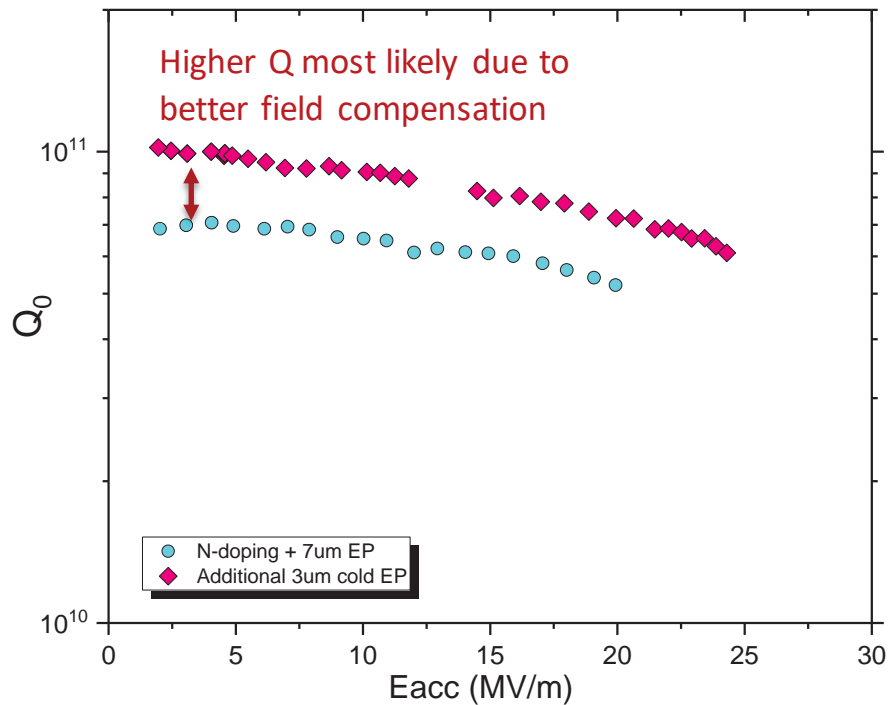
2/6 N-doping + 7um **cold** EP



Cavity limited by FE at ~23MV/m, quench field may be higher

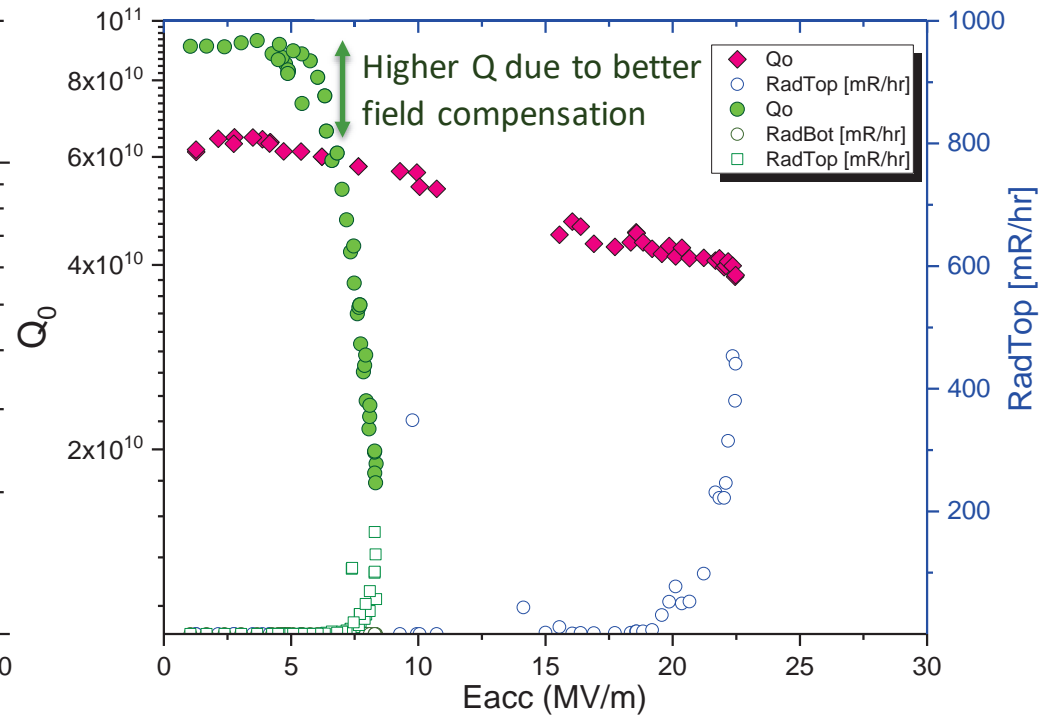
Cold EP on HB650 5-cell cavities

2/6 N-doping + 7um EP + 3um **cold** EP



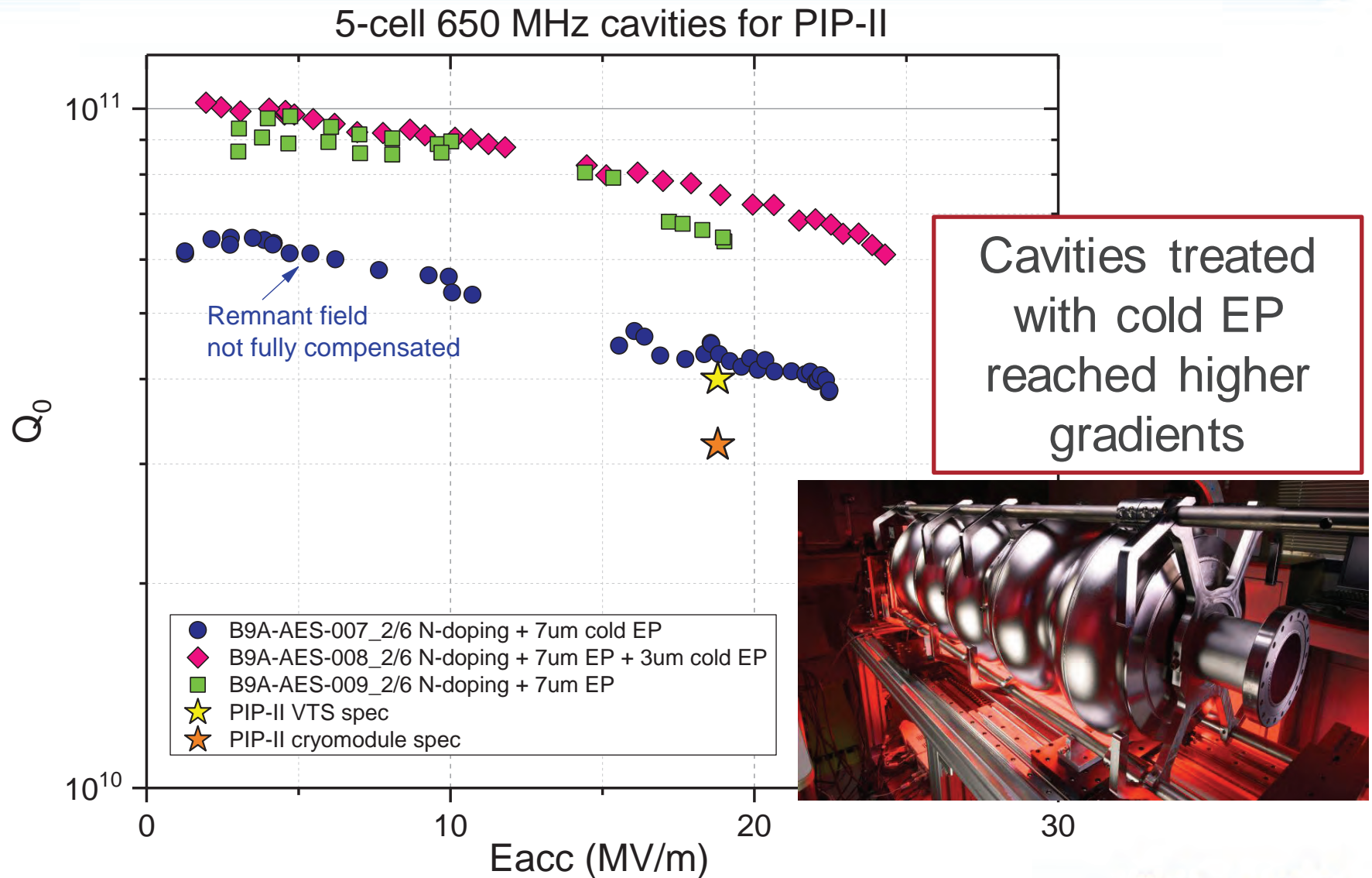
Quench field improved from 20 to 24 MV/m by adding 3um of cold EP

2/6 N-doping + 7um **cold** EP



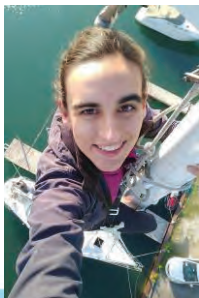
Cavity re-tested after HPR at Lab2 showed large FE starting from low field

Summary results on HB650 5-cell cavities

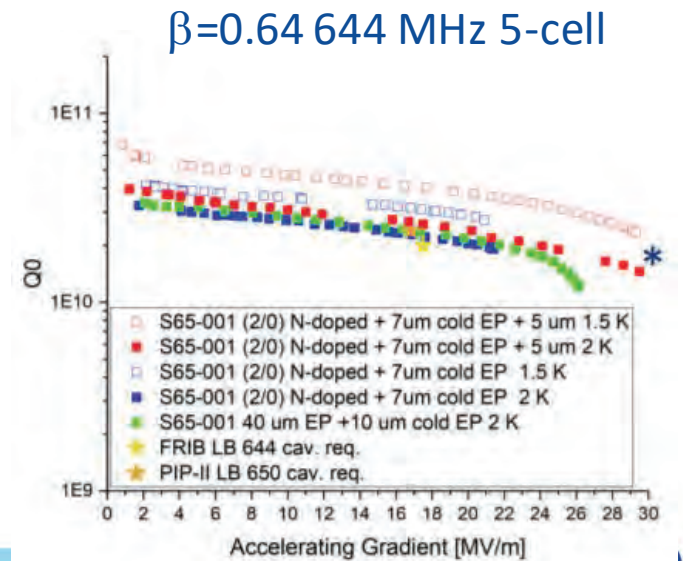
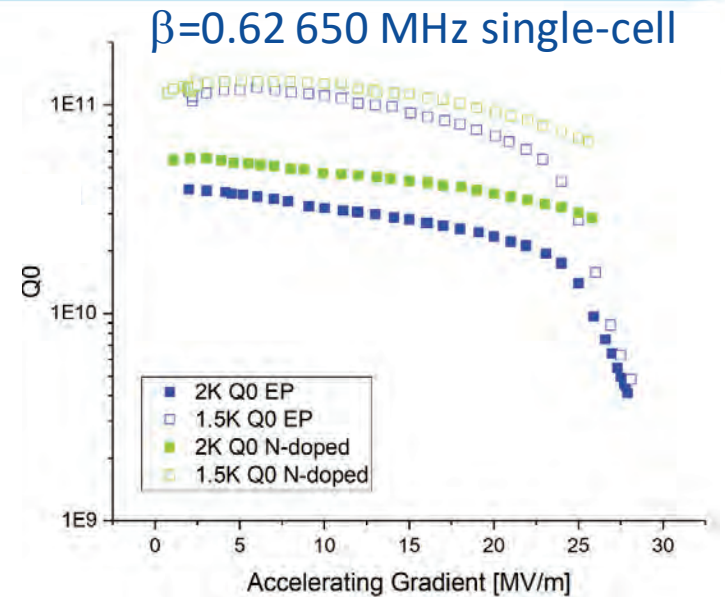


LB650 and 644 MHz cavities studies

- Collaboration between FNAL and INFN to optimize Q-factor on PIP-II LB650 ($\beta=0.62$) cavities, first results in N-doped single-cell cavity very promising
- Collaboration between FNAL, MSU and ANL through an Accelerator Stewardship Award focused on transferring the doping technology to 5-cell LB 644MHz cavities. Results also very promising!



Kellen McGee
PhD student from MSU now working at FNAL
Recipient of DOE Office of Science Graduate Student Research Fellowships



HWR and SSR1 pCM assembly



HWR cryomodule cold mass assembly



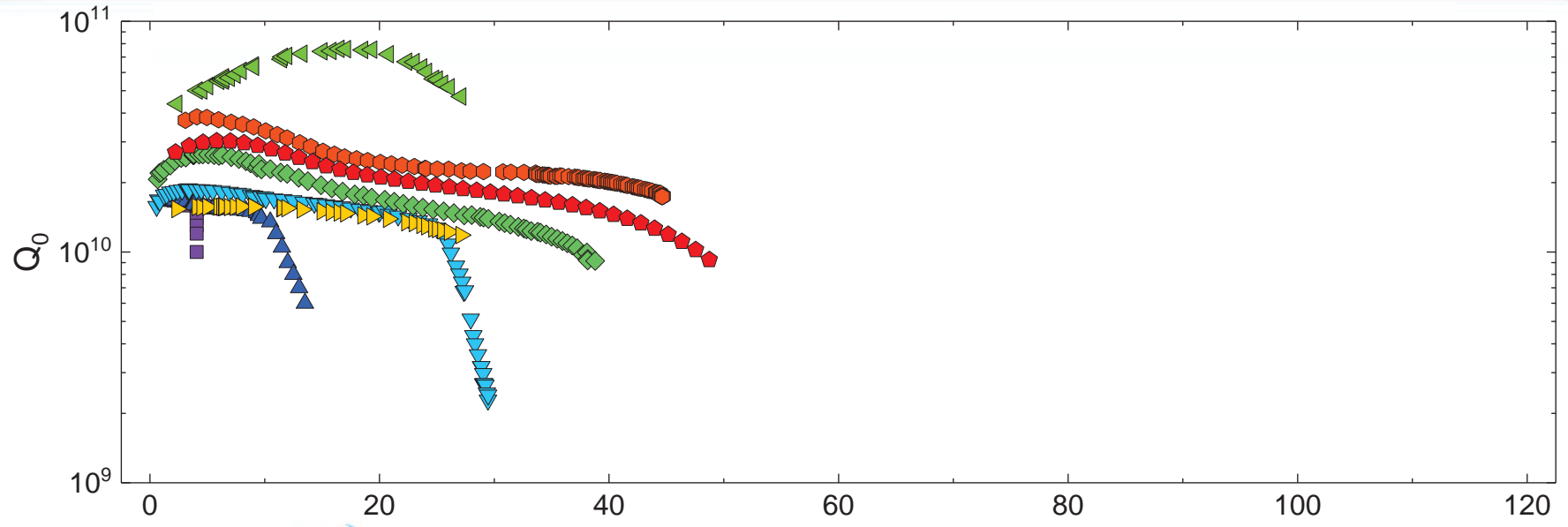
SSR1 cryomodule cold mass assembly



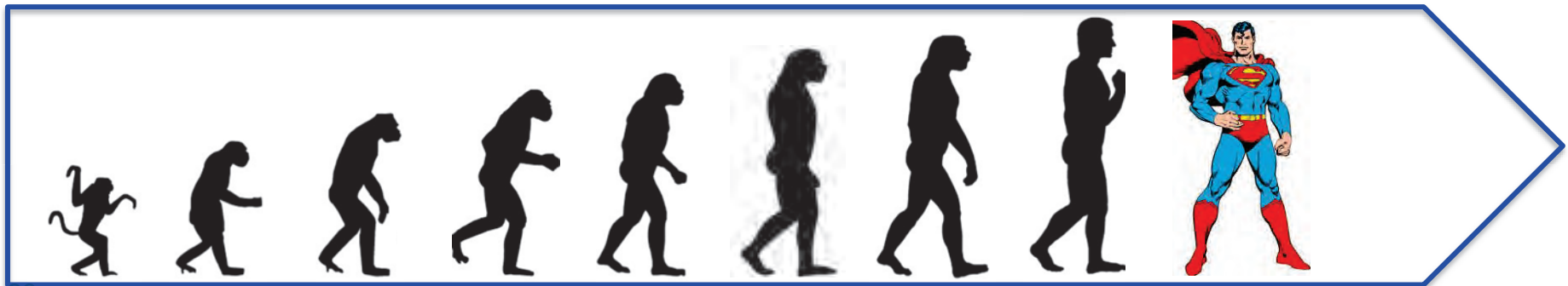
Outline

- Introduction to SRF cavities
- State-of-the art surface treatments
- Toward a better understanding
 - BCS surface resistance: new insights and optimization
 - Residual resistance: understanding and minimizing degradation due to trapped flux
- Technology improvement for SRF-based projects
- **Conclusions**

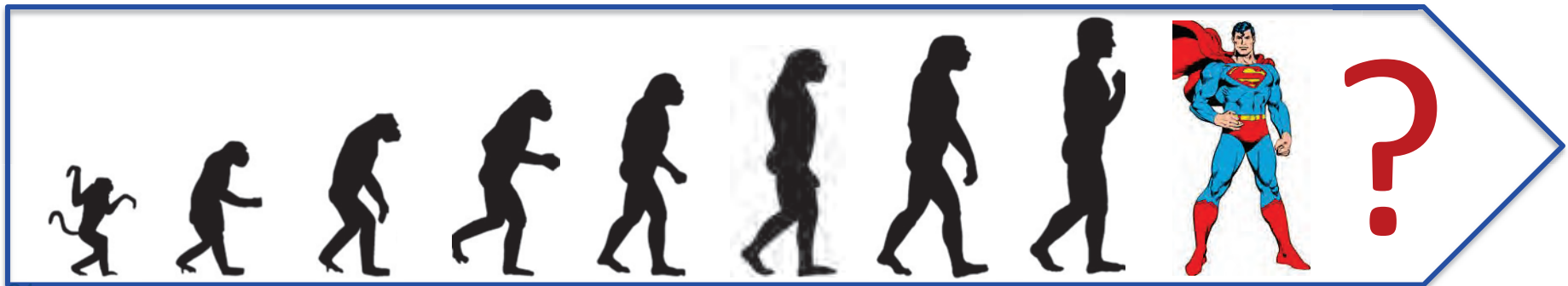
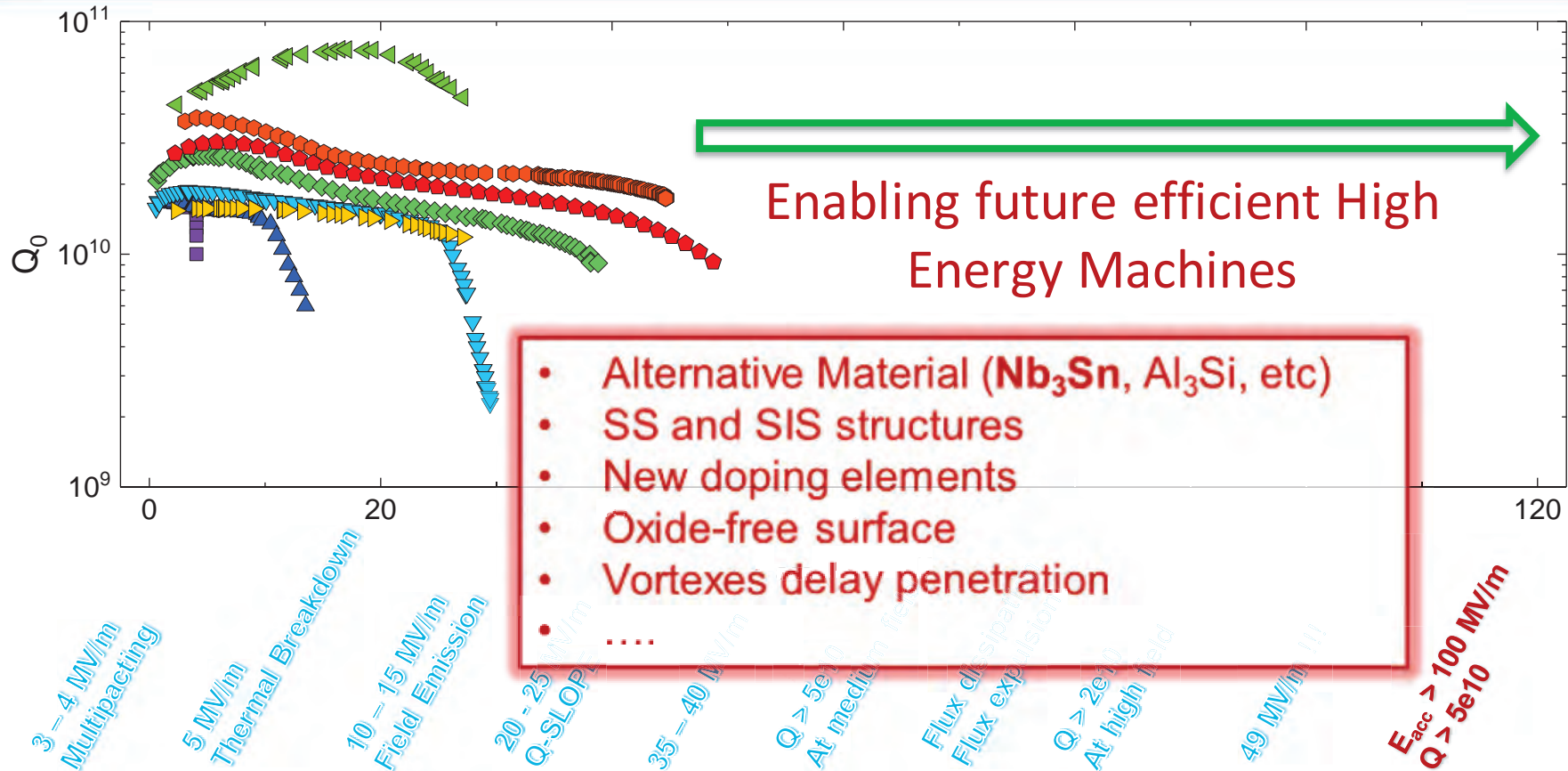
SRF performance: past, present and future



3-4 MV/m Multipacting
 5 MV/m Thermal Breakdown
 10-15 MV/m Field Emission
 20-25 MV/m Q-SLOPE
 35-40 MV/m
 $Q > 5e10$ At medium field
 Flux dissipation / Flux expulsion
 $Q > 2e10$ At high field
 49 MV/m !!!



SRF performance: past, present and future



Thanks to the whole FNAL SRF team!



Thank you for your
attention!